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## Distribution and Orientation of Steel Fibres in UHPFRC

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# **Distribution and Orientation of Steel Fibres in UHPFRC**

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**DCE Technical Report No. 151**

# **Distribution and Orientation of Steel Fibres in UHPFRC**

by

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April 2013

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## **PREFACE**

The technical report “Distribution and Orientation of Steel Fibres in UHPFRC” is a part of a larger research project entitled “Design of Transition Pieces for Bucket Foundations for Offshore Wind Turbines”. This report is divided into five numbered sections, and a list of references, situated after the last section.

The work within this report has only been possible with highly appreciated support and collaboration with Contec ApS, Technical University of Kaiserslautern and Fraunhofer ITWM, Kaiserslautern, Germany. We would personally like to acknowledge Mr. Bo Serwin from Contec ApS for supplying the CRC® material for testing, Mr. Franz Schreiber from Fraunhofer ITWM, Kaiserslautern, Germany, for his highly appreciated help with performing Computed Tomography (CT) scanning and analysis of the concrete samples, Mr. Frank Schuler and Dr. Jürgen Schnell, from Technical University of Kaiserslautern, Germany, for their consultative and advisory role to this report. Their contribution to this work is sincerely acknowledged.

Aalborg, April 11, 2013

Anastasia Nezhentseva, Eigil V. Sørensen, Lars V. Andersen

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# **1. INTRODUCTION**

## **1.1 Background**

Lately, the suction caisson (bucket) foundation has been introduced as the new technology for offshore wind turbine foundations. This concept, which is a hybrid of a monopile and a gravity-based foundation, is believed to be an economically attractive environmentally-friendly alternative to offshore wind turbines foundations used nowadays primarily in shallow waters. The bucket foundation has been widely used for installation of anchors in the offshore oil and gas industry in the North Sea for more than 30 years. However, the loading on the bucket foundations used for offshore wind turbines is more complex and will dramatically differ from the loading conditions (wave and current loading) observed on suction caisson foundations employed in oil and gas industry which are primarily subjected to compression–tension.

New generation of offshore wind turbines (with 6–7 MW already under consideration) will require taller and larger support structures. Consequently, exceptional structural strength and stiffness will be required for these structures in order to carry additional weight of the wind turbine, as well as to resist higher wind and wave forces acting on deeper water depths. In order to provide these characteristics, larger cross sectional diameters of the support structures will also become essential. This can, however, create difficulties with transportation of the wind turbine elements using public transportation system currently limited to transporting units of ~4.5 m in diameter [1] as well as lifting during installation. Development of segmented structural elements can help to solve both problems; however, it will create additional challenges at the joints between the individual segments. This together with other positive features discussed further can contribute to making concrete a competitive material for the production of the structural elements for offshore wind turbines. Concrete has many advantages such as high performance, low maintenance, high fatigue resistance and dynamic performance, low environmental impact, configuration, construction and mix design flexibility, as well as cost-competiveness [1]. Additionally, complex or curved concrete elements can be cast in-situ which can be an effective solution. Moreover, according to the specific strength requirements and application, the properties of concrete can be easily adjusted and utilized better (compared to e.g. steel) from normal structural grade to very high performance grade by using conventional concrete (plain, reinforced or prestressed), High Performance Concrete (HPC), Ultra-High-Performance Concrete (UHPC) or Ultra-High-Performance Fibre-Reinforced Concrete (UHPFRC).

The main challenge for the offshore concrete structures is the corrosive external environment due to seawater. Generally, lack of norms and internationally accepted standards for design of the structures made of special types of HPCs limits application of these materials for wind

turbine structures. Until now, concrete has already been used in the production of the following structures for onshore and offshore wind turbines: wind turbine foundations (on-shore, offshore gravity based foundations and pile caps (a large concrete block with heads of the piles cast into it) [1–3]), wind turbine towers made of prefabricated or cast in-situ reinforced, prestressed (onshore) [1,2] or UHPFRC (offshore) concrete elements [4], as well as high-strength grouted connections of monopile foundations [5]. An overview of international standards and guidelines as well as current design of concrete support structures for offshore wind turbines is discussed and presented, among others, by Klose [6]. As a conclusion, making project specific considerations and using engineering judgment, knowledge and experience was suggested while combination of standards should be avoided [6]. Moreover, dense concrete cover is normally required in order to protect the reinforcement steel against corrosion. For the bucket (caisson) foundations a special examination should be made of its floating stability during transportation and safety against uplift during installation at the construction site [6]. Comparison of the steel and concrete wind turbine towers for onshore and offshore applications is presented in [1,2].

## **1.2 Research problem formulation**

The focus of the current project is on the design of the structural element called “transition piece” (TP), connecting the offshore wind turbine tower to the bucket foundation. Transition piece structures for bucket foundations are currently made of triangular steel flange-reinforced shear panels (stiffeners or bracings) with flanges on top and sometimes with cutaways (circular holes) in the middle (for saving some extra material) as shown in Figure 1. These stiffeners are radially positioned and welded to the bucket lid. This solution is, however, believed to have several shortcomings. Production and corrosion protection of the steel TPs is rather expensive, labour-intensive and time-consuming due to the complex geometries and multiple welded connections. Moreover, when exposed to cyclic loading produced by the combined action of wind and waves, fatigue may occur at the welded joints of the TP.



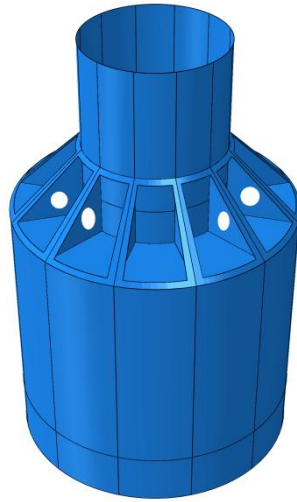


Figure 1 –Traditional steel design of the transition piece (TP) with stiffeners

Using concrete for building transition piece structures for bucket foundations can be an advantageous alternative solution to steel because:

- it responds well to fatigue caused by cyclic wind/wave loads;
- it requires little structural maintenance;
- impermeable concrete matrix (when using UHPFRC) protects the main reinforcement from aggressive sea environment, hence, allowing to save money on the anticorrosive maintenance painting required for steel structures;
- higher material damping and more tolerant of random resonance.

However, the expected large size of these secondary TP structures made of concrete (diameter  $\sim \text{Ø}18\text{--}20$  m and height  $\sim 9\text{--}25$  m, Nezhentseva et al. [7,8]) will require finding a reasonable solution for production of the TPs either as prefabricated units or in-situ. Achieving a cost-effective and competitive solution for the production of the TP capable of carrying and transferring large wave and wind loads onto the bucket foundation will require the use of alternative material solutions, such as steel/concrete hybrids or special types of high-strength concrete. Possibilities of using concrete/steel hybrid and UHPFRC for production of onshore and offshore wind turbine towers has been investigated and reported in the literature [1,4]. UHPFRC can be advantageous compared to steel for production of the transition pieces connecting an offshore wind turbine tower to the bucket foundation. Transition pieces made of UHPFRC could be a cost-effective technology with enhanced loading capacity, environmental resistance, and very low maintenance requirements, affecting directly the final energy price.

### **1.3 Focus and objectives of the current research**

When casting TPs in-situ, providing efficient vibration could be challenging. Therefore, it would be preferable to find a material with a good workability which can be cast without vibration. This will result in a solution with a high paste volume and relatively high autogenous shrinkage leading to fine cracks along the rebar negatively affecting load carrying capacity and durability. Avoiding these problems requires certain features of the UHPFRC material, i.e. providing a good bond between matrix and rebar for the load carrying system to work, and avoiding the surface cracks, which have been studied and documented in the present report. Moreover, a TP structure must transfer large stresses from the wind tower to the bucket foundation. Thus, the following UHPFRC properties are of special importance to the TP structures:

- 1) mix design vs. workability (to achieve a self-compacting material);
- 2) fibre distribution (homogeneity);
- 3) fibre orientation (minimize microcrack width for sufficient stress transfer between matrix and rebars, and for sufficient durability);
- 4) fibre type;
- 5) casting procedure.

The objective of the present study is to investigate workability and compressive strength of UHPFRC, to inspect the cracks using destructive testing method and visual observation, to study fibre distribution, concentration, orientation and presence of voids/cracks using a non-destructive technique – X-ray Computed Tomography (CT).

## **2. HPC, UHPC and UHPFRC: MATERIAL PROPERTIES**

### **2.1 The matrix material: HPC and UHPC**

Production of the taller, larger in diameter and more slender structural elements (towers, transition pieces, foundations) for the upcoming generations of the offshore wind turbines requires such material properties as high strength, durability, low maintenance, fatigue resistance, avoidance of resonance problems and resistance to the aggressive environmental impacts. In this respect, concrete has shown a promising potential and numerous benefits as a high performance structural material to be used offshore. Owing the exceptional challenging aggressive environmental conditions existing offshore and extra loads from combined wind, wave and current action, careful consideration should be given to alternative concrete mix designs that will have the above mentioned properties and that will be competitive with materials currently used for offshore structural elements.

Unreinforced concrete material is a complex non-homogeneous (i.e. heterogeneous) structure consisting of different phases (solid–cement paste plus aggregates, water and air filled voids or pores), all having various mechanical properties. However, in conventional structural design non-homogeneous microstructure of concrete is normally disregarded and considered continuous. When compression load is applied, differences in the mechanical properties of the aggregates and the cement paste material contribute to appearance of stress concentration, resulting further in formation of multiple microcracks in the interfacial transition zone (ITZ) between the hardened cement paste and the aggregate. When subjected to loading independently, cement paste and aggregates will demonstrate so-called brittle elastic behaviour, i.e. linear deformation followed by a sudden failure. Quite the opposite, concrete as a composite material will show a quasi-ductile behaviour characterized by increase of the load bearing capacity of the material beyond the linear elastic stage, primarily owing to microcracking phenomenon in the ITZ. In their study, Richard & Cheyrezy [9] have shown that the size of the microcracks is directly proportional to the diameter of aggregates.

High Performance Concrete (HPC) is a series of specialized concretes made of high-quality ingredients according to an optimized mixture design using a low ratio of water to cementitious material, leading to a higher strength compared to conventional concrete, i.e. a design compressive strength at 28 days of 70 MPa and higher [10]. The strength of the aggregates utilized in HPC is an essential factor affecting the final strength of a high-strength concrete. Moreover, adequate bond between the aggregates and the paste is also important. Using rougher angular crushed-stone aggregates rather than smooth and rounded gravel aggregates of the same size has been shown to provide a better adhesion between the paste and the aggregates, thus, resulting in higher compressive strength of HPC [10]. Furthermore, the optimum strength of high-strength concretes has been achieved at nominal maximum-size aggregates of 9.5–12.5 mm [10].

Compared to conventional concrete, HPCs have several advantages such as: high workability, high early strength, high ultimate strength, low permeability, chemical, freeze/thaw and abrasion resistance, high durability in severe environments, volume stability, compaction without segregation, reduced maintenance, inhibition of bacterial and mold growth, and extended life cycle. The main materials used in HPC include, see e.g. [10]:

- 1) Cement/cementing materials: Portland cement, fly ash, slag, and possibly silica fume;
- 2) Optimal graded aggregates – to improve workability;
- 3) Admixtures: superplasticizers (High-Range Water-Reducing Admixtures, synthetic surface active agents) – for extra flowability and for reducing the water/binder (W/B) ratio.

Like conventional concrete, HPC exhibits tensile strengths ( $\sim 5$  MPa, see e.g. [11]) which are only a small fraction of the compressive strengths, and, consequently, reinforcing steel bars (rebar) must be placed in the tensile zone of the HPC structural members. Normally, HPC has lower porosity, oxygen permeability, as well as a low W/B ratio of 0.20–0.35 by mass [10] than conventional concrete. Lower W/B ratio helps to avoid bleeding in HPC, whereas adding superplasticizers making the HPC mixture more fluid and increasing its workability contributes to a dense packing of the cement particles. However, some segregation can be observed in some HPC mixtures typically caused by excessive amounts of superplasticizers. For most HPC applications adequate workability is provided with a slump of  $\sim 200$  mm [10,12] with majority of HPCs placed at slumps of 180–220 mm with some vibration applied [10].

Adding ultra-fine particles of silica fume (microsilica powder) to a concrete mixture can help to achieve a very dense packing of the particles in the binder phase resulting in a very high compared to HPC compressive strength ( $\geq 150$  MPa, [13–15]). This new concrete material having also a more outstanding durability than HPC was called Ultra High Performance Concrete (UHPC). UHPC has normally a low W/B ratio of 0.18–0.25 [10]. Typically, smaller and stronger aggregates are used in UHPC compared to HPC, with the maximum aggregate size of  $\leq 7$  mm [15,16], or even sometimes  $\leq 2,5$  mm according to the Japanese recommendations [14], which can considerably decrease the size of microcracks forming in the ITZ when the load is applied [9]. Moreover, UHPC is a more homogeneous material due to the smaller difference between the stiffness of the aggregates and the cement paste.

A cement-based UHPC material called DSP (“Densified Systems containing homogeneously arranged, ultra-fine Particles”) developed in 1978 by H.H. Bache and his colleagues at Aalborg Portland, Denmark, has even a smaller water/cement + fine powder ratio of 0.12–0.18 by weight [17]. This UHPC made with ordinary natural aggregates (quartz sand  $\sim 0,1$ – $1,5$  mm and crushed granite of  $\sim 2$ – $5$  mm) has a compressive strength of 120–160 MPa or even up to 180–200 MPa, when utilizing exceptionally strong natural aggregates (see e.g. Bache, [18]). Reaching even higher compressive strengths (typically in the range of 220–270 MPa, [18]) is possible using stronger aggregates, such as e.g. calcined bauxite. Tensile strength of the UHPC lies within  $\sim 6$  MPa [11]. However, UHPC has a very brittle behavior which is generally unacceptable in structures. In order to avoid this problem, various types of reinforcement in the form of fibres, bars and wires can be incorporated in the UHPC matrix. This will assure high fracture energy absorption and, as a result, greater ductility of the structural members.

## 2.2 The matrix material: UHPFRC

In order to fully utilize the high compressive strength of HPC – and UHPC – a large amount of steel is required in the tensile zone of a structural member. This may lead to small distances between neighbouring reinforcing bars, and to cracking and spalling of the brittle UHPC material in the tensile zone during loading of the member. To avoid that situation to occur the mortar fraction of the UHPC may be reinforced by addition of steel fibres which drastically increases the ductility of the material – which is then termed Ultra High-Performance Fibre-Reinforced Concrete (UHPFRC). Having an insufficient structural ductility, large unreinforced UHPFRC can demonstrate a brittle behaviour characterized by appearance of local cracks and sudden failure. Solution to this problem is adding a higher dosage of fibres as well additional main reinforcement (conventional or high-strength) to the section. This has been implemented, for instance, in a Compact Reinforced Composite (CRC®), also called “The New Concrete”, invented by Bache at the Cement and Concrete Laboratory of Aalborg Portland A/S in 1986 [17–19] and subsequently produced by Hi-Con A/S. Different other types of UHPFRC currently presented on the market are described in details by, e.g. Spasojević [15].

Advantages of UHPFRC matrix material and reinforced UHPFRCs:

- provides higher compressive and tensile strength, durability, and increased tensile ductility (fibre dependent) compared to UHPC;
- allows utilization of 5–10 times more main reinforcement (compared to conventional concrete), thus, resulting in much stronger structures;
- minimization of the overall wall thickness due to small cover thickness (5–15 mm vs. 50 mm for the conventional concrete) and dense arrangement of the rebar in order to provide the necessary degree of reinforcement for the loads in question;
- has a low chloride permeability due to highly dense microstructure with possible “clogging” effect over time;
- initially expensive formwork can be reused over long production of structural elements, therefore, giving lower costs per unit.

Until now, UHPFRC has been primarily used for production of slender structures (balcony slabs, staircases, walkways), pavements, industrial floors, cargo ships [20,21], architectural concrete and slabs-on-grade, in-situ cast joints between prefabricated structures, and strengthening steel bridges with strong, ductile steel–UHPFRC hybrid constructions [22]. In the onshore and offshore wind industry ultra-high performance grouting mortars (e.g. Ducorit® [23] and MASTERFLOW®9500 [24–36]) have been applied for grouted connections between transition pieces and monopiles, tripiles, tripods and jacket foundations resulting in high bearing capacity and

very good corrosion protection. Prefabricated UHPFRC structures can be joined in-situ with the so-called CRC® JointCast which, compared to conventional joints, has a smaller joint width and an anchorage length of ~5–10 times the diameter of the reinforcement bars [5]. This technology can be used for joining prefabricated elements made of UHPFRC, conventional concrete, as well as steel.

The above mentioned outstanding material properties of the reinforced UHPFRCs are of special importance to the offshore TP structures for bucket foundations. Therefore, the following concerns related to the production of the TPs made of UHPFRC are being addressed in the present project:

- Fabrication: providing adequate workability of the mixture;
- Transfer of (large) stresses in the structure: fiber distribution (to provide homogeneity) and fibre orientation (to minimize microcrack width for sufficient stress transfer between the matrix and the rebar, and for adequate durability).

Different types of binders can be used in UHPFRCs, such as Densit® (invented by Bache in 1978, [17–19] and further utilized in steel rebar reinforced CRC®), BSI-Ceracem© by Sika-Eiffage [28] and Ductal® by Lafarge and Bouygues [29–31], among others. In general, the binder has lower effect on the compressive strength of UHPFRC compared to the type, size, composition and quality of the aggregates, as well as adding additional fibres. According to Contec ApS, a supplier of Secutec 6 binder for UHPFRC, using bauxite as an aggregate allows to reach a compressive strength of 180-240 MPa depending on the water content, compaction, amount and type of fibres etc.

Generally, enhancement of the strength and durability of UHPFRC can be partially explained by the greater homogeneity of its matrix material microstructure compared to conventional and high-strength concretes. A further insight into the composition and the material properties of UHPFRC is given on the example of CRC® composite material with high concentration of main steel reinforcement, based on the Densit® matrix material [17–19]. An extremely ductile CRC® has a high potential to be used in structures with large dimensions (with spans of 500–1000 meter) exposed to strong compressive and tensile forces (see, e.g. [22]). The higher packing density of Densit® binder particles (e.g. cement and microsilica) compared to a conventional concrete binder (70–75 vs. 30–50 % per vol., see Bache [18]) provides a much stronger fixation of the fibres, which, together with a higher percentage per volume of main reinforcement and fibres in the CRC®, leads to a different mechanism of fracture behaviour. When subjected to tension, a local fracture zone with large visible cracks will appear in the tension zone of the element made of conventional concrete. However, a global fracture zone with small multiple microcracks propagating throughout the element will be observed in CRC® with no visible cracks

in the tension side [22]. Moreover, tests on CRC® panels performed with grenade explosives showed that structures made of CRC® can well resist very heavy impact loads, e.g. earthquakes.

CRC® matrix has typically a water/binder (W/B) ratio of 0.13–0.18 and a large content of microsilica (see, e.g. [32]). Lower W/B ratio in the CRC® matrix means that there is a large number of unhydrated cement particles which will react with the water in cracks, thereby, helping to heal them [32]. Moreover, the CRC® matrix is very dense with a low porosity of ~ 1.5 % and capillary porosity of ~ 0.2 %, and extremely low permeability [33]. The results of tests performed on loaded beams in salt water showed that the rate of chloride penetration decreased over time due to a “clogging” effect. No corrosion was observed even when adding salt in the mixing water because of the lack of capillary porosity limiting transportation of oxygen and water [33]. Furthermore, closing and filling up of the cracks with hydration products was observed after exposing the beams preloaded up to yielding with noticeable cracks to salty water environment. The unhydrated cement particles reacted with the water entering the crack and protected it from rebar corrosion. This phenomenon was confirmed after 3–4 years of exposure and various testing conditions at Force Institutterne in Denmark and Instituto Eduardo Torroja in Spain [33].

CRC® matrix combines large contents of 0,15×6 mm or 0,4×12,5 mm steel fibres (typically 2–6 vol. %) and closely spaced main reinforcement made of high strength steel, traditional rebar, carbon etc. (5–20 %). Adding fibres to concrete can increase its tensile strength, decrease micro cracks, and help to control cracking caused by drying (or autogeneous) shrinkage. Normally, the CRC® matrix contains aggregates with a diameter up to 1/3 of the space between the main reinforcement. A cover layer of 10–15 mm and a comparable distance between individual rebar are used in CRC®. Compared to conventional concrete, 2–5 times more reinforcement can be utilized in CRC® [18]. In this respect, structures made of CRC® will have mechanical properties comparable to steel structures, but weigh only about a half [22]. Moreover, such typical problems affecting durability of the steel structures such as corrosion and fatigue are not of concern for structures made of CRC®. Material properties of the UHPFRC Densit® matrix and the heavily reinforced CRC® are summarized in Table 1.

Table 1 – Properties of UHPFRC Densit® matrix material and reinforced CRC® developed by Bache et al. at Aalborg Portland, Denmark (adapted and translated from Table 1.1, see Bache [18]))

|                            | UHPFRC                             |                |                   |
|----------------------------|------------------------------------|----------------|-------------------|
|                            | concrete/matrix material (Densit®) |                | CRC®              |
|                            | 0 – 2% fibres                      | 4 – 12% fibres |                   |
| Compressive strength, MPa  | 120–270                            | 160–400        | 160–400           |
| Tensile strength, MPa      |                                    |                |                   |
| Flexural strength, MPa     | 6–15                               | 10–30          | 100–300           |
| Shear strength, MPa        |                                    |                | 100–400<br>15–150 |
| Density, kg/m <sup>3</sup> | 2500-2800                          | 2600–3200      | 3000–4000         |
| E-modulus, GPa             | 60–100                             |                | 60–110            |

### 2.3 Uniaxial tensile response of UHPFRC

As mentioned previously, steel fibres increase fracture energy and ductility, compressive and tensile strength while reducing the tendency for cracking. Adding 2 % steel fibres by vol. will provide satisfactory increase in the tensile strength of the UHPFRC matrix, as reported by Markovic [34]. Large contents of fibres might, however, have a negative effect on workability and homogeneity of the mix. In general, adding steel fibres to UHPC matrix will results in a tensile behaviour which can be divided into three main stages and shown in Figure 2 (see e.g. Spasojević [15], Li and Fischer [35]):

1) *Linear-elastic response* – This stage takes place until the point when the stress level in the UHPFRC matrix corresponds to the matrix tensile strength and a first microcrack appears.

2) *Pseudo strain hardening* – During this stage no real plastic microstructural changes happen, therefore, it is also called “*pseudo plastic*”. On the contrary, deformations continue to increase intensely as a result of formation of the numerous tiny cracks (or microcracks ~ 1 µm, [15]) in the matrix, whereas the uniaxial tensile stress does not change a lot, or increases considerably slower compared to the linear-elastic stage. The process of multiple microcracking with uniformly distributed openings tied together by fibres in UHPFRC is similar to strain hardening or plastic behaviour, thereby, giving the name to this stage. The stage continues until the maximum deformation is achieved (normally ~ 2–3 ‰, depending on the fibre content and properties [15]). Macrocracks can also start to develop during this stage.

3) *Strain softening* – this stage starts when one of the sections of the matrix (the weakest) fails to transfer the average stress of the same intensity, meaning that the strain hardening capacity of UHPFRC matrix is reached. During this stage local deformations in the weakest section



occur, leading to macrocracks, along with general decrease of the stresses and deformations taking place in other parts of the matrix. Moreover, the macrocracks will start opening governed by the fibre action. This will lead eventually to reaching the deformation capacity, normally addressed as half the length of the longest fibre [15].

In the FEM analysis performed in ABAQUS by Nezhentseva et al. [7,8] the behaviour of the CRC® matrix material was modelled separately for the Densit® matrix and the steel reinforcement bars using the Damaged Plasticity Model [36]. This was performed in order to model more accurately the development of stresses and strains in various parts of this composite material. The distribution and orientation of steel fibres in the CRC® matrix was modelled uniform in all directions, i.e. isotropic. Due to the lack of information available on the tensile behaviour of the CRC® matrix, tensile properties of a comparable UHPFRC material called BSI-Ceracem© (by Sika-Eiffage) with 0,16×20 mm steel fibres 2.4% vol. were taken from the results of the experiments carried out by Redaelli and Muttoni [28] (as summarized in Table 3, Nezhentseva et al. [7]).

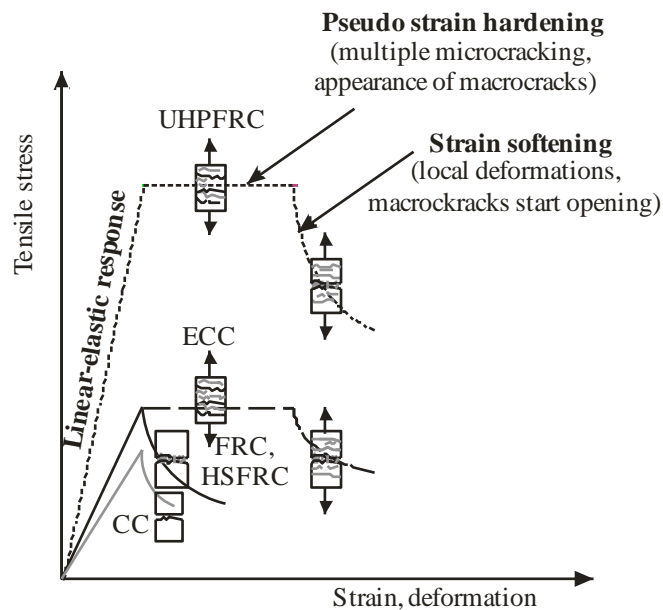


Figure 2 – Response of UHPFRC in uniaxial tensile stress state compared to conventional (CC), fibre-reinforced normal or high-strength concrete (FRC, HSFRC), as well as engineered cementitious composites (ECC) with short polymeric (PE, PVA) fibres (adapted from Spasojević [15], Li and Fischer [35])

In accordance with the CRC® Patent [37], the concept of cracks can be divided into two categories: surface cracks and cracks passing/not passing the reinforcement. The definition “reinforcement” is used for the main reinforcement (such as steel bars), while the CRC® material also contains matrix reinforcement in the form of fibres which reinforce the base matrix. Thus, there are distinguished two possible types of cracks passing (or not passing) the main reinforcement [37]:

- 1) *a base matrix crack* – passing the fibres, which are still able to transfer the load across the crack.

2) *a complete crack* – when the fibre-reinforced matrix does not transfer any tensile forces due to the fibres in the matrix being either broken or pulled out;

Moreover, passing the main reinforcement a crack of a size smaller/equal (at the worse) to the distance between the very fine, rigid and strong steel fibres can still be capable of transferring the tensile loads due an increased combined strain capacity of the base matrix material, fibres and main reinforcement. For this reason CRC® is also known as a material with a fine crack-free behaviour even at very large loads [37]. Short fibres can control and bridge microcracks around the reinforcement, while long fibres can be beneficial when bridging large macrocracks appearing at higher loads. On the other hand, long fibres can potentially cause workability problems due to “balling”.

Fibre orientation is known to have a large effect on tensile and flexural characteristics of UHPFRC. The results of the research on the fibre orientation in UHPFRC carried out by Barnett et al. [38] and supported by X-ray CT imaging, has shown that fibres tend to align perpendicular to the flow of the concrete. Moreover, they have addressed that the behavior of the longer fibres, commonly used in UHPFRC, can be expected to differ from the behavior of the shorter fibres. Besides, considerable differences in fibre orientation based on the ways of casting of UHPFRC, either parallel or perpendicular to the flexural tension, were discovered by Kim et al. [39]. The best distribution and orientation of fibres has been achieved for the samples cast in the flexural tension direction. It has also been confirmed that for larger flow distance from the points of casting the fibres are likely to align perpendicular to the flow. Research done by Pansuk et al. [40] based on the analysis of the fibre orientation numbers has showed, however, a conflicting tendency. It has been observed that the fibres tend to arrange themselves in the direction parallel to the concrete flow. This effect has been, however, more obvious for the longer fibres and longer flow distances.

Motivation of the present study is, therefore, to assess the fibre orientation and to check the validity of the above mentioned observations based on the CRC® samples. Desired outcome and purpose will be as follows:

- Uniform fibre distribution, fibre orientation and fibre volume in the CRC® specimens;
- No air pockets and adequate fibre orientation around main reinforcement bars;
- Reduction/elimination of surface cracks by use of shorter fibres and/or various ways of casting (based on naked-eye observations of the specimens);
- Recommendations on surface crack protection in case of cracks observed.

### **3. UHPFRC SPECIMEN PREPARATION**

#### **3.1 Matrix material, mixing and casting**

The focus of the work in the present section was to provide the following characteristics of the UHPFRC mixture and samples:

- Satisfactory workability of the mixture;
- Sufficient compressive strength.

Workability of concrete is highly affected by the amount of water added to the mixture. For UHPFRC, however, increase of the w/c ratio is not possible, since it will worsen the properties of the final product. Therefore, sufficient workability of paste phase is achieved by using large dosages of superplasticizing admixtures. Additionally, workability is also affected by the shape, angularity and surface texture of the particles of the mixture (i.e. fine and coarse aggregates). Geometric properties of the particles influence the particle size distribution of the fine and coarse aggregates in the mixture (“grading”). Generally, smooth and rounded particles will tend to pack together better than rough angular or subangular ones, thus, assuring the best workability. Typically, particles of different sizes are used in concrete in order to ensure better filling of the spaces between the larger particles by the smaller particles, i.e. better packing. On the other hand, workability will decrease if the aggregate particles are packed too tight, since the cement paste volume will be limited to separate the aggregates properly. Furthermore, increasing the size of the aggregates in the concrete is known to have a positive effect on workability as less cement will be required in a mixture. This solution, however, has a limited application in UHPFRCs as the maximum aggregate size is restrained. As a result, it is essential to find a balance between the workability and the efficient (optimal or ideal) grading (aggregate packing).

Initially, sieve analysis was performed for both coarse and fine aggregates. Coarse and fine aggregate samples were first dried in a drying oven at 105° C to a constant weight. Furthermore, they were shaken through a set of sieves of various diameters with a sieve having the largest openings placed on top, and a finest sieve positioned at the bottom. After sieving was performed, masses of material retained at each sieve and in the bottom pan were obtained, and the total initial weight of the samples was controlled. Besides, percent collected on each sieve and total percent of each sample passing every single sieve were calculated. The results are shown graphically on a grading chart (grading curve) in Figure 3.

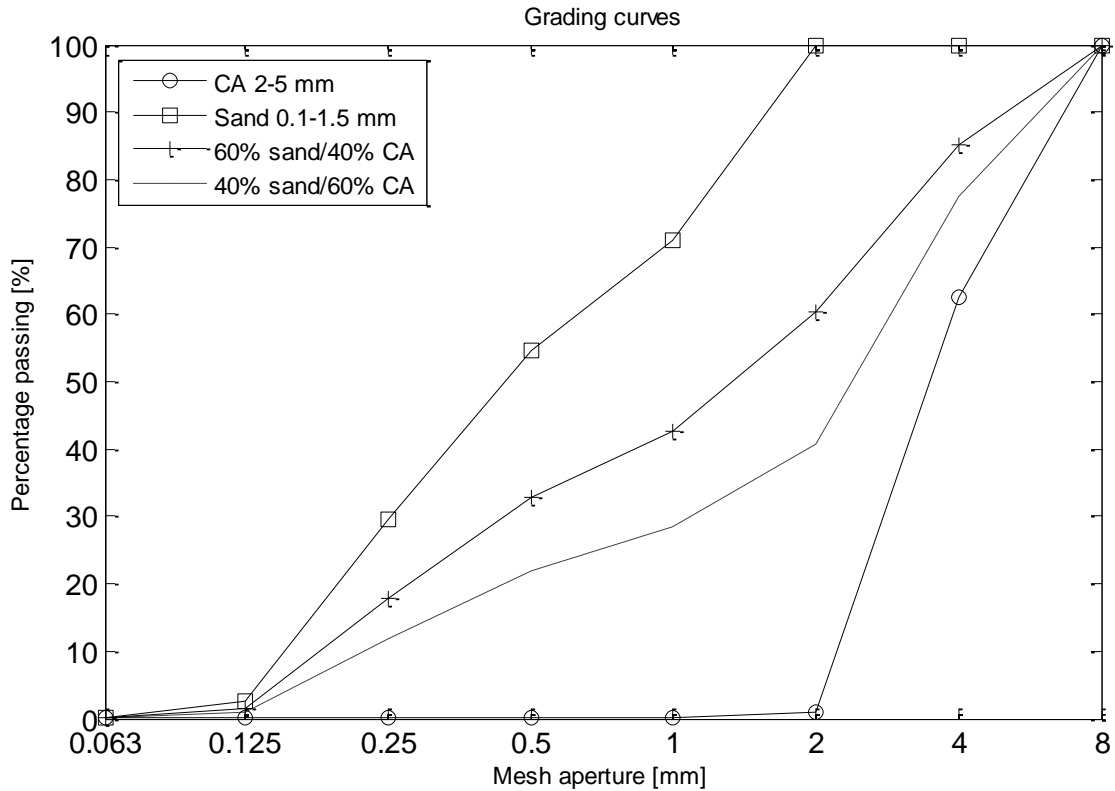


Figure 3 –Grading curves based on the sieve analysis of sand and coarse aggregates

Following, natural packing porosities of sand, coarse aggregates (CA) and their mixtures were measured. The natural packing porosity is a dimensionless quantity of the material calculated as the ratio between the pore (void) volume,  $V_p$ , and the total volume  $V_t$  of the natural packed material expressed either as a fraction or as a percentage. For concrete mixes it is especially important to evaluate the volume of voids between the solid aggregate particles (CA and sand) that can be filled with cement paste material. When the combined void content of the components of the concrete increases, more of the expensive cement paste will be required to fill the voids between the aggregates. Also, autogeneous and drying shrinkage are lower when the paste content is lower. Thus, it is desirable to find a good combination between the coarse and the fine aggregates to keep the void content as low as possible.

Total porosity is, generally, larger for soils and materials with fine texture (e.g. silt or clay), whereas, it is lower for coarser materials (e.g. sand, gravel, coarse aggregates). Normally, evaluation of the pore volume, i.e. the volume of water required to completely saturate the sample material, is rather complicated. In this research, porosity was measured using the following procedure: at first, coarse and fine aggregates were weighed and dried in a drying oven at 105° C until a constant mass was achieved. Next, the specimens were cooled in air and further placed in preweighed graduated glass cylinders to fill them to a mark indicating a total volume,  $V_t$ . Moreover, combination of the two aggregate materials (sand +CA) in different ratios specified in Table 2 was

also placed in the preweighed glass cylinders. Weight of the individual specimens,  $W$ , was calculated as the difference between the total weight of the glass cylinder with the specimen and the self-weight of the measuring glass cylinder. The density of the sand and CA grains,  $\rho_s$ , was also known (2.65 g/cm<sup>3</sup>). Finally, calculation of the natural porosities of sand, CA and the proposed combined grading of the two aggregate materials (sand +CA) in different proportions was performed indirectly using the following expression and the results are summarized in Table 2:

$$p_t = \frac{V_p}{V_t} = \left(1 - \frac{V_s}{V_t}\right) = \left(1 - \frac{W/\rho_s}{V_t}\right) \quad (1)$$

where  $V_p$  – pore volume, [m<sup>3</sup>];  
 $V_t$  – total volume, [m<sup>3</sup>];  
 $V_s$  – material particle volume, [m<sup>3</sup>];  
 $V_s/V_t$  – particle concentration;  
 $W$  – weight, [kg];  
 $\rho_s$  – density of grains, [kg/m<sup>3</sup>].

**Table 2 –Natural porosity of sand, coarse aggregates and their mixes**

| <b>Parameter</b>                                      | <b>100%<br/>0,1-1,5 mm Silica<br/>sand</b> | <b>100%<br/>2-5 mm CA<br/>(Hyperit)</b> | <b>40% 0,1-1,5 mm<br/>sand +<br/>60% 2-5 mm CA</b> | <b>60% 0,1-1,5 mm<br/>sand +<br/>40% 2-5 mm CA</b> |
|---|--|---|--|--|
| Total volume, $V_t$ [m <sup>3</sup> ]                 | 3·10 <sup>-4</sup>                         | 2·10 <sup>-4</sup>                      | 4.5·10 <sup>-4</sup>                               | 4.5·10 <sup>-4</sup>                               |
| Weight, $W$ , [kg]                                    | 0.5025                                     | 0.3123                                  | 0.817  | 0.8139   |
| Density of grains, $\rho_s$ ,<br>[kg/m <sup>3</sup> ] | 2650                                       |   |  |  |
| Particle concentration,<br>$V_s/V_t$                  | 0.63                                       | 0.59                                    | 0.69   | 0.68   |
| Natural porosity, $p_t$                               | 0.37                                       | 0.41                                    | 0.31   | 0.32   |

A mixture of 40% of 0,1–1,5 mm Silica sand and 60% of 2–5 mm CA (Hyperit granite) was used in the following mix design.

Four mixes of UHPFRC with Secutec S6 filling mortar by Contec ApS [41] and 0,4×12,5 mm fibres (2 vol. %) were initially tried in the AAU concrete lab in spring 2011. The results of mix optimization are summarized in Table 3. Normally, Secutec S6 consists of Secutec Binder [42] and calcined bauxite aggregates. Secutec binder is a specially developed extremely dense binder with special admixtures providing high compressive strength (up to 150–230 MPa), extremely low water permeability and a setting time, which is the time from the moment of adding water to the mixture until the paste is no longer fluid and plastic, of 4–8 hours at 20°C [42]. For a reference, a typical setting for DSP-concrete, achieved at 20°C is 6–18 hours [17]. Using calcined bauxite as an aggregate in HPFRC with Contec binder allows to achieve compressive strength of 220–270 MPa, and up to 200–400 MPa when adding steel fibres (in UHPFRC) [43].

In this study in order to reduce the cost of the final UHPFRC product bauxite aggregates were replaced by cheaper local silica sand and natural Hyperit granite aggregates. Hyperit granite of 2–5 mm used in this study can be characterized as a CA material having subangular, rough and flaky grains which means that it can negatively affect the workability. The initial mix was performed according to Contec ApS's recipe with w/b=0.18 and a paste volume of 42.8 %. Mixing was performed in accordance with procedure described by the manufacturer [41]. The UHPFRC Secutec S6 castable was mixed for 4 minutes in a pan-type mixer before the steel fibres were added. Then, the mixture was mixed additionally for 3 minutes.

Workability of UHPFRC mixes was measured using the cone test method according to ASTM C230 [44]. The cone mould with top and bottom diameters of 63 and 102 mm, respectively, and a height of 52 mm was used. The cone mould was placed on the middle of the wooden plate of the flow table apparatus and filled during slight compaction performed by using a tamping rod. Then the cone mould was removed by lifting it vertically, thus, allowing the concrete to flow freely. The diameter of the resulting horizontal spread of concrete was measured in two perpendicular directions and presented in Table 3. As recommended in the technical data sheet of Contec ApS [41], workability can be slightly improved by vibration. Thus, the vibration flow table was subjected to jolting by lifting it up and dropping 20 times. This further caused the concrete to flow and the horizontal projection of the spread was measured and documented one more time (see Table 3), and the relative evaluation of workability of the concrete mixes was performed.

The observed workability of the first mix was very low (see Table 3). A second trial mix had a higher w/b ratio (w/b=0.23) with the same paste volume. This solution helped to increase the workability, but the results were still not quite satisfactory. According to Contec ApS, the typical w/b range for Secutec S6 lies within 0.19–0.21. Therefore, in the third and fourth trial mixes the w/b ratio was lowered to 0.20 while the paste volume was increased to 50 and 60 %, correspondingly, as shown in Table 3. Based on the results summarized in Table 3, the third mix was selected for further investigation with a paste volume of 50% and a water/binder ratio w/b=0.2.

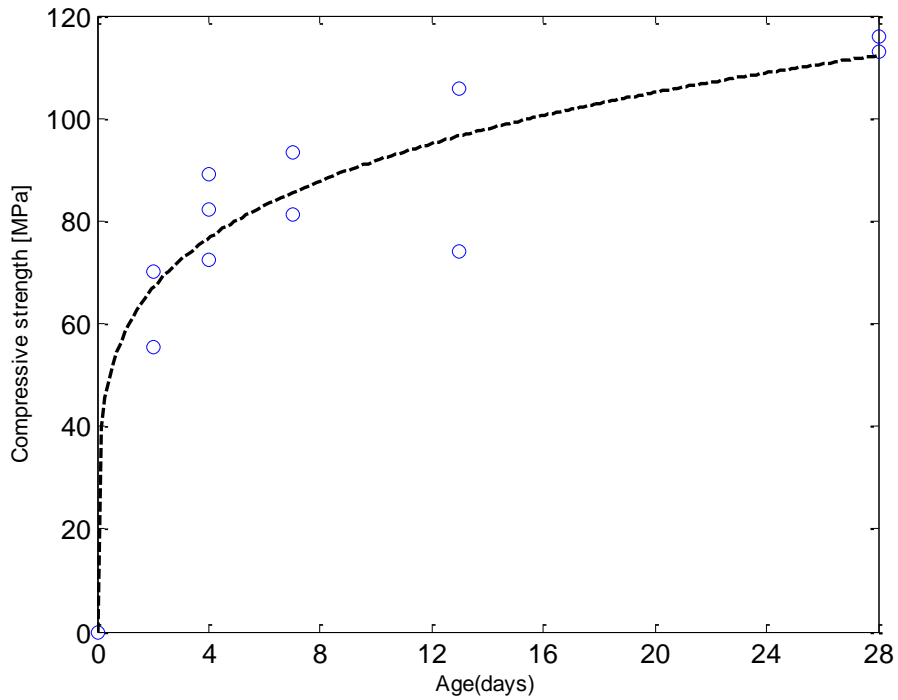
Table 3 – Optimization of the standard Secutec S6 concrete mix (manufacturer: Contec ApS)

| Parameter                               | Density, kg/m <sup>3</sup> | Content kg/m <sup>3</sup> | Content m <sup>3</sup> /m <sup>3</sup> |
|---|----------------------------|---------------------------|--|
| <b>1. Paste volume- 42.8%, w/b=0.18</b> |                            |                           |  |
| Binder (Secutec)                        | 2900                       | 816                       | 0.281                                  |
| Water                                   | 1000                       | 147                       | 0.147                                  |
| Silica sand                             | 2650                       | 585                       | 0.221                                  |
| CA (Hyperit granite)                    | 2650                       | 877                       | 0.331                                  |
| Steel fibres 0,4x12,5 mm                | 7850                       | 157                       | 0.020                                  |
|   | <b>Σ</b>                   | <b>2582.0</b>             | <b>1</b>                               |

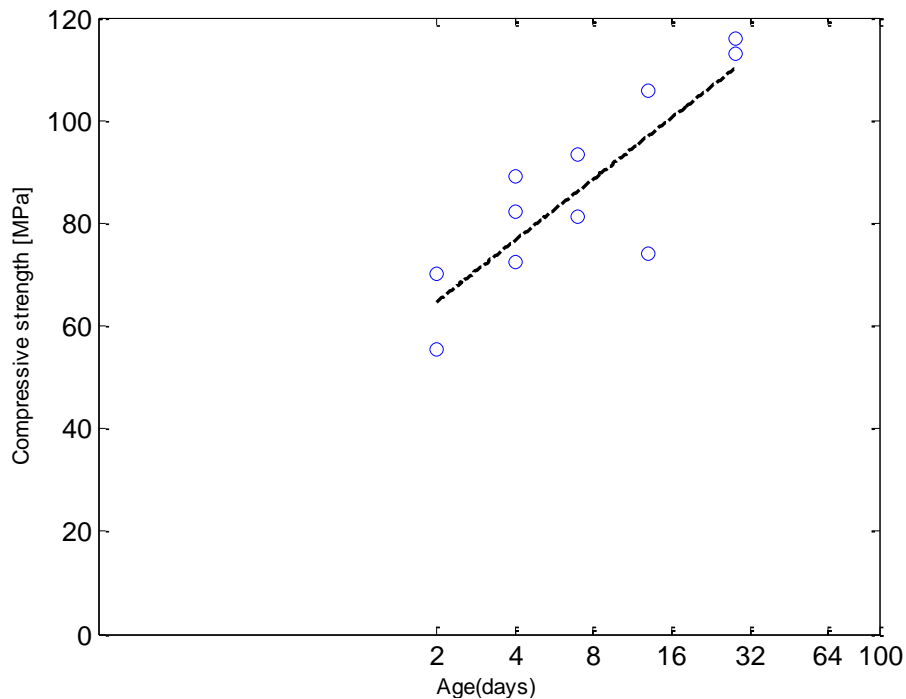
| Parameter                               | Density, kg/m <sup>3</sup>  | Content kg/m <sup>3</sup> | Content m <sup>3</sup> /m <sup>3</sup> |
|---|---|---------------------------|--|
| <b>Results</b>                          | Cone test (free flow) <b>Ø 100 mm</b><br>After 20 strokes <b>Ø 100–103 mm</b>     |                           |  |
| <b>2. Paste volume- 42.8%, w/b=0.23</b> |   |                           |  |
| Binder (Secutec)                        | 2900  | 745                       | 0.257                                  |
| Water                                   | 1000  | 171                       | 0.171                                  |
| Silica sand                             | 2650  | 585                       | 0.221                                  |
| CA (Hyperit granite)                    | 2650  | 877                       | 0.331                                  |
| Steel fibres 0,4x12,5 mm                | 7850  | 157                       | 0.020                                  |
|   | <b>Σ</b>  | <b>2535</b>               | <b>1</b>                               |
| <b>Results</b>                          | Cone test (free flow) <b>Ø 100–105 mm</b><br>After 20 strokes <b>Ø 105–125 mm</b> |                           |  |
| <b>3. Paste volume- 50%, w/b=0.2</b>    |   |                           |  |
| Binder (Secutec)                        | 2900  | 918                       | 0.316                                  |
| Water                                   | 1000  | 184                       | 0.184                                  |
| Silica sand                             | 2650  | 509                       | 0.192                                  |
| CA (Hyperit granite)                    | 2650  | 763                       | 0.288                                  |
| Steel fibres 0,4x12,5 mm                | 7850  | 157                       | 0.020                                  |
|   | <b>Σ</b>  | <b>2531</b>               | <b>1</b>                               |
|   | Cone test (free flow) <b>Ø146–150 mm</b><br>After 20 strokes <b>Ø166–170 mm</b>   |                           |  |
| <b>4. Paste volume- 60%, w/b=0.2</b>    |   |                           |  |
| Binder (Secutec)                        | 2900  | 1101                      | 0.380                                  |
| Water                                   | 1000  | 220                       | 0.220                                  |
| Sand                                    | 2650  | 403                       | 0.152                                  |
| CA (Hyperit granite)                    | 2650  | 604                       | 0.228                                  |
| Steel fibres 0,4x12,5 mm                | 7850  | 157                       | 0.020                                  |
|   | <b>Σ</b>  | <b>2485</b>               | <b>1</b>                               |
|   | Cone test (free flow) <b>Ø187–192 mm</b><br>After 20 strokes <b>Ø207–209 mm</b>   |                           |  |

In order to get the design properties of the optimized UHPFRC mixture with natural aggregates it was important to measure its compressive strength. Based on results of tests on 12 concrete cylinder samples (100×200 mm) carried out at AAU in spring 2011, a mean compressive strength at 28 days was identified and presented in Table 4. The cylinders were first cast without vibration, cured in moulds for 1 day at 20° C and then stored in a water bath at 20° C until testing. Compressive strength was determined according to EN 12390-3 [45]. The early strength of the concrete was measured by performing 1-, 2-, 4- and 7-days tests. Measuring the 1-day strength was unsuccessful, since the tested sample crushed into pieces. The delayed strength increase can be explained by the initial set occurring approximately 15 hours, sometimes even up to 36 hours after mixing [46]. On the second day the average compressive strength was 62.8 MPa, which is

comparable to 70 MPa reported e.g. by Graybeal [46]. A mean compressive strength of 114.4 MPa was achieved for the cylinders tested after 28 days (see Table 4). Consequently, the mean compressive strength of the UHPFRC matrix was approximately one half of that of the comparable Densit® matrix with stronger artificial aggregates used for CRC® (see e.g. Table 1) or UHPFRC with Secutec S6 filling mortar and bauxite aggregates by Contec ApS [43]. Based on the results of the cylinder tests a strength development curve was plotted as shown in Figure 4.



a)



b)

Figure 4 – Compressive strength development curve: a) linear time scale; b) log time-scale (paste volume 50 %, W/B= 0.2)



Table 4 – Compressive strength for the Secutec S6 mix with a paste volume of 50% and w/b=0.2

| Compressive strength, MPa         |                       |                   |                   |                   |                   |                    |
|-----------------------------------|-----------------------|-------------------|-------------------|-------------------|-------------------|--------------------|
| Curing time/# of specimens tested | 1 day<br>(25.03.2011) | 2 days<br>(26.03) | 4 days<br>(28.03) | 7 days<br>(31.03) | 13 days<br>(6.04) | 28 days<br>(21.04) |
|                                   | 1                     | 2                 | 3                 | 2                 | 2                 | 2                  |
| 1                                 | 0                     | 70.16             | 82.12             | 81.11             | 105.93            | 112.94             |
| 2                                 |                       | 55.51             | 72.45             | 93.33             | 73.98             | 115.86             |
| 3                                 |                       |                   | 89.13             |                   |                   |                    |
| <b>Mean</b>                       |                       | 62.84             | 81.23             | 87.22             | 89.96             | 114.4              |

### 3.2 Mixing and casting of UHPFRC specimens for fibre orientation analysis

For the proper functioning of the TP structure it is important to have:

- Uniform fibre distribution, fibre orientation and fibre volume in the UHPFRC specimens;
- No large air pockets and adequate fibre orientation around main reinforcement bars;
- Reduction/elimination of surface cracks by use of shorter fibres and/or various ways of casting (based on naked-eye observations of the specimens).

To investigate this, two types of additional tests on UHPFRC with main reinforcement were carried out:

- Destructive with steel main reinforcement – for visual inspection of cracks;
- Non-destructive tests (CT-scanning) with FiReP®REBAR FRP main reinforcement [48] – for analysis of fibre orientation, fibre concentration, and presence of voids/cracks.

First, an UHPFRC sample was cast with 2 vol. % steel fibres 0,4×12,5 mm and the dimensions 480×480×90 mm as shown in Figure 5. The wall thickness of the sample corresponded to the representative wall thickness of the full-scale TP structures based on the FEM analysis calculations carried out by Nezhentseva et al. [7,8]. The panel was cast in a mould with a very high concentration of steel rebar Ø 16 mm closely tied together and arranged in two layers in the horizontal direction and one layer in the vertical direction. The horizontal rebar was located closer to the surface, while the vertical rebar was centrally positioned in the middle of the formwork. The distance between the centres of the neighbouring bars was 60 mm, i.e. there were 7 rows of reinforcement in each direction [7,8]. Casting was performed vertically without vibration by continually filling the formwork along the top. A transparent front side of the formwork made of Plexiglas allowed to monitor the flow of the mixture during the casting.

After removing the formwork, several hairline surface cracks were observed along the horizontal reinforcement bars. These cracks are feared to potentially create a threat of corrosion of the main reinforcement when exposed to aggressive sea water conditions. In order to further inspect

possible cracks or voids inside the body of the specimen, two cylindrical samples were drilled out from the top and the bottom parts of the sample where the hairline surface cracks were observed. The drilled cores are shown in Figure 6. The specimens split into smaller fragments alongside the main reinforcement after an attempt was made to saw them in halves. Having these samples inspected with the naked eye, multiple cracks were detected around the reinforcement bars inside the body of the top sample (Figure 6, marked in red). A possible origin of these cracks could be from disturbance caused by drilling and further sawing of the samples. Yet, based on these observations a decision to investigate this problem in more detail was made using a non-destructive testing technique (CT scanning). Additionally, fibre orientation, fibre distribution and actual fibre volume could be studied in different parts of the specimens (i.e. above, around and underneath the rebar).

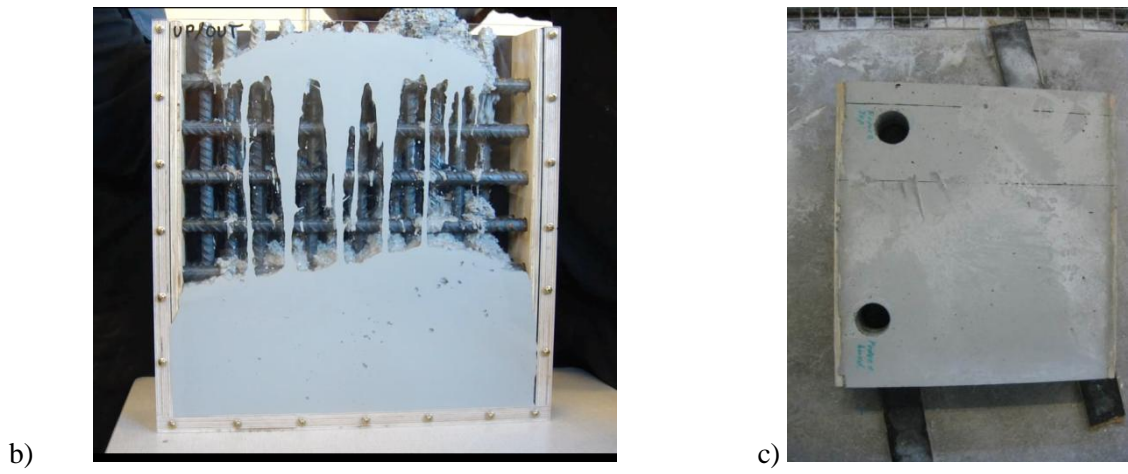
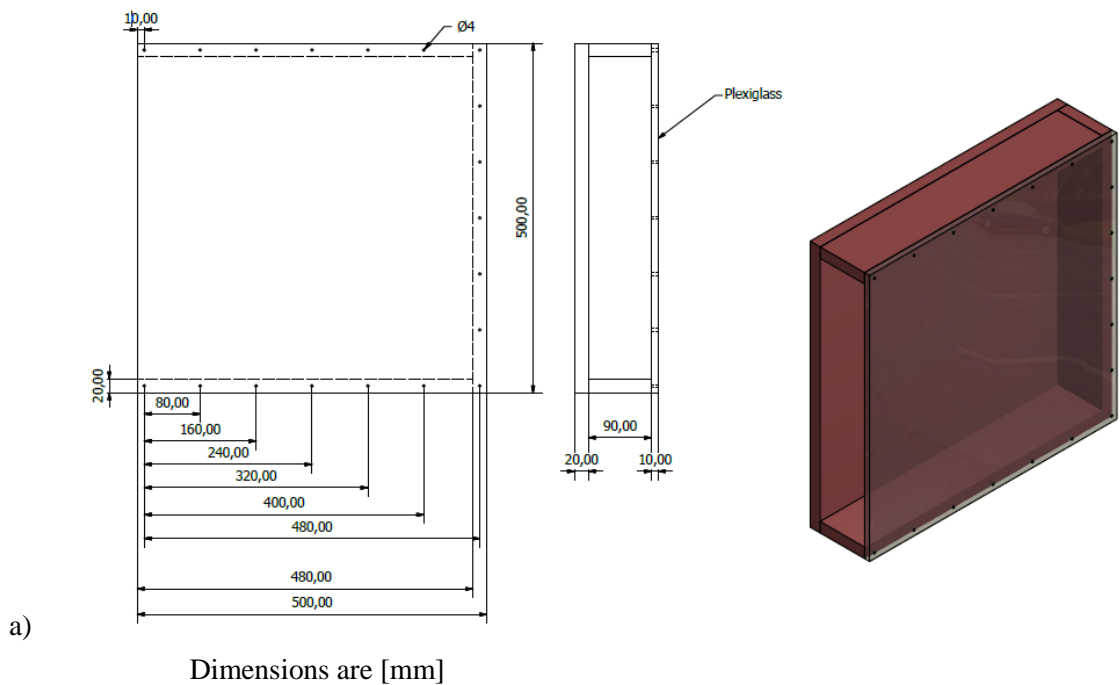


Figure 5 – UHPFRC sample with steel reinforcement cast at AAU in spring 2011: a) formwork; b) casting; c) final product with positions of the drills: sample 1 (bottom) and sample 2 (top).

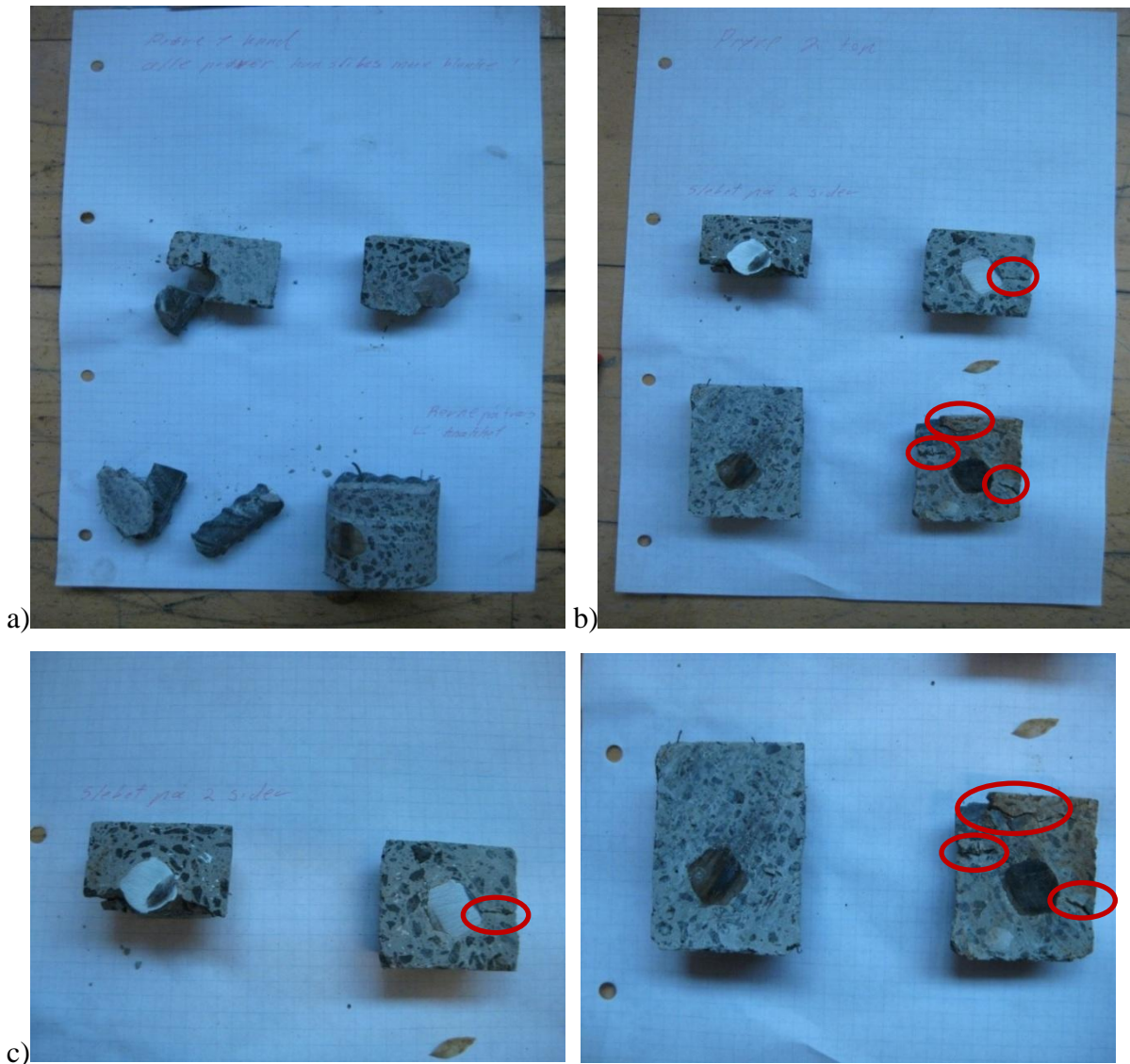
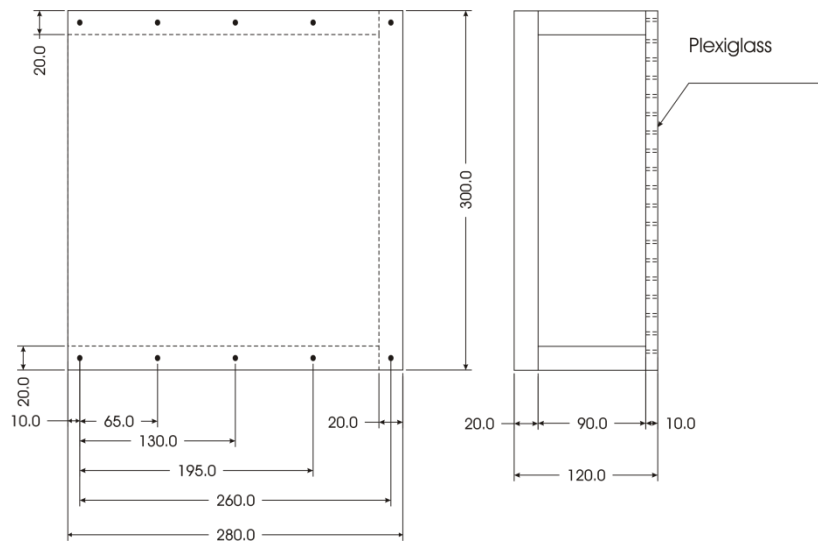


Figure 6 – a) sample 1 (bottom); b) sample 2 (top); c) cracks in sample 2 (top)

### 3.3 Specimen types for CT-scanning

Two types of steel fibres, longer –  $0,4 \times 12,5$  mm and shorter –  $0,16 \times 6$  mm were selected for casting a batch of UHPFRC samples at AAU in spring of 2012 and for further non-destructive examination of voids and cracks using CT-scanning. The same mix composition was selected for simplicity as described in Section 3.1, Table 3 (mix 3) maintaining a fibre dosage of 2 vol. %. The workability and the flow of the new mix with shorter fibres were not measured but were simply judged visually to be adequate to perform casting without vibration. A decision to replace the main steel reinforcement by fibre-glass (aligned glass fibres held together by a polymer) FiReP®REBAR FRP reinforcement [48] of  $\text{Ø}20$  mm was made as the main reinforcement made of steel would «consume» excessive energy when performing X-ray CT-scanning, thus, severely disturbing the analysis. The samples were cast in horizontal and vertical moulds of various sizes with FRP

reinforcement arranged in three layers closely tied together as shown in Figure 7. The same distance between the centres of the neighbouring bars was used, i.e. 60 mm, and the cover thickness (the distance between the surface and the first layer of reinforcement) was 15 mm.



a)

Dimensions are [mm]



b)

Figure 7 – UHPFRC samples with FiReP®REBAR FRP reinforcement cast at AAU in spring 2012: a) formwork for small samples; b) arrangement of reinforcement in horizontal, vertical and “tall vertical” samples

Casting was performed without any vibration in the following ways (see Figure 8):

a) Vertically:

- free fall from 2 diagonal points at the top of shorter (260×260×90 mm) and taller samples (260×1000×90 mm);

- fictitious pumping from the bottom using free fall technique and vertical pressure variation. This attempt was unsuccessful due to segregation of UHPFRC material. The problem might be explained by a probable loss of workability as pumping was delayed after casting due to challenges of arranging the test setup and performing the tests. It is still believed that UHPFRC can be pumped if casting is performed rather fast, preferably within 15–20 minutes from the time of mixing the

product, as the major loss of workability will occur shortly after casting [49,50]. As a result, this sample was discarded.

b) Horizontally by filling a 260×260×90 mm sample from two diagonally located points.



Figure 8 – Casting of the UHPFRC samples with FiReP® reinforcement (AAU, spring 2012): a) 2 point casting of a vertical sample; b) finished vertical sample cast vertically from 2 points (shown in red); c) 2 point casting of a horizontal sample.

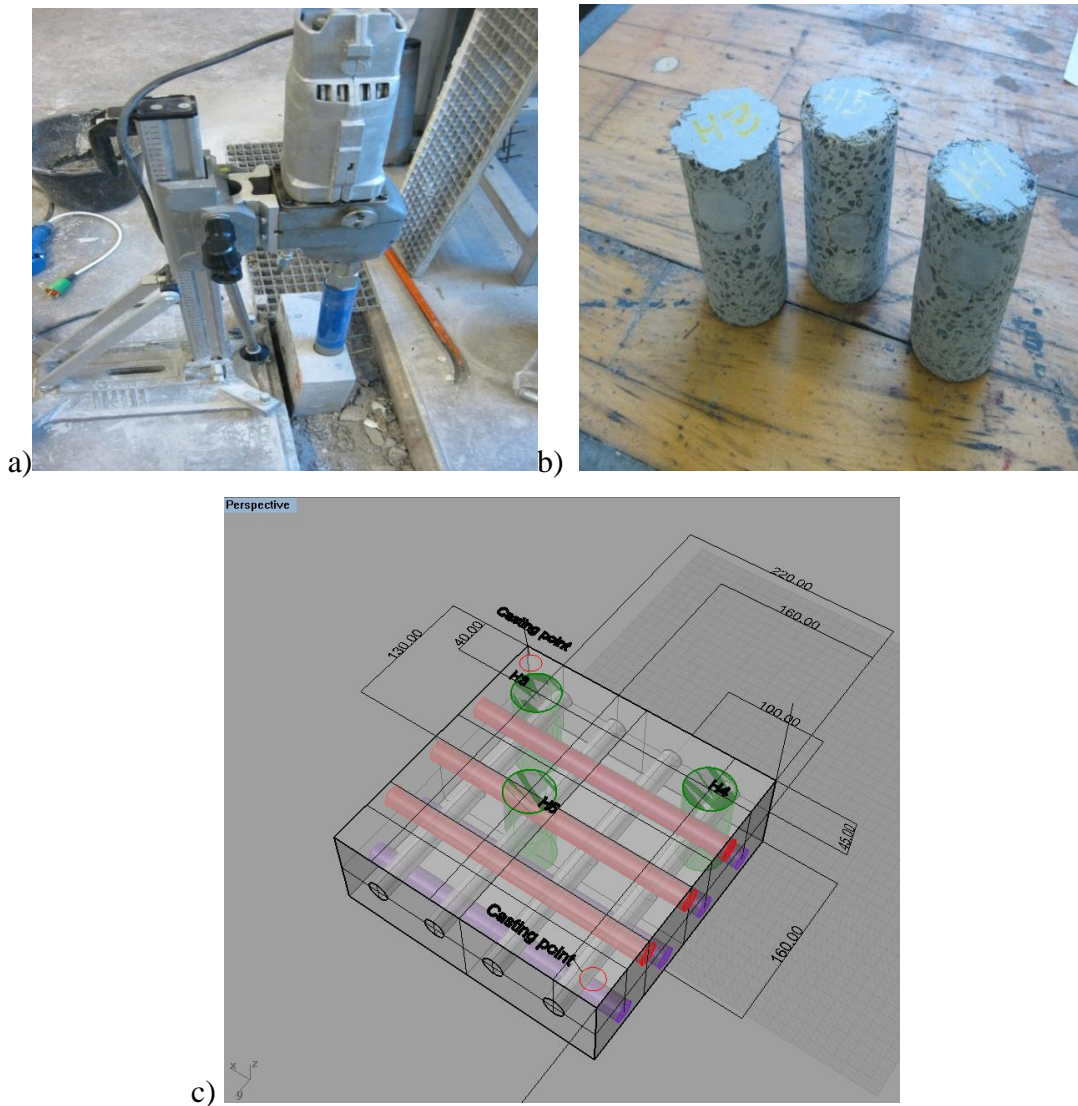


Figure 9 – a) Drilling of a sample; b) UHPFRC specimens H3-H5 with FiReP® reinforcement; c) a sketch of the positioning of the specimens for CT scanning in a sample cast horizontally.

Subsequently, 16 smaller specimens of  $\varnothing 45$  mm and a height of 90 mm (sample thickness) were drilled out of the UHPFRC samples (see Figure 9) for future CT-scanning and examination of inner voids and cracks. The selection of the cuts in the specimens was made with the purpose of experimental investigation of homogeneity and isotropy using the following considerations: proximity to the point of casting (as near as possible to the point of casting, in the opposite corner and approximately in the centre of a specimen) and positioning of the intersections of the reinforcement bars or individual FiReP® bars (longitudinal or transverse) in the centre of the specimens. This was made to study patterns of the dispersion and orientation of steel fibres in the most representative and potentially problematic areas (e.g. far corners), as well as to revise fibre distribution and/or presence of air pockets and cracks around individual or intersecting rebar. Only nine samples of major importance were finally selected for the CT-scanning examination due to the high cost of testing (H3–H5, VB1, VB5, HB1, HB3, VA1 and VA4):

- 1) **sample 1 (0,26×0,26×0,09m, 0,4×12,5mm fibres, cast horizontally):**

- a) next to the point of casting (H3),
  - b) in the opposite corner (H4),
  - c) almost in the middle of the specimen (H5)
- 2) **sample 2 (0,26×0,26×0,09m, 0,4×12,5mm fibres, cast vertically):**
- a) at the top next to the point of casting (V6),
  - b) at the bottom in the opposite corners (V4 and V3),
  - c) approximately in the middle of the specimen (V5),
  - d) long specimen along the vertical rebar (V1)
- 3) **sample 3 (0,26×0,09×1m, 0,4×12,5mm fibres, cast vertically):**
- a) at the top next to the point of casting (VB1),
  - b) at the bottom in the corner (VB5)
- 4) **sample 4 (0,26×0,26×0,09m, 0,16×6mm fibres, cast horizontally):**
- a) next to the point of casting (HB1),
  - b) in the opposite corner (HB3)
- 5) **sample 5 (0,26×0,09×1m, 0,16×6mm fibres, cast vertically):**
- a) at the top in the middle (VA1),
  - b) at various heights in the middle of the specimen (VA2, VA3,VA4).

## 4. CT SCANNING

### 4.1 Results

Several surface cracks were observed in the UHPFRC samples cast vertically different from the horizontally cast specimens. This could partially be explained by a smaller fibre concentration closer to the surface (i.e. in the cover layer) due to a smaller effect of the matrix fibres (see, e.g. CRC® Patent [37]). Perry et al. [50] addressed the importance of the mould design on the initial autogeneous (drying) shrinkage of the fresh UHPFRC material not yet being able to achieve internal tensile capacity. The results of their experiments indicated that filling vertical (or nearly vertical) specimens from the bottom would assure the best results of the surface of the UHPFRC [50].

The biggest concern is, however, how far these visible surface cracks have actually penetrated into the main body of the samples, and whether they pass the main reinforcement. This is particularly significant for the TP structures of special importance where no surface cracks are acceptable, and therefore should be avoided.

## 4.2 Computed Tomography (CT)

Non-destructive testing (NDT) methods currently used for evaluation of fibre distribution in concrete include:

- microscopic examination of polished sections and manual counting of fibres (see, e.g. Tue et al. [51]);
- electrical measurement methods: electrical resistivity (see Lataste et al. [52]) and Alternate Current Impedance Spectroscopy (AC-IS) (Ozyurt et al. [53], Woo et al. [54] );
- magnetic inductive method (Ferrara et al. [55,56], Torrents et al. [57])
- X-ray CT investigations (Schnell et al. [58,59])

Electrical resistivity is a NDT technique providing information about steel fibre distribution and local orientation of steel fibres based on the identification of more or less electrically resistant regions within an UHPFRC specimen. Testing is performed by means of two electrodes introducing a low frequency electrical signal of a predefined value to a concrete sample. When the current passes through the UHPFRC material a potential difference will be generated. This potential difference will further be detected and measured by two additional electrodes. Electrical resistivity method was used, for example, for characterisation of fibre distribution in Ductal® (by Lafarge) UHPFRC concrete with 2 vol. % steel fibres performed by Lataste et al. [52].

AC-Impedance Spectroscopy is a new electrical measurement method for non-destructive monitoring of orientation and arrangement of conductive fibers in both fresh and hardened fibre-reinforced cement-based materials [53,54]. AC-IS is a promising technique based on the intrinsic conductivity approach. An excitation voltage with a range of frequencies is initially applied to a specimen. The magnitude and the phase of the current are further measured. Each frequency generates a single point of the real and imaginary values of impedance converted from the acquired data and further presented on special plots [53,54]. This technique is, however, sensitive to the dispersion of fibres, i.e. clumping, segregation and orientation of fibres.

Magnetic inductive non-destructive testing technique [55–57] is based on the principle of ferromagnetic induction and ferromagnetic properties of steel fibres modifying by their presence the magnetic field lines. Apart from being easy to use, having a good sensitivity and relying on simple equipment, this method allows determination of content and orientation of steel fibres in the concrete matrix regardless of its age and moisture content. However, this method can only detect steel fibres and is not able to provide information about other components in the concrete matrix.

According to Schnell et al. [58] CT scanning technique is advantageous to the other methods as it allows analysing and investigating the fibre-orientation and the fibre-distribution in the entire volume of a specimen without a need for sample preparation. Moreover, it is also possible to



examine and describe local fibre qualities and micro crack propagation. On the other hand, high-resolution CT scanners for testing microstructure of civil engineering materials (soil, cement, reinforced concrete, rock, asphalt) are large, very expensive, and require special safety measures to operate. This results in a high price of a single sample scan.

X-ray Computerized Tomography (XCT), Computed Tomography or CT scan, is a non-destructive imaging procedure developed initially for medical diagnosis of internal organs by G. Hounsfield in 1970 [60,61]. XCT combines two-dimensional (2D) or three-dimensional (3D or volumetric) X-ray projections with powerful computer algorithms. It is based on penetrating electromagnetic radiation in terms of computer-processed X-rays having a wavelength in the range 0.01–10 nm to obtain tomographic high-resolution images for volumetric inspection of a tested sample from inside. If required, information about crystal structure, chemical composition, and physical properties of particular materials comprising the sample can also be obtained. CT scanning is performed by making numerous 2D “slices” (images) of single planes of the specimen taken around a specified rotational axis followed by a digital geometry processing procedure to create a volumetric image [62], see e.g. Figure 10. These images are created by means of measuring the scattered intensity of X-rays passing through matter, reaching different material components, e.g. steel fibres, coarse aggregates and FRP reinforcement in UHPFRC, and partly being absorbed in transmission. X-rays penetrate through an object and interact with atoms and molecules of different materials in various ways depending on the energy of the X-rays and material composition [63].

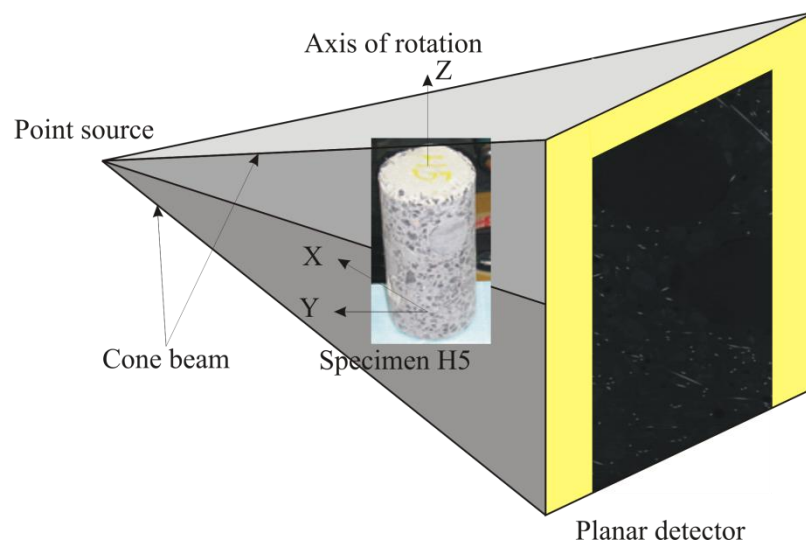


Figure 10 – Three-dimensional computer tomography – a schematic view (adapted from Schnell et al.[59])

The Beer-Lambert law [64,65] describes the relationship between absorption of X-ray photons (particular light particles on the atomic level) produced by an X-ray tube and the properties of the media through which the X-ray photons are travelling (see, e.g. Köser et al. [66]):

$$I = I_0 \cdot e^{(-\mu x)} = I_0 \cdot e^{(-\mu^* \rho x)} \quad (2)$$

where  $I_0$  – the rate of the incident radiation intensity (photon beam);  
 $I$  – the rate of the transmitted (emerging) radiation intensity (photon beam);  
 $\mu$  – linear attenuation coefficient of a certain material characterizing how easily the X-ray photon can penetrate medium, i.e. the photon absorption or scatter per unit length ( $\text{cm}^{-1}$ );  
 $\mu^*$  – mass attenuation coefficient, measurement of how strongly a certain material absorbs or scatters X-ray photon light at a given wavelength, per unit mass ( $\text{cm}^2 \text{g}^{-1}$ );  
 $\rho$  – material density ( $\text{g cm}^{-3}$ );  
 $x$  – material thickness or distance travelled (cm).

$$\mu^* = \mu / \rho \quad (3)$$

Different CT scanners have various photon energy spectra, which can be normalized in Hounsfield units:

$$H = 1000 \cdot \frac{\mu - \mu_{\text{water}}}{\mu_{\text{water}}} \quad (3)$$

where  $H$  – Hounsfield units (HU) or CT numbers (for water –0 HU, air – 1000 HU);  
 $\mu_{\text{water}}$  – linear attenuation coefficient for water.

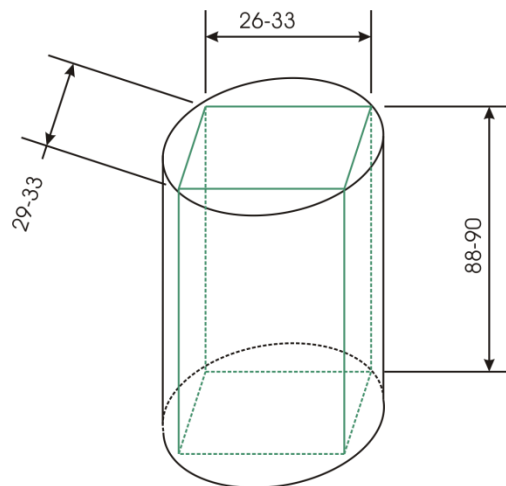
Normally, different densities of components of UHPFRC matrix correspond to a range of CT numbers (or Hounsfield units (HU)) assigned to each single point of the matrix (pixel) in various intensities or shades of gray scale. Higher CT numbers will normally be associated with white regions, and vice versa.

In this study, X-ray computed tomography (CT) was used as a tool to evaluate the fibre orientation, distribution and volume, as well as to detect surface and inner cracks, and voids in the UHPFRC specimens. The samples were scanned with a high-resolution 3D X-ray CT scanner at Fraunhofer ITWM in Kaiserslautern, Germany. The CT scanner used for this study was manufactured by YXLON. It was equipped with a FeinFocus X-ray tube assuring the highest resolution for fiber-reinforced materials. The maximum acceleration voltage and power provided by the CT scanner are – 225 kV and 25 W, respectively. The scans were performed at 400 angles as 8 bit images (8 bits per sampled pixel, i.e. 256 different intensities or shades of gray) with several slices with a slice thickness of  $60 \cdot 10^{-6}$  m according to “resolution” of  $x \mu\text{m}/\text{pixel}$  within each sample at 168.9 kV voltage and X-ray exposure of 101  $\mu\text{A}$ . The reconstruction of the data was performed in Volex XRayOffice software using 400 directions with 2 pictures taken from each direction. Maximum size of the cylinder specimens that could be tested was 60 mm in diameter, 15 mm in height, alternatively 100 mm in diameter and 100 mm in height depending on the required resolution (geometric magnification).

The computer software MAVI 1.4.1 (Modular Algorithms for Volume Images) [66] developed at Fraunhofer ITWM, Germany, was used for processing and analysis of volume images

produced by micro computed tomography. As the X-rays propagated through each pixel of the image of the sample, they were scattered or absorbed in various ways depending on the density and the atomic number of the material they passed. This corresponded to different gray levels in the CT scanned images. In order to separate steel fibres, air pockets around the rebar and cracks from the concrete matrix, so-called threshold images had to be used in the segmentation process. This was performed by setting a threshold level in such a way that a total number of the isolated pixels in each phase was minimized. As for the steel fibres, pixels having higher gray values (white regions corresponding to the areas with higher CT numbers) were interpreted as belonging to the fibre phase, whereas the threshold levels (dark regions with lower CT values) were interpreted as belonging to the concrete phase.

Raw 3D image data from the CT scanning was imported to MAVI. In order to reduce the size of the raw data, cropping of the scanned images was performed on three coordinate axes in such a way, that both the top and the bottom of each specimen were included, while the circular sides of the scanned cylindrical specimens were cropped and the specimens turned into 3D rectangular prisms (see Figure 11).



Dimensions are [mm]

Figure 11 – Cropping of the UHPFRC specimens in MAVI (dimensions in mm).

Furthermore, morphological transformation of the cropped image was performed using an opening filter. Generally, filtering (opening) can be used to remove small objects, to reduce the noise and to enhance image contrast of gray value images. Moreover, a binary operation of subtraction was performed for the cropped and the filtered images. This operation allowed taking pixel values from both images, computing the difference of their grey values and later on writing the result in the output image. The result of subtraction was further subjected to segmentation. Meanwhile, an input file containing grey values was transformed into an output binary image containing only zeros and ones as values depending on whether the pixel's value lied within the

given range (global thresholding). The threshold value was selected using trial and error technique until a desired averaged diameter of the steel fibres was achieved (i.e. 0,4 mm and 0,16 mm). The accuracy of the results of the fibre thickness measurements was further validated. After the CT-analysis of a single specimen was performed, an additional CT-analysis was completed for the same specimen turned 180° (upside down) in the CT scanner. A common source of error would be underestimation of the fibre thickness due to an indistinguishable transition of the gray values from the matrix to the fibre phase due to the poor quality of binarisation. However, working directly on a gray value image would help to solve this potential source of error [58].

Quantitative geometric analysis of the steel fibres was performed using an Open Foam and Field Features tools. The following characteristics were determined: mean diameter of the edge (Open Foam Feature), volume density, surface area, specific fibre length, length of total projections corresponding to the directions of the projections (Field Feature). For selected images, several additional actions were performed with the binary image in order to show a better quality image of the steel fibre distribution. Initially, a simple linear transformation called “spreading” was carried out which allowed to transform the range from minimum to maximum value presented in the input 2-value binary image (the dynamic range) to the full range of the output image (GRAY8 image). The final step was to remove the noise from the spread image using a smoothing filter called Mean Filter with a filter mask size 3. Lastly, volume rendering view allowed visualization of the 3D-data presented in the following section.

### 4.3 Fibre distribution

Fibre distribution in the 0,26×0,26×0,09m and 0,26×0,09×1m UHPFRC specimens cast horizontally and vertically, respectively, with both short and long fibres is represented in Figures 12–13. Furthermore, fibre orientation and fibre volume in the UHPFRC specimens based on the quantitative geometric analysis performed in MAVI is summarized in Table 5. A fibre orientation number,  $\eta_\varphi$  (see e.g. Schnell et al. [59]), was defined as:

$$\eta_\varphi = \frac{L_{p,\varphi}}{L_V \cdot V} \quad (4)$$

where  $L_{p,\varphi}$  – total projected fibre length on the corresponding direction  $\varphi$  (corresponding to axis x, y or z), [m];  
 $L_V$  – specific fibre length, or total fibre length per unit of volume, [m/m<sup>3</sup>];  
 $V$  – total volume of the specimen, [m<sup>3</sup>].

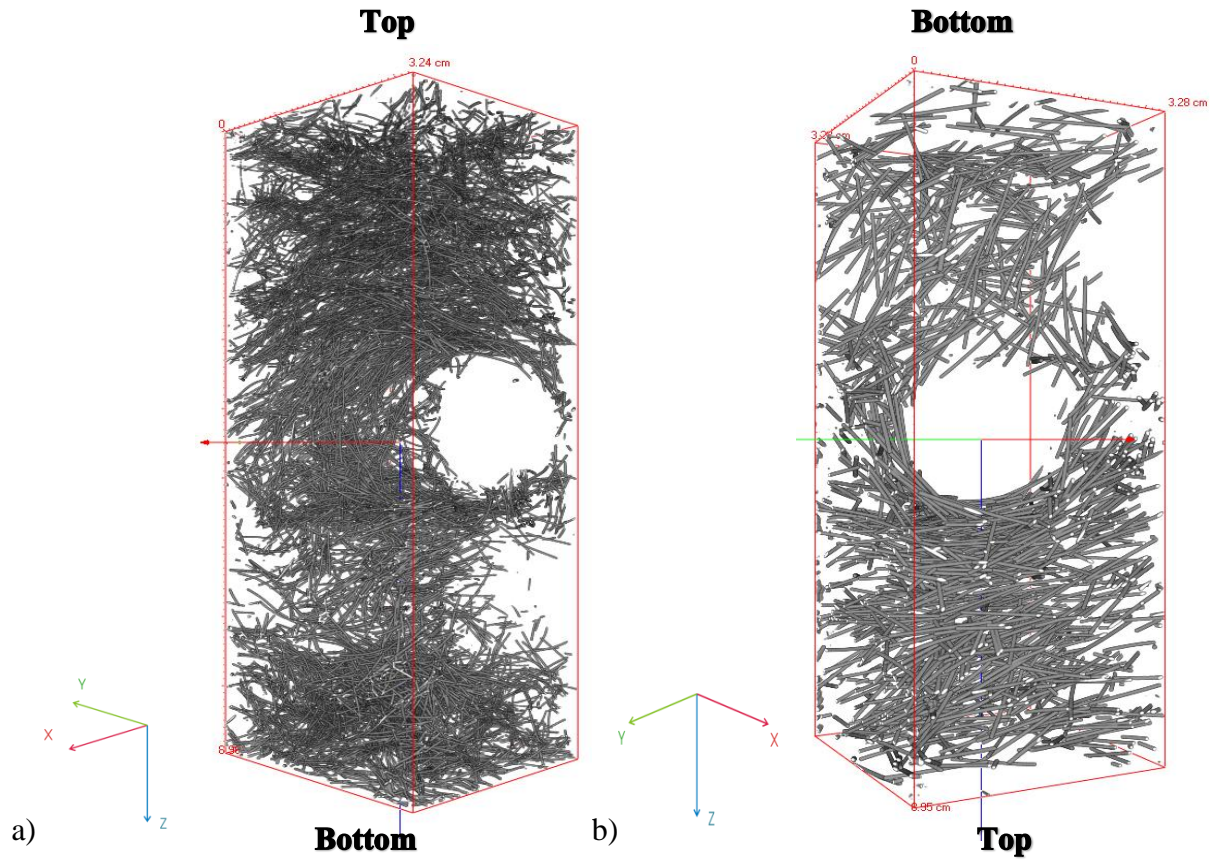


Figure 12 – Visualisation of the fibre distribution in the UHPFRC specimens cast horizontally (0,26x0,26x0,09 m). Specimens cut out at the corners opposite to the corners where casting was performed: a) HB3 with 0,16x6 mm fibres; b) H4 with 0,4x12,5 mm fibres.

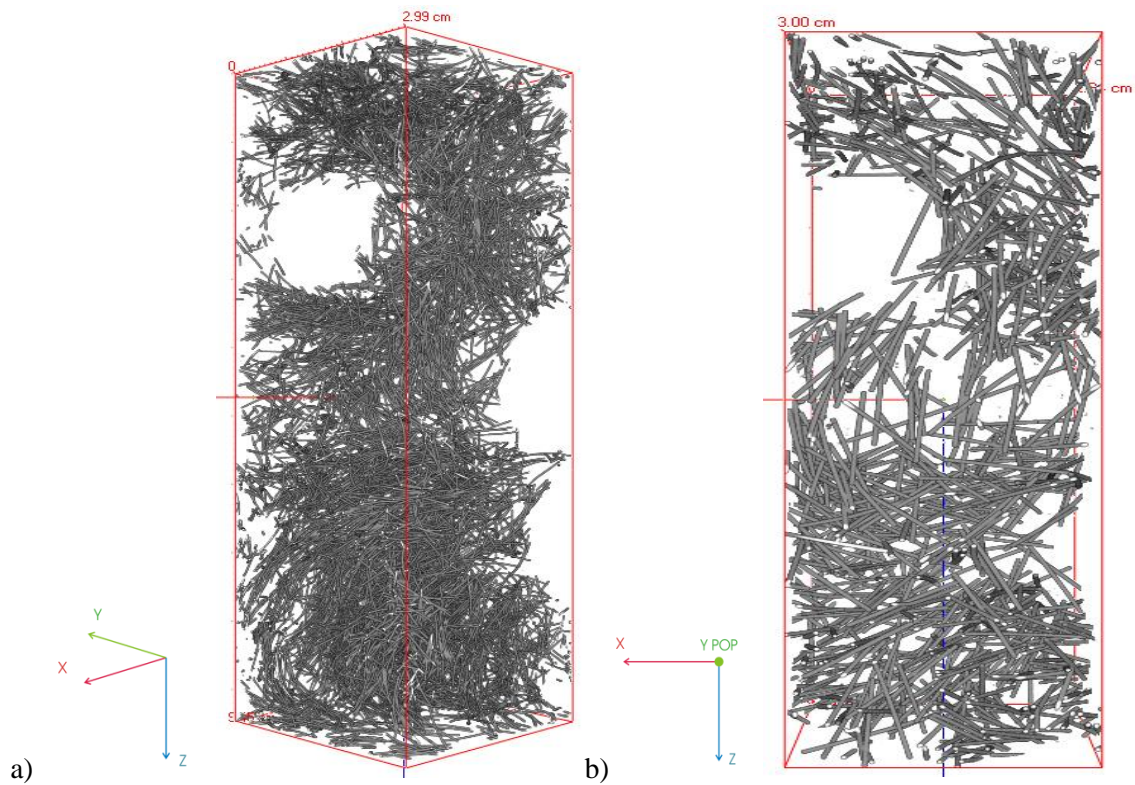


Figure 13 – Visualisation of the fibre distribution in the UHPFRC samples cast vertically (0,26x1x0,09 m). Specimens cut out: a) in the middle – VA4 with 0,16x6 mm fibres; b) at the bottom – VB5 with 0,4x12,5 mm fibres.

Table 5 – Fibre orientation and fibre volume in the UHPFRC specimens

| Specimen  | fibre orientation number ( $\eta_\phi$ ) |      |      | fibre volume, % |
|---|--|------|------|-----------------|
|   | x  | y    | z    |                 |
| <b>H3 (next to the point of casting)</b><br>(cast horizontally (0,26×0,26×0,09 m, 0,4×12,5 mm fibres))            |  |      |      |                 |
| -complete   | 0.63                                     | 0.54 | 0.33 | 1.45            |
| -top (bottom)   | 0.62                                     | 0.60 | 0.24 | 2.30            |
| -bottom (top)   | 0.64                                     | 0.54 | 0.30 | 1.82            |
| -rebar  | 0.62                                     | 0.51 | 0.37 | 1.25            |
| <b>H4 (in the opposite corner)</b><br>(cast horizontally (0,26×0,26×0,09 m, 0,4×12,5 mm fibres))                  |  |      |      |                 |
| -complete   | 0.51                                     | 0.61 | 0.35 | 1.66            |
| -top (bottom)   | 0.56                                     | 0.61 | 0.29 | 1.93            |
| -bottom (top)   | 0.51                                     | 0.68 | 0.27 | 2.40            |
| -rebar  | 0.49                                     | 0.56 | 0.44 | 1.34            |
| <b>H5 (in the middle)</b><br>(cast horizontally (0,26×0,26×0,09 m, 0,4×12,5 mm fibres))                           |  |      |      |                 |
| -complete   | 0.55                                     | 0.63 | 0.30 | 1.38            |
| -top (bottom)   | 0.53                                     | 0.72 | 0.23 | 2.28            |
| -bottom (top)   | 0.58                                     | 0.62 | 0.25 | 2.04            |
| -rebar  | 0.51                                     | 0.60 | 0.39 | 0.90            |
| <b>HB1 (next to the point of casting)</b><br>(cast horizontally (0,26×0,26×0,09 m, 0,16×6 mm fibres))             |  |      |      |                 |
| -complete   | 0.57                                     | 0.54 | 0.39 | 1.47            |
| -top  | 0.65                                     | 0.52 | 0.33 | 2.23            |
| -bottom   | 0.47                                     | 0.68 | 0.35 | 1.58            |
| -rebar (middle)   | 0.50                                     | 0.50 | 0.51 | 0.96            |
| <b>HB3 (in the opposite corner)</b><br>(cast horizontally (0,26×0,26×0,09 m, 0,16×6 mm fibres))                   |  |      |      |                 |
| -complete   | 0.50                                     | 0.62 | 0.38 | 1.36            |
| -top  | 0.50                                     | 0.66 | 0.34 | 1.92            |
| -bottom   | 0.53                                     | 0.59 | 0.40 | 1.96            |
| -rebar (middle)   | 0.47                                     | 0.60 | 0.45 | 0.91            |
| <b>VB1 (3 cm from top, next to the point of casting)</b><br>(cast vertically (0,26×0,09×1 m, 0,4×12,5 mm fibres)) |  |      |      |                 |
| -complete   | 0.52                                     | 0.56 | 0.42 | 1.40            |
| -top  | 0.51                                     | 0.65 | 0.33 | 2.06            |
| -bottom   | 0.47                                     | 0.57 | 0.46 | 1.73            |
| -rebar (middle)   | 0.60                                     | 0.48 | 0.42 | 0.88            |
| <b>VB5 (3 cm from bottom, in the corner)</b><br>(cast vertically (0,26×0,09×1 m, 0,4×12,5 mm fibres))             |  |      |      |                 |
| -complete   | 0.48                                     | 0.57 | 0.44 | 1.48            |
| -top  | 0.37                                     | 0.79 | 0.29 | 1.74            |
| -bottom   | 0.54                                     | 0.54 | 0.44 | 1.89            |
| -rebar (middle)   | 0.46                                     | 0.52 | 0.52 | 1.10            |
| <b>VA1 (6 cm from top, in the middle)</b><br>(cast vertically (0,26×0,09×1 m, 0,16×6 mm fibres))                  |  |      |      |                 |
| -complete   | 0.48                                     | 0.48 | 0.55 | 1.62            |
| -top  | 0.50                                     | 0.59 | 0.42 | 1.66            |
| -bottom   | 0.51                                     | 0.48 | 0.52 | 1.90            |

| Specimen   | fibre orientation number ( $\eta_\phi$ ) |      |      | fibre volume, % |
|--|--|------|------|-----------------|
|  | x  | y    | z    |                 |
| -rebar (middle)  | 0.46                                     | 0.45 | 0.58 | 1.45            |
| <b>VA4 (15 cm from the bottom, ~ in the middle of the sample)</b><br>(cast vertically (0,26×0,09×1 m, 0,16×6 mm fibres)) |  |      |      |                 |
| -complete  | 0.53                                     | 0.52 | 0.46 | 1.32            |
| -top   | 0.48                                     | 0.64 | 0.39 | 1.60            |
| -bottom  | 0.53                                     | 0.52 | 0.46 | 1.66            |
| -rebar (middle)  | 0.55                                     | 0.46 | 0.49 | 0.80            |

The calculated fibre orientation number ranges between "0" and "1", where  $\eta_\phi = 1$  means that all the fibres are aligned with the direction  $\phi$  (axis x, y or z),  $\eta_\phi = 0$  indicates that all fibres are directed perpendicular to this direction. Fibre orientation number and fibre volume were initially calculated for the whole volume of the specimen ("complete" in Table 5), as well as for the smaller subareas each specimen was subdivided into, namely: "top" (the upper area just before reinforcement), "bottom" (the bottom area below reinforcement), and "rebar" (the area including reinforcement bars crossing a specimen). Position of the horizontal samples H3–H5 was flipped 180° (upside down) in the CT–scanner (Figure 12 b)) compared to their initial casting position in the formwork indicated in brackets (bottom, top) in Table 5.

Fibre volume distribution for all the specimens at the top and bottom locations was within 1.58–2.40 % with generally higher values observed for the specimens cast horizontally. Calculation of the complete fibre volume concentration was affected by the area around rebar where the fibre volume was at its lowest value, which could also mean that the fibre volume would be close to 2% if that was taken into consideration. Indeed, fibre volume in the top and the bottom part of the samples was not disturbed by the presence of the main reinforcement and was close to 2%, which gives credibility to the CT method. A slightly lower fibre volume concentration was observed in the specimens VA4 and VB5 (cut out ca. in the middle and at the bottom of the vertical samples) compared to the specimens VA1 and VB1 from the top of the samples in the middle and next to the point of casting.

Based on the data provided in Table 5, the average fibre orientation numbers for various axes and the average fibre volume were calculated and summarized in Table 6 for the specimens with the same fibre type, cast either horizontally (i.e. H3–H5 and HB1–HB), or vertically (i.e. VB1, VB5 and VA1, VA4). The area around the rebar and the complete area of the sample (which also included the rebar layer) were included in the results of the calculations, which resulted in a reduction of the value of the fibre volume. According to Table 6, fairly uniform fibre distribution was observed for the specimens cast in both directions regardless of the fibre type with a slight tendency to a perpendicular alignment in the direction of the vertical axis z for the horizontal

specimens. Moreover, slight deviations were observed in different parts of the specimens (i.e. top, bottom and around rebar, see Table 5). Generally, no obvious fibre alignment in any direction was observed based on the ways of casting, thus, resulting in a fairly uniform fibre orientation in the cast samples.

Table 6 – Averaged fibre orientation and fibre volume in the UHPFRC specimens

| Specimen           | fibre orientation number ( $\eta_\phi$ ) |          |          | fibre volume, % |
|--------------------|--|----------|----------|-----------------|
|                    | $\eta_x$                                 | $\eta_y$ | $\eta_z$ |                 |
| Cast horizontally: |  |          |          |                 |
| <b>H3–H5</b>       | 0.56                                     | 0.60     | 0.31     | 1.73            |
| <b>HB1, HB3</b>    | 0.52                                     | 0.59     | 0.39     | 1.55            |
| Cast vertically:   |  |          |          |                 |
| <b>VB1, VB5</b>    | 0.49                                     | 0.59     | 0.42     | 1.54            |
| <b>VA1, VA4</b>    | 0.44                                     | 0.52     | 0.48     | 1.50            |

#### 4.4 Cracks and air voids

Interpretation of the gray images will be more comprehensive if the corresponding CT numbers associated with each component of the UHPFRC matrix (air voids, cracks, CA, steel fibres and FRP rebar) could be recalculated into real densities. In this study this was done by correlating the X-ray densities of the matrix materials to the real densities ( $\text{kg}\cdot\text{m}^{-3}$ ) and associating them with a user-defined non-linear colour scale (as shown in Figures 14–20):

- 1) Air voids/cracks (black) – 1000;
- 2) FRP reinforcement (red) – 2100;
- 3) CA (Hyperit granite) (green) – 2650;
- 4) Steel fibres (blue) – 7850.

Since high density of small steel fibres lead to absorption of energy during CT scanning, disruption of other density signals was observed in the specimens. Therefore, X-ray densities corresponding to the real densities of the matrix materials had to be adjusted for every specimen as a range of values rather than a certain value indicated above. As a result, in addition to the fibre orientation and fibre volume investigation, presence of air voids, air pockets and cracks was monitored in the specimens and presented in Figures 14–21. Linear dimensions of the samples are shown for each specimen in the notation to Figures 14–20 in brackets. During CT scanning all specimens were “sliced” into ~500 slices in two perpendicular directions (XZ and YZ). Only one direction (YZ plane) was chosen for comparison of all the samples (shown in the middle of the specimen VA1 in Figure 14–1). Direction of the X axis was pointing out of plane YZ (POP) and was not displayed in Figures 14–20. For each sample a section with the most dangerous combination of cracks and voids was selected to be presented in Figures 14–20. Generally, for each



specimen only approximately 10 “slices” out of ~500 contained such problematic areas. Thus, the results can be understood as warning rather than characteristic. For example, air pockets and voids are believed to be a warning sign and can theoretically lead to crack formation threatening the load transfer in the material.

In the vertical specimens VA1 and VA4 (0,16×6mm steel fibres) multiple air pockets were traced around rebar, as well as a few discrete pores (air voids) of different size were observed in the entire body of the specimen (Figures 14,15). Knowing the size of the specimens, the size of these pores was measured to be ~5 and 4 mm for the specimens VA1 and VA4, respectively. Furthermore, multiple almost through-going cracks were initiated in the body of the vertical specimen VB1 (0,4×12,5 mm steel fibres) close to the main reinforcement, as well as hairline surface cracks were observed on the left-hand side of Figure 16–1,2. Suggested possible crack location in the specimen VB1 is shown in black in Figure 21. Besides, large air voids were observed on the ribs of the FRP rebar and in the body of the specimen VB1 (~6 mm), as well as between interconnecting longitudinal and vertical rebar (~8 mm). Specimen VB5 cut out at the bottom of the vertical sample (0,26×0,09×1m, 0,4×12,5mm fibres) had only a few small air voids in the body of the specimen (Figure 17).

Similarly to the specimen VB5, only a few small air bubbles were traced in the specimen H3 cast horizontally (0,4x12,5mm steel fibres) and cut out next to the point of casting (see Figure 16–1). As for the specimens H4 and H5 drilled out from the opposite to the point of casting corner and in the middle of the same sample cast horizontally, respectively, several small and larger air voids (up to ~5 mm) were observed, both around rebar and in the body of the specimens (Figure 16–2,3). In the horizontal specimens HB1 air pockets were observed around the rebar, whereas in the specimen HB3 (0,16x6mm steel fibres) a few small air voids were observed in the body of the specimen (Figures 19,20), which is believed to be normal. The post-processing picture for this specimen looks blurry due to a high energy absorption caused by the large amount of small steel fibres.

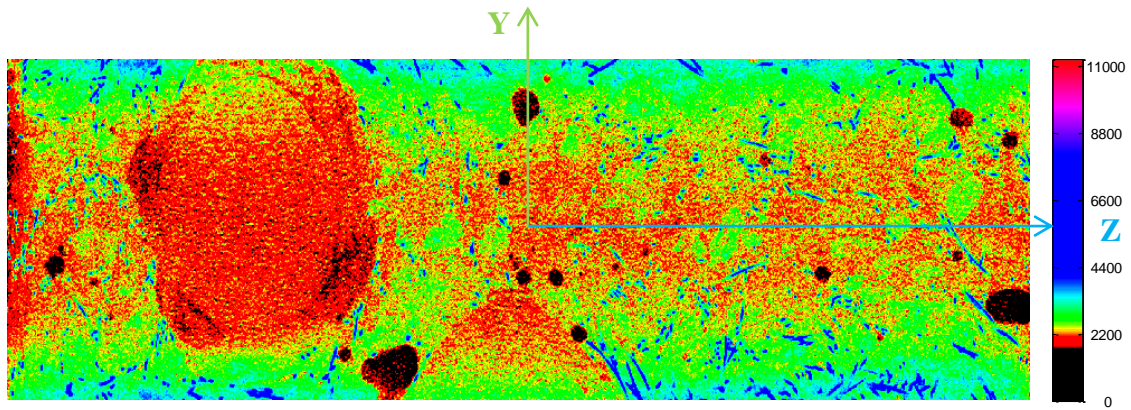


Figure 14 –Specimen VA1 (3,0×3,1×9,0cm) cast vertically (0,16×6 mm fibres): Black regions indicate cracks and air voids, green regions – coarse aggregates (Hyperit), red regions – RFP reinforcement, and blue regions – steel fibres. Recalculated densities are shown on the colour scales to the left ( $\text{kg m}^{-3}$ )

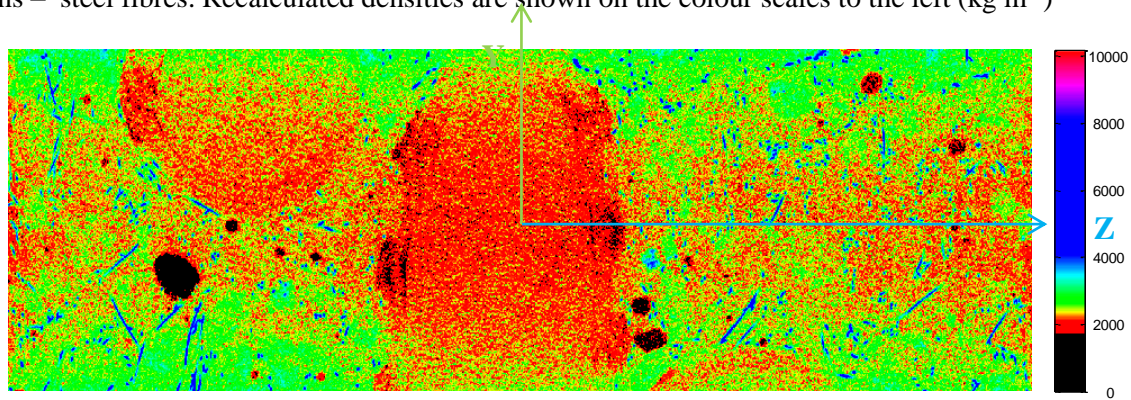


Figure 15 – Specimen VA4 (3,0×2,9×9,0cm) cast vertically (tall, 0,16×6 mm fibres): Recalculation of CT numbers into densities of material components of the UHPFRC specimen. Black regions indicate cracks and air voids, green regions – coarse aggregates (Hyperit), red regions – RFP reinforcement, and blue regions – steel fibres. Recalculated densities are shown on the colour scales to the left ( $\text{kg m}^{-3}$ )

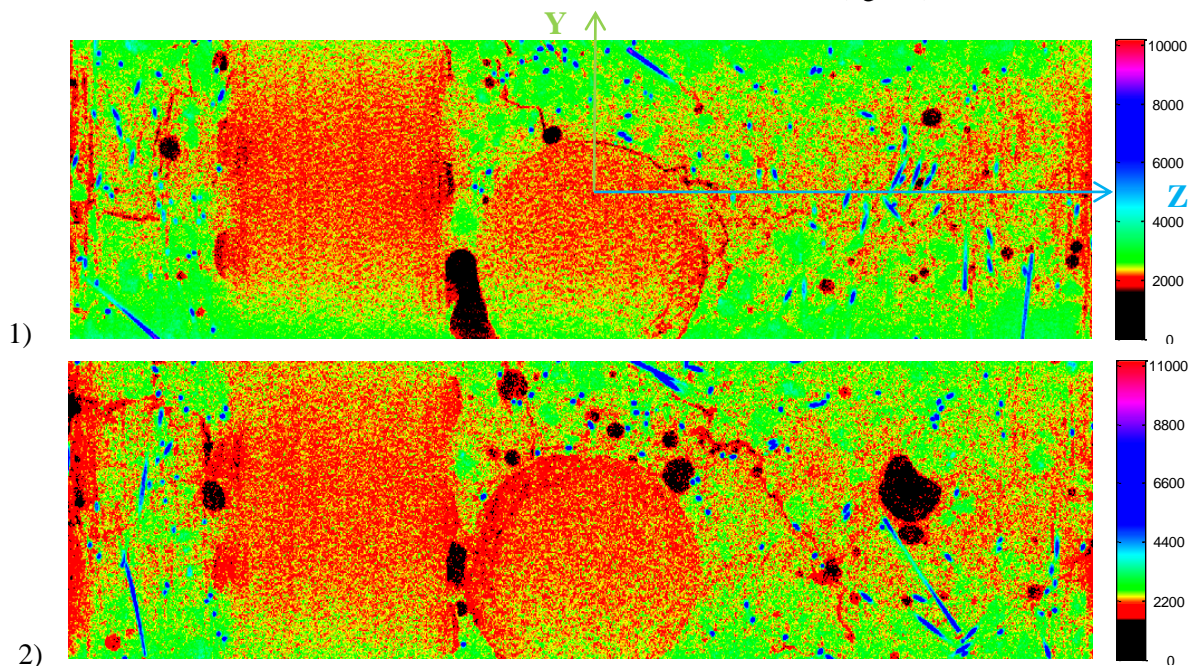


Figure 16 –Specimen VB1 (2,6×3,2×9,0cm) cast vertically (tall, 0,4×12,5 mm fibres): 1), 2) Two slices in YZ direction. Recalculation of CT numbers into densities of material components of the UHPFRC specimen. Black regions indicate cracks and air voids, green regions – coarse aggregates (Hyperit), red regions – RFP reinforcement, and blue regions – steel fibres. Recalculated densities are shown on the colour scales to the left ( $\text{kg m}^{-3}$ )

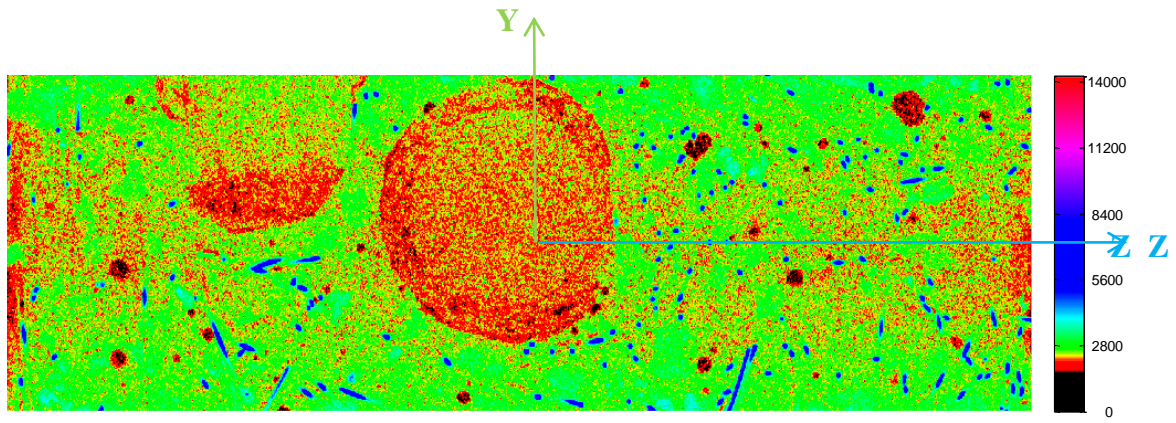


Figure 17 –Specimen VB5 (2,8×3,0×9,0cm); cast vertically (tall, 0,4×12,5 mm fibres): Recalculation of CT numbers into densities of material components of the UHPFRC specimen. Black regions indicate cracks and air voids, green regions – coarse aggregates (Hyperit), red regions – RFP reinforcement, and blue regions – steel fibres. Recalculated densities are shown on the colour scales to the left ( $\text{kg m}^{-3}$ )

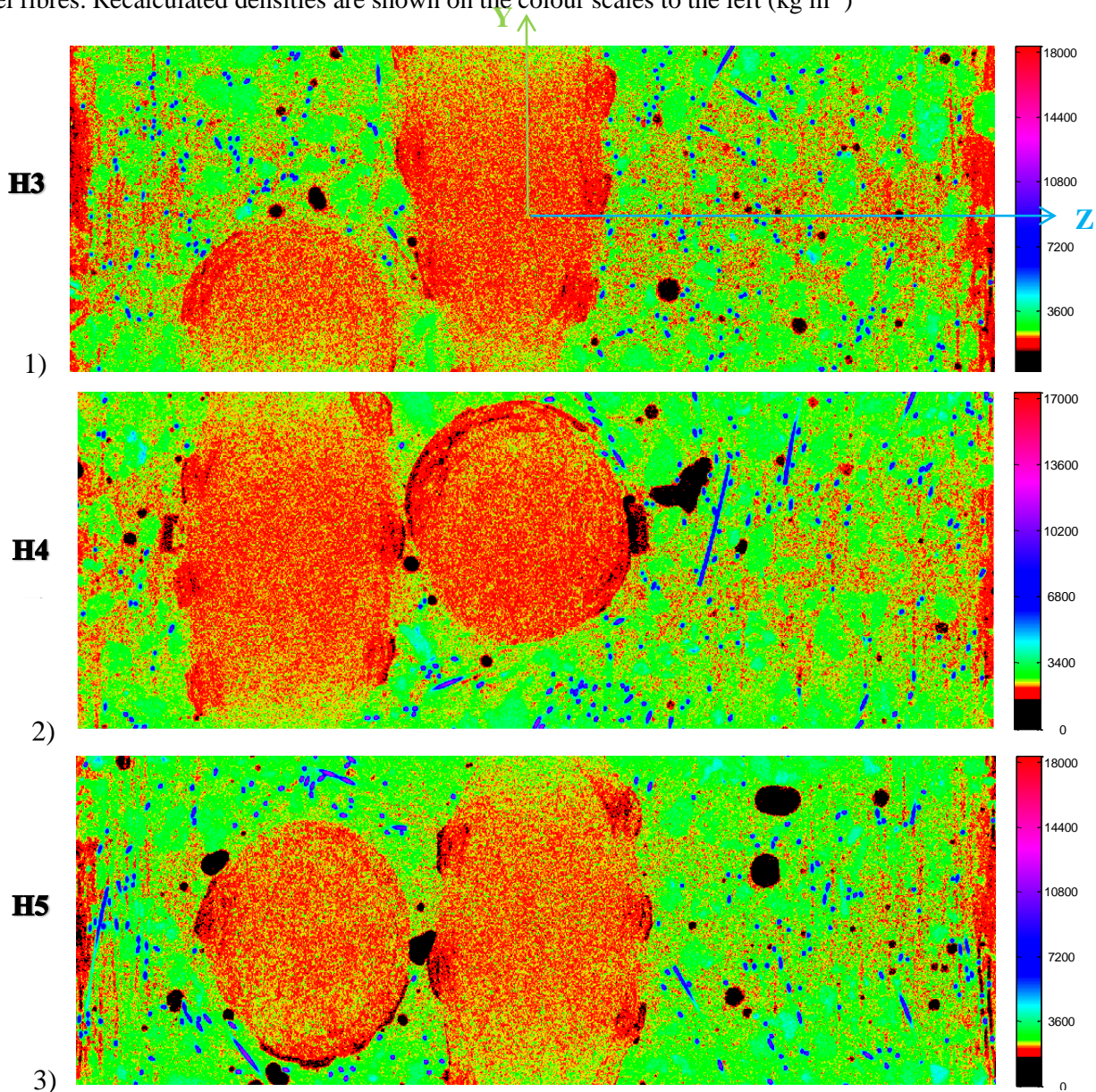


Figure 18 –Specimens cast horizontally (0,4×12,5 mm fibres): 1) H3 (3,3×3,3×9,0cm); 2) H4 (3,3×3,3×9,0cm); 3) Specimen H5 (3,3×3,3×9,0cm): Recalculation of CT numbers into densities of material components of the UHPFRC specimen. Black regions indicate cracks and air voids, green regions – coarse aggregates (Hyperit), red regions – RFP reinforcement, and blue regions – steel fibres. Recalculated densities are shown on the colour scales to the left ( $\text{kg m}^{-3}$ ).

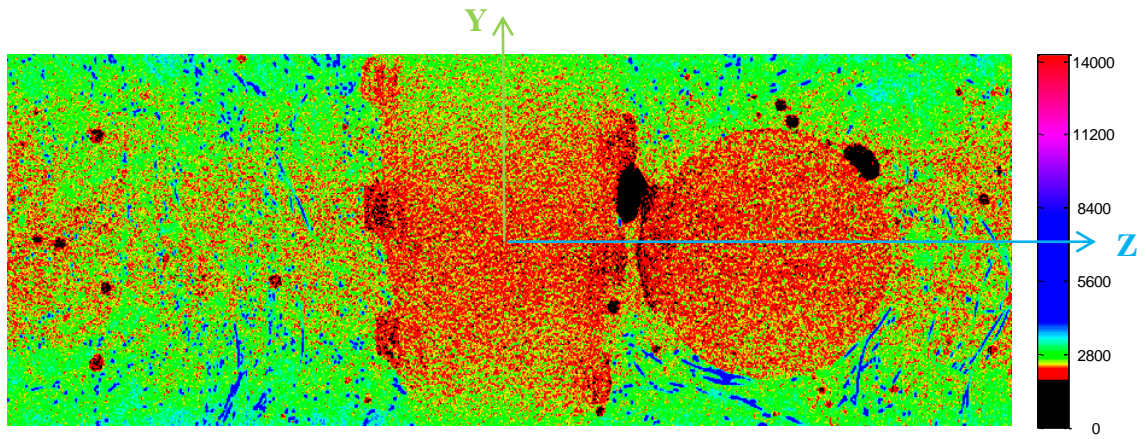


Figure 19 –Specimen HB1 (2,8×3,1×8,8cm) cast horizontally (0,16×6 mm fibres).

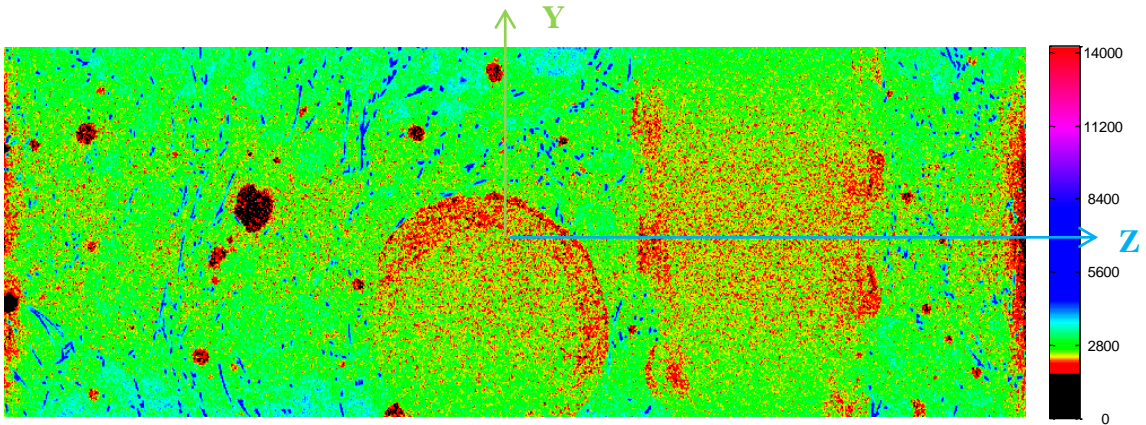


Figure 20 – Specimen HB3 (2,9×3,2×9,0cm) cast horizontally (0,16×6 mm fibres). Recalculation of CT numbers into densities of material components of the UHPFRC specimens. Black regions indicate cracks and air voids, green regions – coarse aggregates (Hyperit), red regions – RFP reinforcement, and blue regions – steel fibres. Recalculated densities are shown on the colour scales to the left ( $\text{kg m}^{-3}$ ).

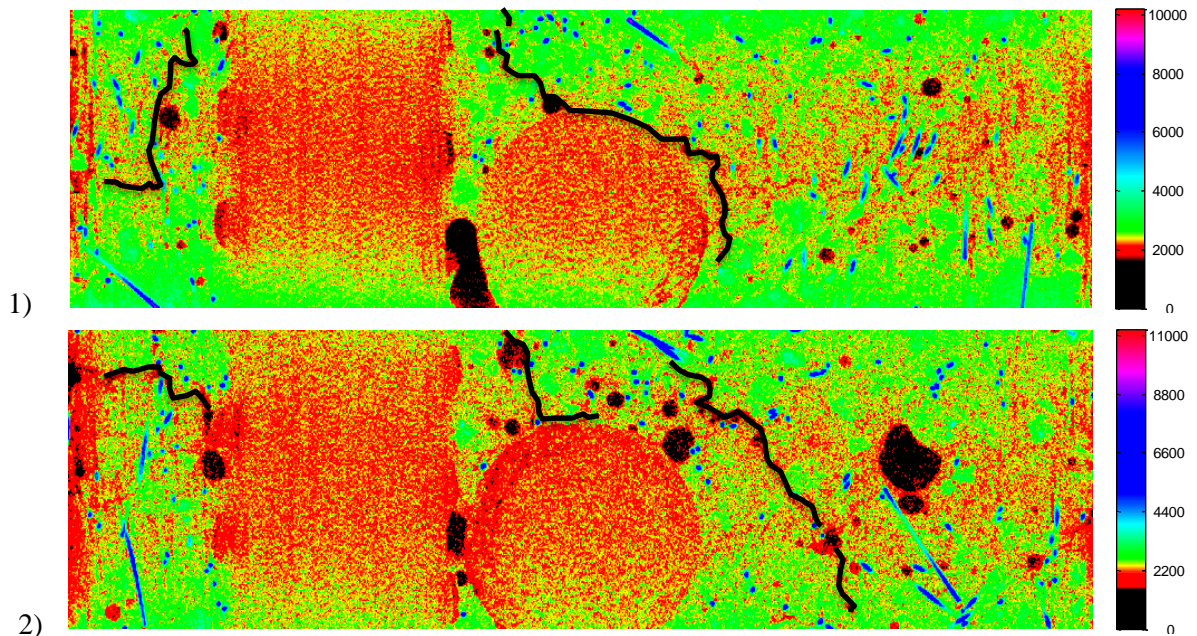


Figure 21 –Specimen VB1 (2,6×3,2×9,0cm) cast vertically (tall, 0,4×12,5 mm fibres): suggested possible crack location

## 5. CONCLUSION

The purpose of the present study was to investigate the suitability of UHPFRC material for construction of TPs for bucket foundation for offshore wind turbines as an alternative to the current welded steel structure solution. One of the prerequisites for the successful application of UHPFRC is that it is possible to use a very high concentration of large reinforcing bars with a durable and crack-free cover layer. Hence, fairly high fibre content is required in the UHPFRC matrix, and the fibres must be able to prevent surface cracking in the TP because of the aggressive marine service environment, as well as to secure adequate reinforcement of the matrix around the main reinforcing bars in order to prevent debonding.

With the purpose of securing a simple placement procedure suited for in-situ casting, the UHPFRC material was designed to have a good workability and cohesion to make the material self-compacting, allowing it to be cast without any need for vibration. The developed UHPFRC material was cast into samples using various placement procedures, and the fibre distribution and orientation was subsequently studied using CT scanning technique. CT scanning is a relatively new method for examination and analysis of volume images of complex microstructures, such as fibre reinforced composites. The CT scanning method yielded excellent results and provided direct pictures of the fibre distribution in the specimens, characterized by the fibre orientation numbers and the fibre volume. Moreover, the calculated fibre content numbers were very close to the real one (2 %) confirming reliability of this method. Also, this technique allowed seeing surface and inner cracks and air voids in the body of a specimen.

Overall, uniform fibre distribution and fibre volume were observed in all the UHPFRC specimens. No particular fibre alignment based on the way of casting was detected with a slight tendency to a perpendicular alignment in the direction of the vertical axis ( $z$ -direction) for the specimens cast horizontally. Fairly uniform fibre distribution was detected for all specimens in all directions. Moreover, the fibre volume at various sections of the specimen was not affected by the fibre type and the way of casting. The behaviour of longer fibres ( $0,4 \times 12,5 \text{ mm}$ ) was not remarkably different from the behaviour of the shorter steel fibres ( $0,16 \times 6 \text{ mm}$ ).

A few normally occurring air pores and relatively large air pockets were traced in some of the tested specimens (with less than 10% of all the pictures of the specimens containing any flaws). "Hairline cracks" were only observed on the surface of the specimen VB1 positioned at the top of the tall sample 3 cast vertically, which could be explained by possible drying of the concrete surface and insufficient hydration caused by water evaporation due to inadequate sealing with plastic sheets. These cracks remained, however, on the surface, i.e. neither did they penetrate deep into the main body of the sample, nor did they cross the main reinforcement. As for the other specimens,

absence of these surface cracks indicated, as expected, that the fine randomly oriented fibre distribution in the cover layer prohibited the occurrence of such cracks. Moreover, should a surface hairline crack appear anyway, its bridging is believed to be taken care of by the fibers crossing the crack plane parallel to the normal of the crack plane or to a lesser extent by the inclined (oblique) steel fibres. Finally, exceptionally durable UHPFRC material can be a promising material for future TP construction.

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