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- 1. 3D J-integral applicable to problems involving large fracture process zones.
- 2. A mode I-II-III decomposed J-integral for large fracture process zones.
- 3. Evaluation of the J-integral using the information from the cohesive zone model.
- 4. Efficient implementation of the mode-decomposed J-integral using cohesive e ements.
- 5. Application of the method to a 3D structure under mode I, II and III loading.

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An evaluation of mode-decomposed energy release rates for a-bitrarily shaped delamination fronts using cohesive ϵ er ents

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Abstract

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Computing mode-decomposed energy release rates in arbit, rily shaped delaminations involving large fracture process zones has not been previously invertigated. The *J*-integral is a suitable method for calculating this, because its domain-independence can be employed to reduce the integration domain to a cohesive interface, and reduce it to a line integral. However, the existing formulations for the evaluation of the mode-decomposed *J*-inverses rely on the assumption of negligible fracture process zones. In this work, a method for the computation of the mode-decomposed *J*-integrals in three-dimensional problems involving '..._o fracture process zones and using the cohesive zone model approach is presented. The formulation is explicable to curved fronts with non-planar crack faces. A growth driving direction criter: n, thick takes into account the loading state at each point, is used to render the integration pe and to decompose the *J*-integral into loading modes. This results in curved and non-planar in e.g. ution paths crossing the cohesive zone. Furthermore, its implementation into the finite element fram, work is also addressed. The formulation is validated against virtual crack closure technique ($\sqrt{\gamma} \in \Gamma$) and linear elastic fracture mechanics (LEFM)-based analytical solutions and the signific nce an ' generality of the formulation are demonstrated with crack propagation in a three-diment formal composite structure.

Delamination of with, Cohesive zone model, Finite element analysis, Energy Release Rate, 3D

Keywords:

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J-Integral

1 1. Introduction

Laminated composite materials are built by stacking plies with different material and reinforcement orientations, e.g. fiber reinforced polymers. During service, excessive in terlaminar stresses can lead to a loss of cohesion between constituent layers. This failure trenches in is known as delamination, and it is one of the most common cause of failure in structure. Inde of layered materials. Therefore, analyzing the onset and growth of delamination is essential for any mechanical application of laminated composites. In this regard, the finite element (FE) method has become an indispensable tool for designing layered composite structures and predicting the service life.

The most common methods for predicting intribuniar failure can be divided into two main approaches: Methods based purely on fracture neclanics and methods based on the concept of the cohesive zone model (CZM) [1, 2]; the latter of which combines the framework of fracture mechanics and damage mechanics.

In the fracture mechanics approach, 'su ily a local Griffith's criterion [3] is used to predict delam-ination growth, i.e., the energy 1^{-1} erg erg \mathcal{G} , is compared to the interlaminar fracture toughness, \mathcal{G}_{c} . Two of the most common e⁻tr, ⁻tion methods for the energy release rate (also called the crack exten-sion force) rely either on the VCCT [4] or the J-integral [5]. Then, applying Griffith's criterion, crack propagation occurs at the points where $\mathcal{G} \geq \mathcal{G}_c$. This local energy balance criterion implies a negligible fracture process zone. Con ersely, CZMs can capture the fracture energy dissipation mechanisms of quasi-brittle materials, uch as the formation of micro cracks ahead of the crack tip before complete separation ϵ , the crack faces occurs. Therefore, the CZM approach is a suitable means of predicting crack propasition when a non-negligible fracture process zone is present. The strain singularity at the tip o' a sharp crack is removed by accounting for a cohesive zone (CZ), where the material un-dergoes degradation until complete decohesion. The mechanical behavior of the interface is modeled by means of a damage variable, which is a measure of the degradation of the mechanical properties

of the material ahead of the crack tip. When the damage variable reaches its maximum value, a new crack surface is created. Moreover, CZMs are particularly suited for simulation if interlaminar cracks in laminated structures because the delamination is confined to propagate between two adjacent plies. Thus, when a progressive delamination simulation is solved using an Lar a lalysis, the potential failure surfaces are known in advance, and the cohesive elements can be officiently located.

Under static loading conditions, existing CZMs [6-12] do p \therefore require the energy release rate to be computed in order to simulate crack growth. However, some of the recently published methods for simulating fatigue-driven delamination based on CZM [13–18] line the rate of the local fatigue damage with any variant of the Paris' law [19]. The Paris' law-n' expressions relate the crack growth rate with a power law function of the loading level in terms of a fracture mechanics parameter [13, 20], usually the stress intensity factor, K, or the energy release rate, \mathcal{G} , where only the latter is relevant for a CZM. Therefore, computing the energy releasing is required in order to integrate the rate of the local fatigue damage. In this regard, the J-in. oral directly equates to \mathcal{G} [21]. In fact, the benchmark study of the simulation methods for 'atigu -driven delamination using a CZM approach presented in [18] showed a better performance or th_{n} m thods using the J-integral as the means of extracting the energy release rate.

The path-independence function two-dimensional J-integral makes it very attractive in practice, since it avoids the need for a cura e computations on the stress field at the crack tip; something which is hard to deal with ir an FL ^cramework. For this reason, considerable effort has been devoted to extending the application ity i the J-integral to three-dimensional (3D) domains [22–32]. The published extensions of the J-integral for its evaluation in three-dimensional problems, where the crack extension force may change along the crack front, commonly employ two approaches. The first is a point-wise evaluation of the -integral on a cross-section normal to the crack front, resulting in the combination of a cont η integral and a surface integral defined over the area enclosed by the contour. See [30] for a detailed description. Computing the surface integral requires accurately calculating the field quantities at the crack tip. For this reason, the boundary element method is commonly used [27, 30].

The second approach is the equivalent domain integral over a finite volume surrounding the crack front [25, 26, 33]. With this method, capturing the singular field near the γa κ tip is not required which is why it is usually applied in a FEM framework. Regardless, the a plicability of most of these J-integral extensions to three-dimensional domains is restricted to ertain assumptions such as plane-strain/stress, i.e., at the vicinity of the crack tip, or plane crack. By employing curvilinear coordinates, Eriksson [34] and Fernlund et al. [35] obtained generalized expressions applicable to curved cracks with non-planar crack surfaces. In [34], a volume-ind pendent integral expression for evaluating the crack extension force is derived from the principle of virtual work. In [35], the decrease of the potential energy with crack extension is employed to obtain a general path-area independent J-integral expression for non-planar cracks with curvea area fronts. In both cases, the fracture process zone is considered negligible and the mode-decomposition is not addressed.

⁶² Delamination propagation can be described to output a combination of the three basic fracture ⁶³ modes (Modes I, II and III) [36], and the fra ture resistance of the interface, under both static and ⁶⁴ fatigue loading, highly depends on the mood mixity conditions. Consequently, the delamination models ⁶⁵ available in the literature [13, 20, ?7] are 'resid on a mode-decomposed definition of the load, expressed ⁶⁶ in terms of the energy release 1...'reside on a mode-decomposed definition of the load, expressed ⁶⁷ integral into fracture modes as retool for extracting energy release rates, becomes necessary.

In this work, a new ploce lure to numerically evaluate mode-decomposed J-integrals in a 3D body undergoing delamination is p. cented. The method is applicable to curved crack fronts with non-planar crack surfaces. More, er, ne method enables, for the first time, the application of the J-integral in 3D problems involvi g large fracture process zones. In addition, in contrast to current cohesive models where the mode mixity is evaluated locally (point-wise) using the interface separation, the presented J_{-} ter al formulation enables defining the mode mixity parameter as a function of the mode dec r posed \mathcal{G}_I , \mathcal{G}_{II} and \mathcal{G}_{III} (global measures). This is of crucial importance to improve the accuracy of the simulation of delamination propagation under quasi-static and fatigue loading.

The formulation is derived from the general expression of the J-integral for 3D curved delaminations

with non-planar surfaces expressed in terms of curvilinear coordinates [35], which re. s on LEFM. Its application to cohesive interfaces is addressed in Section 2, while its application in an FE framework is presented in Section 3 and Appendix A. In Section 4, the to, rulation is applied to a moment-loaded double-cantilevered-beam (DCB) and the mixed-m. 19 -components are compared to the mode-decomposed energy release rates obtained from VCC Γ . In Cection 5, the formulation is applied to an embedded penny-shaped crack in a steel cylinder and two determined mode-components are compared to and validated against an analytical LEFM-Lased olution [38]. In Section 6, the formulation is used to compute the J-integral components of a partially reinforced end-loaded split (ELS) specimen with a non-straight crack front and non-p., "ar crack interface. Finally, the conclusions on this work are presented.

87 2. Formulation of mode-decomposed encestre release rates

In this section, the formulation of the mode decomposed energy release rates in 3D delaminations, modeled using a cohesive zone model appro. ch, is presented. The point of departure is the generalized J-integral for non-planar curved cracks c. * ined by Fernlund et al. [35].

2.1. Assessment of the energy release, ate by means of the J-integral formulation in curvilinear coor dinates

Consider an elastic body (cf. Figure 1), with a crack, subjected to prescribed tractions, T, and displacements, u, a reprint of its boundary surface (Note that T and u are physical entities that are not yet described in any particular coordinate basis). In a general three dimensional domain, both the crack surfices and the crack front may be curved. Let θ^i , i = 1, 2, 3, be an orthogonal curvilinear coordinate. $stem \cdot$ ith origin at a given point P along the crack front. This local coordinate system is oriented 'uc' unat, at point P, θ^3 is normal to the crack surface, θ^2 is the coordinate along the crack front and θ^1 is the direction of crack propagation, which is always tangent to the crack surface and qq perpendicular to θ^2 and θ^3 .

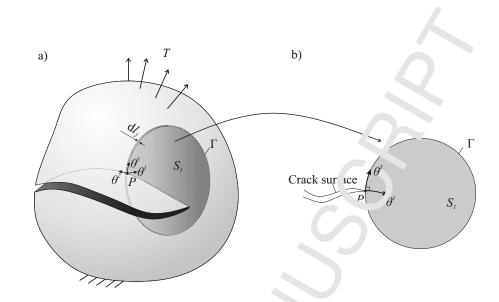


Figure 1: a) Three-dimensional body undergoing a delaminate , with curved front and non-planar crack surfaces. b) The integration domain is a slice of infinites, pal thickness, dl_2 .

Let us focus on a thin slice of elemental thickney, dl_2 , of the cracked body, which contains P (cf. Figure 1). Note that an infinitesimal length seq. ent, dl_i along a curvilinear axis, θ^i is given by:

$$dl_i = \sqrt{g_{ii}} d\theta^i \tag{1}$$

where g_{ij} is the covariant metric tendor. In the absence of body forces, the change in potential energy, II, per unit of newly created mack area is [35]:

$$-\frac{\mathrm{d}\Pi}{\mathrm{d}A} = -\int_{V} \frac{\mathrm{d}W}{\mathrm{d}A} \mathrm{d}V + \int_{S} T^{i} \frac{\mathrm{d}u_{i}}{\mathrm{d}A} \mathrm{d}S$$
(2)

where dA is the element, 'c ack area extension, V is the volume of the slice, S is the surface surrounding V, W is the str in ener y density, T^i are the contravariant components of the traction vector and u_i are the cov riant components of the displacement vector.

The infinite interval thickness of the slice, allows to lump the three-dimensional slice into a surface S_1 , defined by $^2 = 0$ (d $l_2 \rightarrow 0$). Then, by applying Green's theorem, and under the assumption of small deformations, elastic material behavior, symmetry of the stress tensor and equilibrium conditions, the decrease in potential energy per unit area extension is expressed, in [35], as a contour integral and an ¹¹² area integral on the surface S_1 :

$$J = -\frac{\mathrm{d}\Pi}{\mathrm{d}A} = \frac{1}{\sqrt{g_{11}}} \oint_{\Gamma} \left(W n_1 - T^i \frac{\partial u_i}{\partial \theta^1} \right) \mathrm{d}\Gamma - \frac{1}{\sqrt{g_{11}g_{22}}} \int_{S_1} \frac{\partial}{\sigma} \left(\sigma^{i_2} \frac{\partial u_i}{\partial \theta^1} \right) \mathrm{d}S \tag{3}$$

where Γ is the contour enclosing S_1 in the clockwise direction and n in the ortward unit normal vector on Γ . Note that in [35], the curvilinear coordinate system is rotated 90° round the θ^1 -coordinate.

The *J*-integral is equivalent to the energy release rate, ζ , for an elastic material response. In a three-dimensional body, the energy release rate may vary alon. the crack front. Therefore, in order to assess the delamination extension force in three-dimensional problems, it is customary to compute the point-wise value of *J* as a function of the crack from problem, *P*.

119 2.2. Application to cohesive interfaces

Unlike LEFM, the CZM relies on the existence on a band of material ahead of the crack tip (known as the cohesive zone (CZ), where the material c haves nonlinearly [1, 2]. In the CZ, a cohesive traction distribution acts on the separating sv faces, hus avoiding stress singularities at the tip of sharp cracks. The constitutive law that relates the control ve tractions to the displacement jumps at the interface is governed by a scalar damage varial e. The damage variable evolves monotonically with time to ensure irreversibility. To guarante, the proper energy dissipation under mixed-mode conditions, in [11] the cohesive law is formulated in ι one-dimensional space, where the equivalent mixed-mode traction, μ , is related to the norm $\langle i \rangle$ the displacement jump, λ . Thus, the equivalent one-dimensional displacement jump, λ , is defined as:

$$\lambda = \sqrt{\left(\delta_1\right)^2 + \left(\delta_2\right)^2 + \left(\langle\delta_3\rangle\right)^2} \tag{4}$$

and the equival nt one-dimensional interface traction, μ , is related to λ as follows:

$$\mu = \left(1 - \mathcal{D}^K\right) K\lambda \tag{5}$$

where $\mathcal{D}^{K} \in [0, 1]$ is a scalar damage parameter degrading the constitutive tangent stifn. ss, K, and $\langle \rangle$ is the Macaulay bracket ensuring that negative normal opening (interpenet. tio , of crack faces) does not affect damage development.

A sketch of the bilinear cohesive law used in [11] is represented in 7 igure 2. An energy-based damage variable, \mathcal{D}^e , is introduced as the ratio of specific dissipate 1 energy due to fracture, ω_d (Figure 2.b), and the fracture toughness, \mathcal{G}_c (Figure 2.a). Thus, \mathcal{D}^e ranges from σ to 1, and can be understood as the degree of crack development, taking a value of 0 if the degred tion process is yet to start, and a value of 1 if the crack is fully developed.

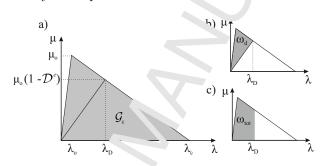


Figure 2: Equivalent one-dimensional coherine law. The shadowed area in a) represents the fracture toughness, \mathcal{G}_c , in b), the specific dissipated energy, ω_d and in c), the total specific work, ω_{tot} , for a given state of damage.

The constitutive law is formed by an initial elastic region, before damage initiation, and a softening region. The onset and proparation of delamination are limited by the onset mixed-mode displacement jump, λ_o , and the critical mixed-mode displacement jump, λ_c , such that the applicability of the energy-based damage variable, \mathcal{D}^e , is restricted to:

$$\mathcal{D}^{e} = 0 \quad \text{for} \quad \lambda_{\mathcal{D}} \leq \lambda_{o}$$
$$\mathcal{D}^{e} = \frac{\omega_{d}}{\mathcal{G}_{c}} \quad \text{for} \quad \lambda_{o} \leq \lambda_{\mathcal{D}} \leq \lambda_{c}$$
$$\mathcal{D}^{e} = 1 \quad \text{for} \quad \lambda_{\mathcal{D}} \geq \lambda_{c}$$
(6)

where $\lambda_{\mathcal{L}}$ is the mixed-mode displacement jump associated to the current damage state.

When applied to delamination modeling in laminated composite materials, the cohesive behavior is lumped into the interface between subsequent plies. In [35], it is demonstrated that the *J*-integral

of Equation (3), generalized in terms of curvilinear coordinates for cracks with cur ed front and non-planar crack surfaces, is path-area-independent. Then, for the measurement of the delamination extension force in 3D laminated structures modeled using a CZM approach the potential the potential of Equation (3) can be employed to reduce the contour Γ to the cohesive interface (cf. Figure 3), similar to what is done with the two-dimensional form of the *J*-integral [5]. Therefore, because of the zerothickness of the cohesive interface, and taking into account the the opening displacements are very small, the differentials $n_1 d\Gamma \approx d\theta^3$ and dS in Equation (3) vanish. Thus, Equation (3) is reduced to:

$$J = -\frac{1}{\sqrt{g_{11}}} \int_{\Gamma} \left(T^i \frac{\partial u_i}{\partial \theta_\perp} \right) d\Gamma$$
(7)

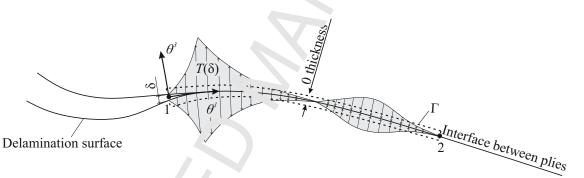


Figure 3: The integration path, (dz) ded line), is reduced to the zero-thickness cohesive interface. Let σ^{ij} be the contravariant c_{ij} population of the cohesive stress tensor. Then, the contravariant traction vector at the clock aces is given by:

$$T^i = \sigma^{ij} n_j \tag{8}$$

where n_j is one outward unit normal vector on the contour Γ , i.e., on the crack surfaces. Thus, n_j vanishes for $\dot{j} \neq 3$ and Equation (7) reads:

$$J = -\frac{1}{\sqrt{g_{11}}} \int_{\Gamma} \left(\sigma^{i3} \frac{\partial u_i^+}{\partial \theta^1} + \sigma^{i3} \frac{\partial u_i^-}{\partial \theta^1} \right) \mathrm{d}\theta^1 \tag{9}$$

where u^+ and u^- are the displacements at the upper (⁺) and lower (⁻) crack surfaces, respectively.

Finally, by introducing the displacement jump as the separation of two initially coinciding points on the interface, defined as:

$$\delta_i = \left(u_i^+ - u_i^-\right) \tag{10}$$

¹⁵⁹ the curvilinear CZ *J*-integral, when applied to cohesive interfaces, can be expressed as:

$$J = -\frac{1}{\sqrt{g_{11}}} \int_{CZ} \left(\sigma^{i3} \frac{\partial \mathcal{I}}{\partial \theta^1} \right) \, \mathrm{d}\theta^{-} \tag{11}$$

Observe, in Figure 3, that the integration path is the entire CZ so that all the cohesive stresses contribute to the CZ J-integral. Further details on the integration path shape and limits in 3D applications are provided in Section 2.3.

163 2.3. Integration paths

As demonstrated in Section 2.2, the integration domain of the curvilinear CZ J-integral applied to cohesive interfaces is a slice of in. ritesim 1 thickness, dl_2 , lumped into the delamination interface. Thus, the integration domain is edu ed to a path contained in the delamination interface that follows the direction of crack proparation, θ . In order to compute the J-distribution in three-dimensional structures, the interface can be divided into infinite slices. Obviously, the J-value of each slice is unique and is obtained . 'n the integration path is covered in its entirety, i.e., going through the entire cohesive zone from the completely damaged zone (point 1 in Figure 3, with zero cohesive stress) to the end of the zone in elastic regime (point 2 in Figure 3, with zero cohesive stress).

In LEFM, the propagation direction, θ^1 , is assumed to be the normal to the crack front at the point *P*, where the crack front is the line separating the damaged and undamaged parts (cf. Figure 4.b). However, the propagation direction as the normal to the crack front does not apply for CZM, due to the existence of a cohesive zone of variable length. The authors have recently introduced the concept of the growth driving direction (GDD) for CZM [39], as an analog to the crack propagation direction in LEFM. The GDD is defined as the gradient vector field of the scalar energy-based damage,

$$\text{GDD} = -\nabla \mathcal{D}^e$$

(12)

Thus, the GDD is normal to the energy-based damage, \mathcal{D}^e , isolinov (cf. Figure 4.a) and it converges with the normal to the crack front in LEFM (cf. Figure 4.b) in the lim ting case where the length of the CZ approaches zero. Therefore, by making use of the criber on presented in [39], θ^1 can be defined according to the GDD. In this way, the integration γ aths, defined along the θ^1 -coordinate, never intersect and the three-dimensional structure can be und estood as the aggregation of infinite individual slices of infinitesimal thickness which coving a ward propagating in the GDD. It is worth mentioning that the damage isolines may not b μ mentioning the CZ, leading to slices with double curvature if, in addition, the cohesive interface mid-surface is non-planar.

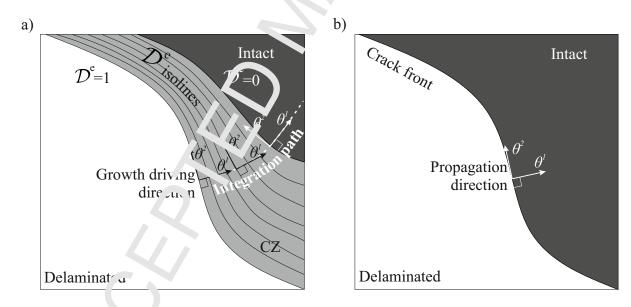


Figure 4: a) The gowth driving direction (GDD) is assumed to be the normal direction to the energy-based damage isolities in the CZM framework. The integration paths are tangent to the local GDD direction. b) The propagation direction is assumed to be the normal direction to the crack front in the LEFM framework.

It is n t d that, to compute the *J*-value in cohesive interfaces using Equation (11), the contribution of the stress, σ^{i3} , and displacement jump slope in the GDD direction, $\frac{\partial \delta_i}{\partial \theta^1}$, in the elastic regime is needed. However, the criterion in Equation (12) for identifying the GDD, based on the negative

gradient of the energy-based damage, \mathcal{D}^e , is only meaningful for $\mathcal{D}^e \in]0, 1[$ (see Equation (6)). Therefore, a new criterion to identify the GDD in the elastic regime must be use. If this regard, another criterion, which is also active before the initiation of the degradation process, proposed in [39]:

$$GDD = -\nabla \left(\frac{\omega_{tot}}{\mathcal{G}_c}\right) \tag{13}$$

where $\frac{\omega_{tot}}{\mathcal{G}_c}$ is the ratio between the total specific work (cf. Fig. re 2 .) ϵ nd the fracture toughness. Both the conservative and the non-conservative work are computed in unis criterion. This implies that as soon as two initially coinciding points separate from each coher ($\lambda > 0$), some elastic energy is stored which makes this criterion active before damage of \ldots fince the damage is initiated, both criteria lead to the same GDD solution.

198 2.4. Mode-decomposition of the CZ J-integral J. r. application to cohesive interfaces

A crack can grow under a combination of the ploading modes [36]: the opening mode (mode I), the sliding mode (mode II) and the tearing mode (mode III). Mode I is defined as normal to the cohesive interface mid-surface, mode II, tangent we the mid-surface in the propagation direction and mode III, tangent to the mid-surface and perpendicular to mode II. In this work, the crack propagation direction is defined as the GDD (cf. pectrum 2.3). This implies that the mode II direction is also defined as the GDD, and the mode III direction is defined as the direction perpendicular to the mode I and mode II direction.

For the mode decon, γ s² ion of the *J*-integral, the integrands in Equation (11) must be decomposed according to the local besis vectors, aligned with the three loading modes directions. Thus, θ^1 is locally coincident with the GDD (i.e. tangent to the mid-surface), θ^3 is normal to the mid-surface, and θ^2 is normal to ν^2 and θ^3 . Moreover, since θ^i are orthogonal curvilinear coordinates, the local covariant and contreparation basis vectors are collinear.

At an interface modeled using a CZM approach, only three uncoupled components of cohesive stresses (σ^{13} , σ^{23} and σ^{33}) result from the displacements jumps between crack faces (δ_1 , δ_2 , δ_3). The

quantities σ^{13} and $\frac{\partial \delta_1}{\partial \theta_1}$ contribute to mode II, σ^{23} and $\frac{\partial \delta_2}{\partial \theta_1}$, to mode III, and σ^{33} and $\frac{\partial \delta_3}{\partial \theta_1}$, to mode I crack loading. Hence, the mode-decomposed CZ *J*-integrals are defined coording to the local θ^i coordinate system such that the terms with i = 3 are attributed to Mode I, $\dot{i} = 1$, to Mode II and i = 2, to Mode III:

$$J_{I} = -\frac{1}{\sqrt{g_{11}}} \int_{CZ} \left(\sigma^{33} \frac{\partial \delta}{\partial \theta} \right) e^{i t}$$

$$J_{II} = -\frac{1}{\sqrt{g_{11}}} \int_{CZ} \left(\sigma^{13} \frac{\partial \delta_{1}}{\partial \theta^{1}} \right) e^{i t}$$

$$J_{III} = -\frac{1}{\sqrt{g_{11}}} \int_{C} \left(\sigma^{2z} \frac{\partial \delta_{2}}{\partial z} \right) d\theta^{1}$$
(14)

Note that Equation (14) represents an expression \mathbf{r} to the evaluation of the mode-decomposed energy release rates in arbitrarily shaped delamination. The volving a large fracture process zone modeled using a CZM approach. The integration paths are curved lines crossing the CZ formed according to the GDD, which is rendered taking into are a part the loading state at each point. Moreover, the mode II is collinear with the GDD and mode \mathbf{h} , is perpendicular to it. This results in the mode directions not being constant along the integration path 3. On the contrary, in LEFM approaches, mode II and mode III directions are the normal and tangent to the crack front, respectively.

For 3D planar cracks i.e. ribed by a rectangular Cartesian coordinate system, the work by Rigby and Aliabadi [30] and Ei. sson [40] propose equivalent expressions for the mode-decomposed Jintegrals, which are 'n agreement with those presented in Equation (14) in the limiting case where the length of t' e CZ ands to zero. Moreover, by limiting the integration domain to the cohesive interface, the error committed in the decomposition of the far-field quantities due to the out-of-plane stress gradents [2^{\circ}, 40] is avoided, thus, allowing the integration domain of the CZ J-integrals to extend to large inacture process zones.

²³¹ 3. FE-discretized mode-decomposed CZ *J*-integrals

In the following, the formulation presented in Section 2 is applied in an FL immework. The CZM used in this work, and its implementation to FE, was presented by Turca eval. in [10, 11]. Complying with the cohesive element definition, the interfacial tractions and displatment jumps are expressed in a local Cartesian coordinate system, x_i , located on the deform d mid- inface, \overline{S}_{coh} , defined as the average distance between two initially coinciding points, P^- and $-^+$ (cf. Figure 5). The direction cosines of the local Cartesian coordinate system are the no. mal, \hat{c}_3 , and tangential, \hat{e}_1 and \hat{e}_2 , unit vectors to \overline{S}_{coh} . Furthermore, employing the criterion deve. red in [39], the local tangential coordinates can be oriented in such a way that x_1 and x_2 are the tangen all and normal coordinates to the GDD, respectively.

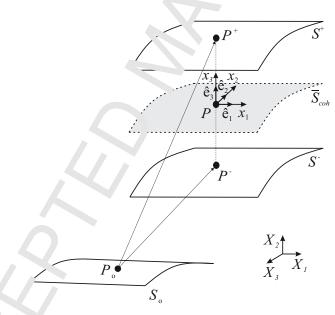


Figure 5: Description of the undeformed, S_o , and deformed, S^+ and S^- , configurations of the delamination interfaces. The quantities of the CZM are calculated at the deformed misurface, \overline{S}_{coh} , in terms of the local Cartesian coordina $\propto r$ P is a point located at the mid-surface in the deformed configuration, while points P^+ and P^- are points belonging to the upper and lower crack surfaces, respectively. P, P^+ and P^- coincide at P_o in the undeformed configuration.

To nu ν rically integrate Equation (14), trapezoidal integration is employed (although any other numerical integration method could be used). Thus, the curved integration pathline is discretized into small linear subintervals tangent to the curvilinear coordinate θ^1 . The quantities in the integrand

of Equation (14) must, therefore, be defined according to the local Cartesian coordin. e system, x_i , with a locally coincident direction with the covariant and contravariant basis rectors of the orthogonal curvilinear coordinate system, θ^i . Further details on the discretization of the normalization with the FE method, such as the tracking of the integration path, as well as its linities are addressed in Appendix A.

After the discretization of the cohesive interface into FE. Le numerical integration of Equation (14), performed under the trapezoidal rule, reads:

$$J_{I} \simeq -\sum_{k} \left[h^{k} \left(\frac{\sigma_{3}^{k} \frac{\partial \delta_{3}^{k}}{\partial x_{1}} + \sigma_{3}^{k-1} \frac{\partial \delta_{3}^{k+1}}{\partial x_{1}}}{2} \right) \right]$$

$$J_{II} \simeq -\sum_{k} \left[h^{k} \left(\frac{\sigma_{1}^{k} \frac{\partial (x_{1}^{k}) + \sigma_{13}^{k+1} \frac{\partial \delta_{1}^{k+1}}{\partial x_{1}}}{2} \right) \right]$$

$$J_{III} \simeq -\sum_{k} \left[h \left(\frac{\partial (x_{2}^{k} \frac{\partial \delta_{2}^{k}}{\partial x_{1}} + \sigma_{23}^{k+1} \frac{\partial \delta_{2}^{k+1}}{\partial x_{1}} \frac{\partial \delta_{2}^{k+1}}{\partial x_{1}}}{2} \right) \right]$$

$$(15)$$

where h^k is the integration interval h^{ot1} , approximated to the Euclidean distance between two consecutive points along the integration path, P^k and P^{k+1} .

The accuracy on the cor.p. ⁺ation of the CZ *J*-integral depends both explicitly on the integration interval length, and implicit 7 on the size of the cohesive elements due to the discretization of the displacement field in the FL model.

4. Compariso 1 with mode-decomposed energy release rates extracted by VCCT

The capabilities of the CZ *J*-integral formulation presented are assessed by comparing the energy release rate n. do components of a moment-loaded DCB model obtained by VCCT. The specimen is 30 mm lo. c, 6 mm wide and 3 mm thick (Figure 6). The elastic properties, corresponding to a unidirectional laminate made of a carbon fiber reinforced polymer (CFRP) material used in aeronautical applications, are listed in Table 1. The fracture properties of the interface are presented in Table 2.

The fracture toughnesses, \mathcal{G}_{Ic} , \mathcal{G}_{IIc} and \mathcal{G}_{IIIc} , are close to typical values for this mater. I. The inter-laminar strengths, τ_{Ic} , τ_{IIc} and τ_{IIIc} , have been selected such that the fractu. Dr cess zone is small, to enable a fair comparison between the VCCT and the cohesive zone model which ensuring a minimum number of 3 damaged elements spanning the cohesive zone, to provide an accurate distribution of the tractions ahead of the crack tip [41, 42]. The specimen arms are nodeled in the commercial FE code ABAQUS [43] using C3D8I hexahedral elements. The undeformed elements are 0.4 mm wide, 0.2 mm long and 0.5 mm thick. The delamination front is completely straight and located at the mid-surface at a distance of 15.1 mm from the loading application edges. A combined I, II and III fracture mode is created by applying four force pairs (Figure 6). M1 and $1^{\prime}2$ generate uneven opening Y-moments at the upper and lower arms, respectively. M3 and M4 goverate even tearing Z-moments at both arms. The resultant bending moments are listed in Table 3

Lamina 'e _P operties		
E_{11} : longitudinal Young's \dots	154	GPa
$E_{22} = E_{33}$: transversal Young 5 modulus	8.5	GPa
$G_{12} = G_{13}$: shear modulus in the longitudinal planes	4.2	GPa
G_{23} : shear modulus is the ι answersal plane	3.0	GPa
$\mu_{12} = \mu_{13}$: Poisson's pefficient in the longitudinal planes	0.35	-
μ_{23} : Poisson's coefficient . The transversal plane	0.4	-

Table 1: Elastic properties of the la.. [:] ate used in the simulation studies of the moment-loaded DCB and the ELS specimens.

Interface properties		
\mathcal{G}_{Ic} : mode 1. cture toughness	0.3	N/mm
$\mathcal{G}_{IIc} = \mathcal{C}_{IIIc}$ modes II and III fracture toughness	3	N/mm
τ_{Io} : m de / inte laminar strength	10	MPa
$\tau_{IIo} = \tau_{III}$, r odes II and III interlaminar strengths [11]	31.62	MPa
η : benzeg ragn-Kenane's interpolation parameter [44]	2	-
K: penalty stiffness	10^{5}	N/mm^3

Table 2: Fre ture p. perties of the interface used in the simulation study of the moment-loaded DCB specimen.

In the F containing the VCCT [4], the energy release rates are evaluated locally, at every node forming the delamination front, using the nodal forces, F_i , and relative displacements between released nodes on the upper and lower crack faces, $u_i^{upper} - u_i^{lower}$:

=

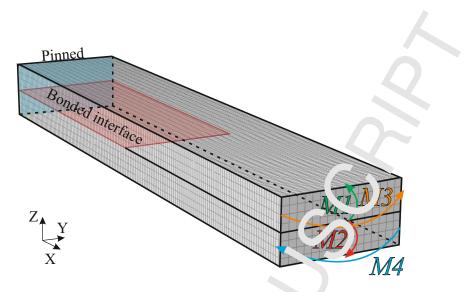


Figure 6: DCB specimen dimensions with four force pairs: 11 a.^{12} generate uneven opening Y-moments, while M3 and M4 generate even tearing Z-moments.

Bending moment	[Nmm]
M1	270
M2	135
M3	960
<u>M4</u>	960

Table 3: Bending moment resultants from the apple tion of the four force couples to the double-cantileveredbeam model.

$$\mathcal{G}_{I} = \frac{1}{2l_{1}^{e}l_{2}^{e}}F_{3}\left(u_{3}^{upper} - u_{3}^{lower}\right)
\mathcal{G}_{II} = \frac{1}{2l_{1}^{e}l_{2}^{e}}F_{1}\left(u_{1}^{upper} - u_{1}^{lower}\right)
\mathcal{G}_{III} = \frac{1}{2l_{1}^{e}l_{2}^{e}}F_{2}\left(u_{2}^{upper} - u_{2}^{lower}\right)$$
(16)

where l_i^e is the lement length in the *i*-direction. A local crack coordinate system, x_i with i = 1, 2, 3, defines the r⁻ le-components, such that mode II (x_1 -direction) and mode III (x_2 -direction) are normal and tangent all to t' e delamination front, respectively, and mode I (x_3 -direction) is normal to mode II and III circe tons. For a straight front, like the one under study, the orientation of this local coordinate system is constant along the front and aligned with the mesh [45]. The same results are obtained using the built-in implementation available in the commercial FE code ABAQUS [43].

To evaluate the *J*-values, the interface undergoing delamination has been modeled using userdefined cohesive elements. To this end, the method presented in [10, 11] has been enhanced with the formulation for the numerical evaluation of the mode-decomposed CZ $_{\circ}$ integrals presented in Appendix A. For the purpose of comparison with VCCT, a fixed CD $_{\circ}$ is defined normal to the straight delamination front.

The mode-decomposed energy release rate distributions *e*' is the width of the specimen, from both the VCCT and the CZ J-integral extraction methods. are plot⁺ed in Figure 7. Both results are in good agreement, although there are small differences at some points. However, determining which is the most accurate is not straightforward. On the one . and, in a real specimen, a damage process zone develops ahead of a crack tip, thus increasing the compliance of the specimen. Using the VCCT approach, the development of a damage process zon of nead of the crack tip is neglected. Using cohesive elements, the development of this damage process, one is captured and therefore, the compliance of the specimen increases with respect to the con. viance of the VCCT specimen. On the other hand, the penalty stiffness of the cohesive law c.n in oduce an error into the computation of the energy release rate [46], especially when the dam $_{\text{ge p. }}$ ϵ s zone is not fully developed. However, it is worth noting that the initial stiffness that has been selected is very high to minimize this effect. In any case, the good agreement between both \neg proaches validates the methodology presented here.

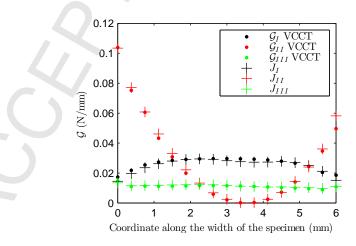


Figure 7: Comparison of the mode-components of energy release rate between VCCT and CZ J-integral extraction methods.

Furthermore, the standard formulations for VCCT require having orthogonality of the mesh with the delamination front in order to obtain accurate energy release rate con γ or ents [47]. Therefore, its application to three-dimensional FE models requires the option of being at. to move meshes that conform according to the delamination front, something which is no av ilable in commercial finite element codes [48]. Alternative solutions that enable the use of station ry meshes are presented in [49, 50]. These techniques consist of tracing a smooth virtual ^e and around the stepped front. Either way, the basic assumption of these formulations is that the noder at the delamination front will propagate along a normal vector to the current front. However, we en the delamination originates from an artificial initial defect, e.g. caused by a Teflon insert, or "hen the loading conditions change, there is a transient stage during which the shape of the crack fro. changes according to the current propagation conditions. The formulation for the evaluation of b GDD does not depend on the geometry of the crack front (which is historical information), but ther on the current displacement field. Further details are given in [39]. Thus, any variatio. in the displacements due to a change in the loading scenario is captured by the GDD cri error, at the current time. Therefore, the mode-decomposition scheme according to the GDD car be $a_{\mu} \rightarrow i$ d during transient propagation.

³¹⁴ 5. Comparison with the . EFM analytical solution of a penny-shaped crack

In this section, the fermu ation of the CZ J-integral is applied to a penny-shaped crack embedded at the centre of a ste $_1$ cylinde, of 20 mm radius, r, and 20 mm height, h (c.f. Figure 8.a). The radius of the penny-shaped c., k. a, is 5.1 mm. A shear force, Q, is applied at the center of each crack face, pointing in opp site directions, as shown in Figure 8.b. The cylinder is modeled in the commercial FE code A AQU^C [43] using C3D8I hexahedral elements. Exploiting Y-symmetry, only one half of elements [16, 11] enhanced with the CZ J-integral formulation (c.f. Appendix A for the finite element implementation). The undeformed cohesive elements are 0.32 mm wide and 0.1 mm long (tangential and radial direction to the crack front, respectively). The elastic and fracture properties used in the

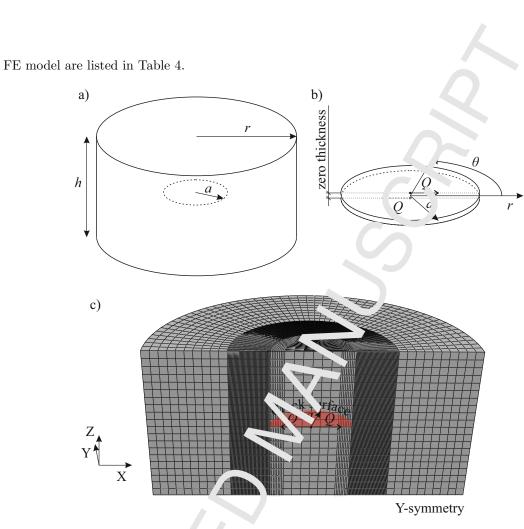


Figure 8: a) Penny-shaped crack (nbe .ded $_{\rm t}$ the center of a cylinder. b) Detail of the penny-shapped crack with the applied shear load [38]. c) r. mc .el.

Properties	5	
E: Young's modulus	210	GPa
μ : Poisson's coefficient	0.3	-
\mathcal{G}_c : fracture toughness	11	N/mm
τ_o : strength	400	MPa
K: penalty stiffness	10^{5}	N/mm^3

Table 4: E. stic ar 1 fracture properties used in the simulation study of the penny-shaped crack.

The CZ *integral* mode II and III components computed according to Equation (15) are represented in *Figure* 9 together with the LEFM analytical solution of a penny-shaped crack in an infinite domain available in [38]. The mode I component is not plotted since it is negligible under these loading conditions. The represented results have been normalized by:

$$F = \frac{\left(\frac{2Q}{(\pi a)^{3/2}}\right)^2}{E} \tag{17}$$

The energy release rates extracted using the CZ J-integral formulat on ϵ^{-1} in good agreement with those from the LEFM analytical solution [38]. However, likewise in the VCCT example presented in Section 4, the LEFM analytical solution does not take into account the development of a damage process zone ahead of the crack tip. Even though the parameters of the cohesive law have been selected such that a fair comparison with LEFM can be m. ¹e (small fracture process zone), there still exist a small discrepancy between the results of the ... or ethods. In any case, the derivation and implementation of the proposed CZ J-integral - integral - inte accuracy.

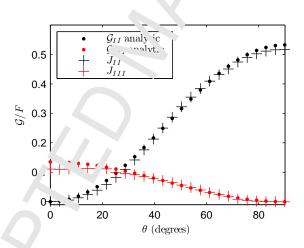


Figure 9: Comparison () the n. de-components of the energy release rate between the LEFM-based analytical solution [38] and CZ _-int/ gral method.

337 6. Application partially reinforced ELS specimen

In [39], an well-baded split (ELS) test on a symmetric run-out specimen is presented. A Teflon insert is placed et the mid-plane of the specimen and acts as an initial delamination. A pulling displacement is applied to the cracked end of the specimen causing the two specimen beams to deflect. The test rig allows the applied displacement to be maintained in the initial direction (usually the vertical direction)

by clamping the opposite end of the specimen between rollers. Consequently, the mc ement in the horizontal direction is not constrained and axial forces are avoided. Because ft' is test configuration, the specimen is subjected to large deflections. Moreover, the particularity of this kind of test is that the delamination shape changes during propagation as it approaches "he stiffened region created by bonded reinforcements on the upper and lower faces (cf. Figure 1)). The reinforcements do not span the entire width of the specimen in order to promote a curved *c* lamination. As a consequence, during propagation, both the delamination front and the crack surfaces are curved. Therefore, the partially reinforced ELS specimen is considered to be suitable to exemplify the applicability of the generalized CZ J-integral methodology for 3D curved and non-plana. delamination fronts.

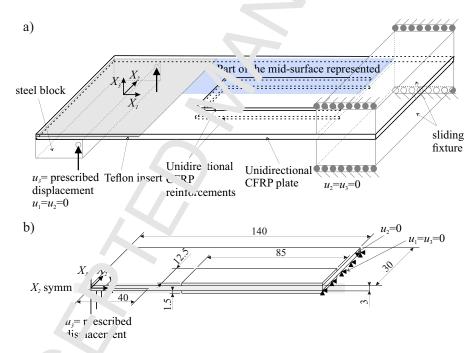


Figure 10: a) Ske ch of the partially reinforced ELS specimen [39], consisting of a CFRP plate with an initial delamination can be a Teflon insert and two CFRP reinforcements bonded to the upper and lower faces. The grey-shadowed area is the part of the mid-surface represented in figures 11, 12 and 13. b) Simplified model for FE simulation and dimensions (units in mm).

The run f ace is modeled using user-defined cohesive elements which incorporate the formulation present d in [10, 11], enhanced with the GDD criterion presented in [39] and the CZ *J*-integral formulation described in Appendix A. The undeformed cohesive elements are 0.27 mm wide, 0.23 mm long and have zero thickness. To reduce the computational resources required, only one half of

Interface properties $\mathcal{G}_{Ic} = \mathcal{G}_{IIc} = \mathcal{G}_{IIIc}$: mode-independent fracture toughness2V/mm $\tau_{Io} = \tau_{IIo} = \tau_{IIIo}$: mode-independent interlaminar strength5MPaK: penalty stiffness10 ⁵ 1/mm ³		
$\tau_{Io} = \tau_{IIo} = \tau_{IIIo}$: mode-independent interlaminar strength \sim MPa	Interface properties	
· · · · · · · · · · · · · · · · · · ·	$\mathcal{G}_{Ic} = \mathcal{G}_{IIc} = \mathcal{G}_{IIIc}$: mode-independent fracture toughness	$2 \sqrt{N/mm}$
K: penalty stiffness 10^5 V/mm^3	$\tau_{Io} = \tau_{IIo} = \tau_{IIIo}$: mode-independent interlaminar strength	Nr MPa
	K: penalty stiffness	10^{5} $1/mm^{3}$

Table 5: Fracture properties of the interface used in the simulation s udy , be ELS specimen.

the specimen is modeled by exploiting X_2 -symmetry. The elastic properties of the laminate and the fracture properties of the interface are listed in Tables 1 and ε_{c} respectively. Note that, as a simple way to check the CZ *J*-integral implementation, the fracture to the sis set to be mode-independent $(\mathcal{G}_c = \mathcal{G}_{Ic} = \mathcal{G}_{IIc} = \mathcal{G}_{IIIc} = 2 \text{ N/mm})$ to ensure a constant *J* value ($J = \mathcal{G}_c$) during static crack propagation. Thus, the sum of the three mode-decompored CZ *J*-integrals in Equation (15) must be constant and equal to 2 N/mm at every integration contour, regardless of the loading mode. In the following figures, only the blue-shadowed area of the indistribution in Figure 10 is represented.

The historical evolution of the 0.5-valued energy based damage isoline is plotted in Figure 11.a. The energy-based damage, \mathcal{D}^e , distribution is projected onto the deformed mid-surface (cf. Figure 11.c) for a prescribed end displacement of 21. mm. Note that a large fracture process zone is developed (the maximum length of the CZ is app. xi nately 20 mm). The GDD distribution within the CZ is represented in Figure 12. As n_{x} ti ned in Appendix A, the CZ J-integral can be evaluated at any point within the CZ and, there, re, infinite integration paths can be tracked. For illustrative purposes, only a few selected integration paths are plotted on top of the GDD distribution. Note that the trajectory of the integration ratio ratio is established according to the GDD. Thus, since the $\frac{\omega_{tot}}{\mathcal{G}_c}$ isolines are not parallel, the ... 'gra ion paths are curved lines throughout the CZ.

The total *J*-value is valuated at each of the 30,000 integration points forming the CZ. The result is represented in Figure 13.a. The step length h^k used is 0.3 mm (1.3 times the element length), where the superscription of the projection on the cohesive interface mid-surface (see Appendix A for furth h^{-1} escription of h^k). Note that the *J*-distribution is constant and equal to the fracture toughness, which, during static propagation and for any mode mixity, amounts to 2 N/mm. The total *J*-value computed is equal to the fracture toughness at all the integration points within the cohesive

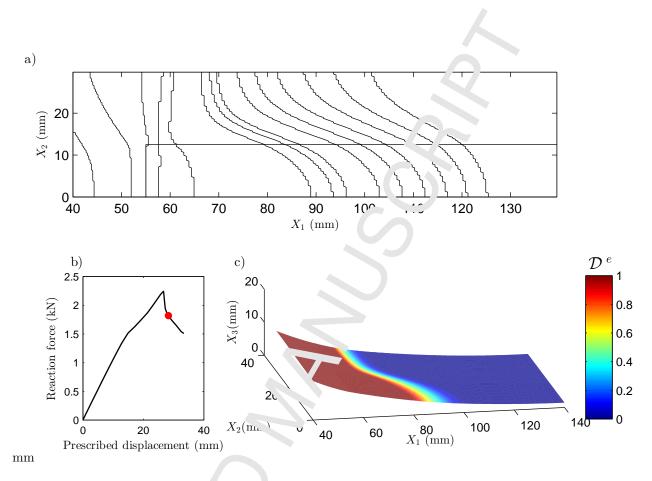


Figure 11: a) Historical evolution of the 0.5-val led energy-based damage isoline extracted at the integration points. b) Reaction force vs prescril ed disp. ement curve with the current loading state highlighted in red. c) Energy-damage projected onto the deformed mid-surface at the current loading stated marked in (b).

zone with a maximum error of 3.7% (cf. Figure 13.a). By reducing $*h^k$, more accurate results may be obtained. However, for such a large CZ, the computational cost increases significantly with the number of segments j which, the integration paths are discretized.

The decompositio. If the CZ J-integral into modes, computed according to Equation (15), is also represented in Figure 13. The mode II and III components of the CZ J-integral are predominant, while mode 1 slightly appears at a small region close to the specimen's edge (cf. Figure 13.b). The contribution γ the J-value of the tangent quantities to the mid-surface is decomposed into modes II and III according to the GDD. The bonded reinforcements cause the loading state to be uneven throughout the specimen's width, leading to a curved crack, so that the GDD amounts to 60° with respect to the X_1 at the zones with the highest delamination front curvature. Due to the test configuration, the

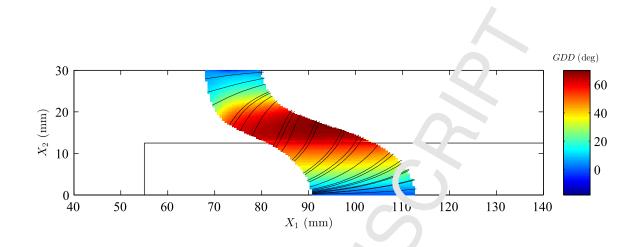


Figure 12: Growth driving direction (GDD) distribution along the con-sive zone and a few selected integration paths (black solid lines) plotted on top of it. The current load. * state is marked in Figure 11.b.

maximum interlaminar shear stress is applied in the global X_1 -direction. For straight cracks where the GDD is aligned with the X_1 -direction, the shear one ponent would be pure mode II. However, in the studied case with a curved delamination from the maximum contribution of the external loading to the mode III CZ *J*-integral is at the regreen with the GDD differs most from the X_1 -direction (cf. Figures 13.c and 13.d).

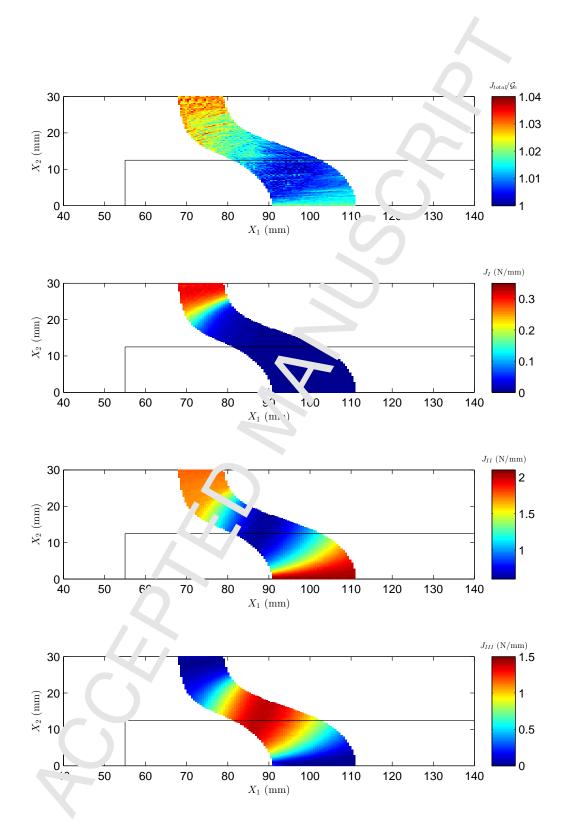


Figure 13: Distribution of a) J_{total}/\mathcal{G}_c (where $J_{total} = J_I + J_{II} + J_{III}$ and $\mathcal{G}_c=2$ N/mm), b) J_I , c) J_{II} and d) J_{III} within the cohesive zone at current loading state marked in Figure 11.b.

392 7. Conclusions

A novel methodology for calculating the mode-decomposed *J*-integrals in the redimensional delamination simulation using a cohesive zone model approach is presented. The methodology incorporates the growth driving direction criterion, recently developed by the authors, or track the integration paths and to determine the local directions of mode I, II and III components. The generality of the formulation makes it applicable to curved fronts with non-planar d tam; in ion interfaces and large fracture process zones. The application of the described methodolog, results in curved integration paths.

The calculation of the *J*-integral is based on dividing the *d*-dimination interface into elemental thickness slices so that the *J*-value of each slice is unique. The curvature of such slices is defined according to the growth driving direction. Since the growth driving direction is mesh independent, the definition of the slices is not affected by the resh sites.

By applying the formulation presented here, c global measure of the energy release rate in three-dimensional structures modeled using a cohesive zone model approach can be obtained. To the authors knowledge, this has not been prevously a dressed. Furthermore, the energy release rate can be decomposed into mode I, II and III omponents. The decomposition of the shear component of the energy release rate into mode II and I. to date, has only been addressed under the assumption of elastic fracture mechanics. In aquition, the new formulation enables a global measure of the mode mixity to be obtained, error ming the limitation of the current 3D cohesive zone model formulations where the mode mix ty i only obtained at integration point level in terms of opening displacements. The limitatic _ of the presented formulation are related to the use of cohesive zone models, and therefore, the ci-ck is onfined to propagate within the interface between layers. The possibility of crack migr ting to nother interface is not accounted for.

Bes^{: 1}⁴¹⁴ Bes^{: 1}⁴¹⁴ be uncertain the immediate applications of the formulation, the authors believe that more applications ⁴¹⁵ will be uncertain future research. The CZ *J*-integral presented here is a decisive contribution to ⁴¹⁶ fracture mechanics-based procedures in a cohesive zone model framework, which will allow the design of ⁴¹⁷ lighter and more reliable structures. In addition, a direct application of the CZ *J*-integral formulation

is its implementation in combination with existing fatigue simulation methods formula ed in a CZM approach that rely on mode-dependent Paris law' like expressions. Thus, the mode-decomposed CZ *J*integral formulation developed becomes a new solution for extracting mode-decomposed energy release rates of real complex three-dimensional structures.

422 8. Acknowledgements

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Appendix A. Discretization with the FE met. d

⁴²⁶ Using Equation (15), the mode-decomposed CZ \sim -integrals, which may vary for every slice, can be ⁴²⁷ evaluated everywhere within the CZ. Moreover, \sim ny point within the CZ belongs to a single slice, i.e. ⁴²⁸ to a single integration path. The integration path is are defined according to the local GDD. Therefore, ⁴²⁹ one can randomly select any location of the \Z and, by means of the GDD, identify the tangent to the ⁴³⁰ integration pathline at that point in order to move, either forward or backward, along the integration ⁴³¹ path. The mode-decomposed CZ \sim int grals corresponding to such slice are obtained when the path ⁴³² is tracked in its entirety.

The procedure for $t^{1} + ev_{7}$.uation of the mode-decomposed CZ J-integral of Equation (15) is shown in Figure A.15 and described in the following. Consider a point, P^k , belonging to the CZ. In order to assess the mode-decomposed of CZ J-integrals at the slice which the point P^k belongs to, the numerical integration of E-uation (14) is performed along the integration path, defined as tangent to the local GDD direc ion and limited by vanishing stress conditions at both ends (cf. Figure 3). In the general case, the initial point P^k is not located at one end of the integration path, i.e. point P^k is located in the middle of the CZ. In this case, the path will be tracked from P^k in the GDD (Loop 1 in Figure A.15) and in the opposite direction to the GDD (Loop 2 in Figure A.15). In other words, in the positive GDD until vanishing elastic stress is reached (point 2 in Figure 3), while in the negative GDD

until the intersection with the 1-valued energy-based damage isoline, where the cohest ve stress also equals zero (point 1 in Figure 3). The condition for vanishing cohesive stress as yeads:

$$\mu < tol$$
 (A.1)

where μ is the norm of the cohesive stresses and *tol* is a user-defined three hold close to zero.

To move along the integration path, the following procedule is p_1 lied. Starting from P^k , the next point along the integration path is established by moving in $straight line a *h^k$ -length step further in the local GDD, which is tangent to the cohesive intertain mid-surface, \bar{S}_{coh} , at P^k . Then, a new point, P^{k+1} , in the space is found. Nevertheless, P^{k+1} is not necessarily placed on the mid-surface, \bar{S}_{coh} . This becomes evident when \bar{S}_{coh} is highle in planar (cf. Figure A.14). Thus, the real next point constituting the integration path, P^{k+1} , is found by projecting P^{k+1} on \bar{S}_{coh} in the normal x_3 -direction of point P^k .

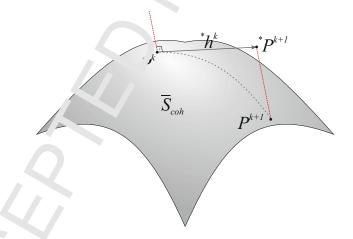


Figure A.14: Point P^k is a point on the integration path of a curved cohesive interface, \bar{S}_{coh} . The following point on the integration path, P^{k+1} , is found by projecting point $*P^{k+1}$ along the normal direction to the interface at pr' + P. Found $*P^{k+1}$ is at an *h distance from P^k in the tangential GDD.

The integrands in Equation (15), σ_{ii} and $\frac{\partial \delta_i}{\partial x_1}$, are evaluated at every point P^k along the integration path. σ_{ii} a \circ the components of the cohesive stress tensor expressed according to the local Cartesian coordinate system. On the other hand, the derivative of the displacement jumps, δ_i , with respect to the local Cartesian coordinate aligned with the GDD, x_1 , is addressed in the following. X_j is the

⁴⁵⁶ Cartesian reference system, x_i is the local Cartesian coordinate system and R_{ii} is the ι ansformation ⁴⁵⁷ tensor which relates the global to the local coordinate system. The derivation of the rotation matrix, ⁴⁵⁸ R_{ij} , with respect to the coordinate x_1 can be approximated to zero by assuming that the curvature ⁴⁵⁹ of the interface within the integration subinterval is small. This is achieved by setting an $*h^k$ -length ⁴⁶⁰ step similar to the element length. Moreover, its derivation would include the complexity of the ⁴⁶¹ formulation without a substantial improvement in the accuration of the solution. Thus, by assuming ⁴⁶² that the derivative of R_{ij} with respect to x_1 can be omitted, the derivative $\frac{\partial \delta_i}{\partial x_1}$ reads:

$$\frac{\partial \delta_i}{\partial x_1} = R_{ij} \frac{\partial \Lambda_{j}}{\partial x_1} \mathcal{O}_n \tag{A.2}$$

where M_{jm} is the transformation matrix that i lines the global displacement jump with the nodal global displacement, Q_m . The size of Q_m is \dots number of degrees of freedom of the element (in the case of 8-noded cohesive elements, m = 1, 24). The derivative of the transformation matrix, M_{jm} , with respect to the local coordinate, x_1 , is obtained by applying the chain rule:

$$\frac{\partial \hat{I}_{jm}}{\partial x_1} = \frac{\partial M_{jm}}{\partial \eta_\alpha} \frac{\partial \eta_\alpha}{\partial x_1}$$
(A.3)

The first partial derivative is the right hand side of Equation (A.3) is the variation of the transformation matrix, M_{jm} , with the isoparametric coordinates of the cohesive element formulation, η_{α} (α =1,2):

$$\frac{\partial M_{jm}}{\partial \eta_{\alpha}} = \left[-\frac{\partial N_{jk}}{\partial \eta_{\alpha}}, \ \frac{\partial N_{jk}}{\partial \eta_{\alpha}} \right] \tag{A.4}$$

where N_{jk} is the slape function matrix and the subscript k runs from 1 to the number of degrees of freedon 100_1 continues of the top and bottom surface of the cohesive element. In the case of an 8-noded element, k = 1...12. In [10, 11], the material coordinates and the displacement fields are interpolated within the domain of the interface element using isoparametric bilinear shape functions:

$$L_{1} = \frac{1}{2} (1 - \eta_{1}) (1 - \eta_{2}); \qquad L_{2} = \frac{1}{2} (1 + \eta_{1}) (1 - \eta_{2})$$

$$L_{3} = \frac{1}{2} (1 + \eta_{1}) (1 + \eta_{2}); \qquad L_{4} = \frac{1}{2} (1 - \eta_{1}) (1 + \eta_{2})$$
(A.5)

474 organized in N_{jk} as follows:

$$N_{jk} = \begin{bmatrix} L_1 & 0 & 0 & L_2 & 0 & 0 & L_3 & 0 & 0 & J_4 & 0 & 0 \\ 0 & L_1 & 0 & 0 & L_2 & 0 & 0 & L_3 & 0 & L_4 & 0 \\ 0 & 0 & L_1 & 0 & 0 & L_2 & 0 & 0 & L_3 & 0 & 0 & L_4 \end{bmatrix}$$
(A.6)

where the local isoparametric coordinates, η_1 and η_2 , i. eq from -1 to 1 over the element domain.

The derivatives $\frac{\partial \eta_{\alpha}}{\partial x_1}$ are the inverse of the der \Im ives of the local coordinate, x_1 , with respect to the isoparametric coordinates, η_{α} , defined as:

$$\frac{\partial x_1}{\partial \eta_{\alpha}} = \frac{1}{c_{1j}} \frac{\partial N_{jk}}{2} \left(C_k^+ + C_k^- + Q_k^+ + Q_k^- \right)$$
(A.7)

where C_k^- and C_k^+ are the global coor linates of the nodes at the lower and upper surfaces, and Q_k^- and Q_k^+ are the nodal displacements, relaxing to the global coordinates, of the lower and upper surfaces.

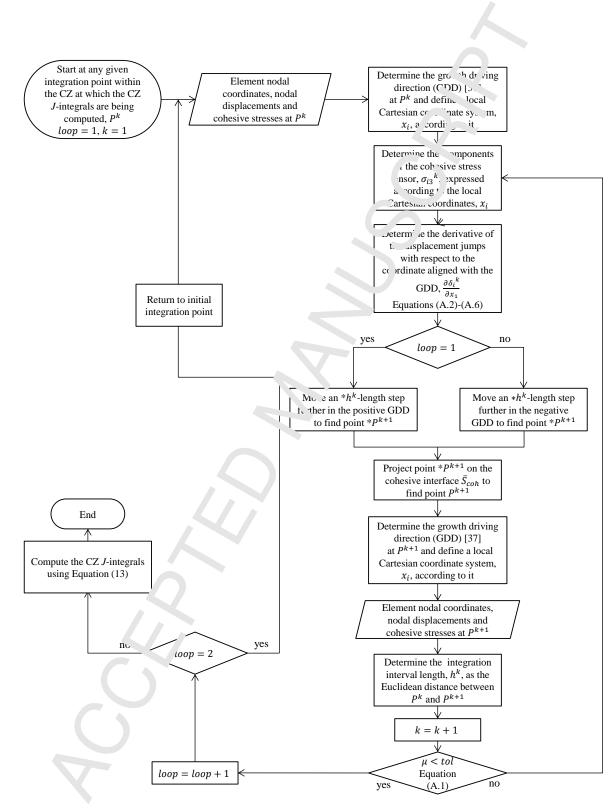


Figure A.15: Flow chart of the calculation of the CZ J-integrals at a given point within the cohesive zone discretized with the FE method.

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