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- 1. 3D J-integral applicable to problems involving large fracture process zones.
- 2. A mode I-II-III decomposed J-integral for large fracture process zones.
- 3. Evaluation of the J-integral using the information from the cohesive zone model.
- 4. Efficient implementation of the mode-decomposed J-integral using cohesive elements.
- 5. Application of the method to a 3D structure under mode I, II and III loading.

 An evaluation of mode-decomposed energy release rates for arbitrarily shaped delamination fronts using cohesive ϵ energy release rates for arbitrarily

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Abstract

Computing mode-decomposed energy release rates in arbita, vily shaped delaminations involving large fracture process zones has not been previously involving. The J-integral is a suitable method for calculating this, because its domain-independence can be employed to reduce the integration domain to a cohesive interface, and reduce it to a line integral. However, the existing formulations for the evaluation of the mode-decomposed J-inc mass rely on the assumption of negligible fracture process zones. In this work, a method for the computation of the mode-decomposed J-integrals in three-dimensional problems involving in a plicable to curved fronts with non-planar crack faces. A growth driving direction criterian, which takes into account the loading state at each point, is used to render the integration passes and to decompose the J-integral into loading modes. This results in curved and non-planar in eg. ution paths crossing the cohesive zone. Furthermore, its implementation into the finite element fram work is also addressed. The formulation is validated against virtual crack closure technique (TOT) and linear elastic fracture mechanics (LEFM)-based analytical solutions and the signific note and generality of the formulation are demonstrated with crack propagation in a three-dimental composite structure.

Keywords:

Delamin attor of both Cohesive zone model, Finite element analysis, Energy Release Rate, 3D

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J-Integral

1. Introduction

Laminated composite materials are built by stacking plies with different material and reinforcement orientations, e.g. fiber reinforced polymers. During service, exce sive in erlaminar stresses can lead to a loss of cohesion between constituent layers. This failure mechanism is known as delamination, and it is one of the most common cause of failure in structure. .nade of layered materials. Therefore, analyzing the onset and growth of delamination is essential for any mechanical application of laminated composites. In this regard, the finite element (FE) me had has become an indispensable tool for designing layered composite structures and predicting beir service life. The most common methods for predicting intally initial relations and the divided into two main approaches: Methods based purely on fracture nechanics and methods based on the concept of the cohesive zone model (CZM) [1, 2]; the latter of which combines the framework of fracture mechanics and damage mechanics. In the fracture mechanics approach, 'su aly a local Griffith's criterion [3] is used to predict delam-ination growth, i.e., the energy 1^{-1} er s er ϵ , ϵ , is compared to the interlaminar fracture toughness, \mathcal{G}_{ϵ} . Two of the most common et a ction methods for the energy release rate (also called the crack exten-sion force) rely either on the VCCT [4] or the J-integral [5]. Then, applying Griffith's criterion, crack propagation occurs at the points where $\mathcal{G} \geq \mathcal{G}_c$. This local energy balance criterion implies a negligible fracture process zone. Con ersely, CZMs can capture the fracture energy dissipation mechanisms of quasi-brittle materials, uch as the formation of micro cracks ahead of the crack tip before complete separation councerrack faces occurs. Therefore, the CZM approach is a suitable means of predicting crack propas tion when a non-negligible fracture process zone is present. The strain singularity at the tip o. a sharp crack is removed by accounting for a cohesive zone (CZ), where the material un-dergoes degradation until complete decohesion. The mechanical behavior of the interface is modeled

by means of a damage variable, which is a measure of the degradation of the mechanical properties

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of the material ahead of the crack tip. When the damage variable reaches its maximun, value, a new crack surface is created. Moreover, CZMs are particularly suited for simulating interlaminar cracks in laminated structures because the delamination is confined to propagate between two adjacent plies. Thus, when a progressive delamination simulation is solved using an Lara alysis, the potential failure surfaces are known in advance, and the cohesive elements can be fficient'y located. Under static loading conditions, existing CZMs [6-12] do not require the energy release rate to be computed in order to simulate crack growth. However, some of the recently published methods for simulating fatigue-driven delamination based on CZM [13-18] lim the rate of the local fatigue damage with any variant of the Paris' law [19]. The Paris' law-n' expressions relate the crack growth rate with a power law function of the loading level in term of a fracture mechanics parameter [13, 20], usually the stress intensity factor, K, or the energy α ease rate, \mathcal{G} , where only the latter is relevant for a CZM. Therefore, computing the energy releas r. a is required in order to integrate the rate of the local fatigue damage. In this regard, the J-in, σ ral directly equates to \mathcal{G} [21]. In fact, the benchmark study of the simulation methods for 'atigu driven delamination using a CZM approach presented in [18] showed a better performance for the methods using the J-integral as the means of extracting the energy release rate. The path-independence fully two-dimensional J-integral makes it very attractive in practice, since it avoids the need for actura e computations on the stress field at the crack tip; something which is hard to deal with ir an Fr. framework. For this reason, considerable effort has been devoted to extending the applicative ity if the J-integral to three-dimensional (3D) domains [22–32]. The published extensions of the J-integral for its evaluation in three-dimensional problems, where the crack extension force may change along the crack front, commonly employ two approaches. The first is a point-wise evaluation of the integral on a cross-section normal to the crack front, resulting in the combination of a cont v integral and a surface integral defined over the area enclosed by the contour. See [30]

for a detailed description. Computing the surface integral requires accurately calculating the field

quantities at the crack tip. For this reason, the boundary element method is commonly used [27, 30].

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The second approach is the equivalent domain integral over a finite volume surrounding the crack front [25, 26, 33]. With this method, capturing the singular field near the rak tip is not required which is why it is usually applied in a FEM framework. Regardless, the a plicability of most of these J-integral extensions to three-dimensional domains is restricted ' γ ertain assumptions such as plane-strain/stress, i.e., at the vicinity of the crack tip, or plane crack. By employing curvilinear coordinates, Eriksson [34] and Fernlund et al. [35] obtained generalized expressions applicable to curved cracks with non-planar crack surfaces. In [34], a volume-independent integral expression for evaluating the crack extension force is derived from the principle of virtual work. In [35], the decrease of the potential energy with crack extension is employed to obtain a general path-area independent J-integral expression for non-planar cracks with curved arck fronts. In both cases, the fracture process zone is considered negligible and the mode-decom, of tion is not addressed.

Delamination propagation can be described through a combination of the three basic fracture modes (Modes I, II and III) [36], and the fracture resistance of the interface, under both static and fatigue loading, highly depends on the mode mixity conditions. Consequently, the delamination models available in the literature [13, 20, ?7] are 'as 3d on a mode-decomposed definition of the load, expressed in terms of the energy release 1.5 (G_I G_{II} and G_{III}). In this regard, the decomposition of the Jintegral into fracture modes as 3 tool for extracting energy release rates, becomes necessary.

In this work, a new place lure to numerically evaluate mode-decomposed J-integrals in a 3D body undergoing delamination is presented. The method is applicable to curved crack fronts with non-planar crack surfaces. Morever, he method enables, for the first time, the application of the J-integral in 3D problems involving large fracture process zones. In addition, in contrast to current cohesive models where the mode mixity is evaluated locally (point-wise) using the interface separation, the presented J-recorrect formulation enables defining the mode mixity parameter as a function of the mode decomposed G_I , G_{II} and G_{III} (global measures). This is of crucial importance to improve the accuracy of the simulation of delamination propagation under quasi-static and fatigue loading.

The formulation is derived from the general expression of the J-integral for 3D curved delaminations

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with non-planar surfaces expressed in terms of curvilinear coordinates [35], which reas on LEFM. Its application to cohesive interfaces is addressed in Section 2, while its and mentation in an FE framework is presented in Section 3 and Appendix A. In Section 4, the to, rulation is applied to a moment-loaded double-cantilevered-beam (DCB) and the mixed-mc 'a -components are compared to the mode-decomposed energy release rates obtained from VCC Γ. In Caction 5, the formulation is applied to an embedded penny-shaped crack in a steel cylinder and the determined mode-components are compared to and validated against an analytical LEFM-based olution [38]. In Section 6, the formulation is used to compute the J-integral components of a partially reinforced end-loaded split (ELS) specimen with a non-straight crack front and non-p. Par crack interface. Finally, the conclusions on this work are presented.

2. Formulation of mode-decomposed energy release rates

 88 In this section, the formulation of the mode 4 ecomposed energy release rates in 3D delaminations,

modeled using a cohesive zone model approach, is presented. The point of departure is the generalized

J-integral for non-planar curved cracks of Lined by Fernlund et al. [35].

2.1. Assessment of the energy release rate by means of the J-integral formulation in curvilinear coor-

92 dinates

Consider an elastic body (cf. Figure 1), with a crack, subjected to prescribed tractions, T, and displacements, u, at the parts of its boundary surface (Note that T and u are physical entities that are not yet described in any particular coordinate basis). In a general three dimensional domain, both the crack surfaces and the crack front may be curved. Let θ^i , i = 1, 2, 3, be an orthogonal curvilinear coordinate a stem in ith origin at a given point P along the crack front. This local coordinate system is oriented auch that, at point P, θ^3 is normal to the crack surface, θ^2 is the coordinate along the crack front and θ^1 is the direction of crack propagation, which is always tangent to the crack surface and perpendicular to θ^2 and θ^3 .

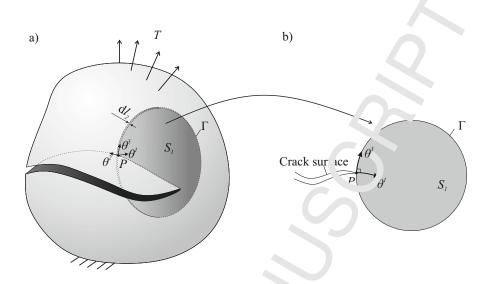


Figure 1: a) Three-dimensional body undergoing a delaminate , with curved front and non-planar crack surfaces. b) The integration domain is a slice of infinites. At thickness, dl_2 .

Let us focus on a thin slice of elemental thickney, dl_2 , of the cracked body, which contains P (cf. Figure 1). Note that an infinitesimal length segn ent, dl_i along a curvilinear axis, θ^i is given by:

$$dl_i = \sqrt{g_{ii}} d\theta^i \tag{1}$$

where g_{ij} is the covariant metric ten or. In the absence of body forces, the change in potential energy, Π , per unit of newly created rack area is [35]:

where dA is the element c ack area extension, V is the volume of the slice, S is the surface surrounding

$$-\frac{\mathrm{d}\Pi}{\mathrm{d}A} = -\int_{V} \frac{\mathrm{d}W}{\mathrm{d}A} \mathrm{d}V + \int_{S} T^{i} \frac{\mathrm{d}u_{i}}{\mathrm{d}A} \mathrm{d}S \tag{2}$$

V, W is the str in ener y density, T^i are the contravariant components of the traction vector and u_i are the cov riant emponents of the displacement vector. The infinite into a surface S_1 , defined by $^2 = 0$ (d $l_2 \rightarrow 0$). Then, by applying Green's theorem, and under the assumption of small deformations, elastic material behavior, symmetry of the stress tensor and equilibrium conditions, the decrease in potential energy per unit area extension is expressed, in [35], as a contour integral and an

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area integral on the surface S_1 :

$$J = -\frac{\mathrm{d}\Pi}{\mathrm{d}A} = \frac{1}{\sqrt{g_{11}}} \oint_{\Gamma} \left(W n_1 - T^i \frac{\partial u_i}{\partial \theta^1} \right) \mathrm{d}\Gamma - \frac{1}{\sqrt{g_{11}g_{22}}} \int_{S_1} \frac{\partial}{\partial \sigma} \left(\sigma^{i2} \frac{\partial \omega_i}{\partial \theta^1} \right) \mathrm{d}S$$
(3)

on Γ . Note that in [35], the curvilinear coordinate system is rotated 90° round the θ^1 -coordinate.

The J-integral is equivalent to the energy release rate, ζ , for an elastic material response. In a three-dimensional body, the energy release rate may vary along the crack front. Therefore, in order to assess the delamination extension force in three-dimensional problems, it is customary to compute the point-wise value of J as a function of the crack from Γ points, P.

where Γ is the contour enclosing S_1 in the clockwise direction and n in the cutward unit normal vector

2.2. Application to cohesive interfaces

Unlike LEFM, the CZM relies on the existence of a band of material ahead of the crack tip (known as the cohesive zone (CZ)), where the material cohaves nonlinearly [1, 2]. In the CZ, a cohesive traction distribution acts on the separating surfaces, thus avoiding stress singularities at the tip of sharp cracks.

The constitutive law that relates the contract vectractions to the displacement jumps at the interface is governed by a scalar damage variable. The damage variable evolves monotonically with time to ensure irreversibility. To guarante the proper energy dissipation under mixed-mode conditions, in [11] the cohesive law is formulated in a one-dimensional space, where the equivalent mixed-mode traction, μ , is related to the norm of the displacement jump, λ . Thus, the equivalent one-dimensional displacement jump, λ , is defined as:

$$\lambda = \sqrt{\left(\delta_1\right)^2 + \left(\delta_2\right)^2 + \left(\langle \delta_3 \rangle\right)^2} \tag{4}$$

and the equ. . at one-dimensional interface traction, μ , is related to λ as follows:

$$\mu = (1 - \mathcal{D}^K) K\lambda \tag{5}$$

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where $\mathcal{D}^K \in [0,1]$ is a scalar damage parameter degrading the constitutive tangent stifn. ss, K, and $\langle \rangle$ is the Macaulay bracket ensuring that negative normal opening (interpenet. 'io' of crack faces) does not affect damage development.

A sketch of the bilinear cohesive law used in [11] is represented in Figure 2. An energy-based damage variable, \mathcal{D}^e , is introduced as the ratio of specific dissipate 1 energy due to fracture, ω_d (Figure 2.b), and the fracture toughness, \mathcal{G}_c (Figure 2.a). Thus, \mathcal{D}^e rar $_{5-3}$ from $_{9}$ to 1, and can be understood as the degree of crack development, taking a value of 0 if the degrad tion process is yet to start, and a value of 1 if the crack is fully developed.

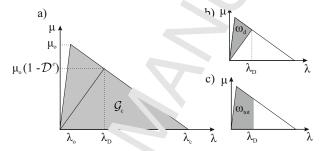


Figure 2: Equivalent one-dimensional cohesive law. The shadowed area in a) represents the fracture toughness, \mathcal{G}_c , in b), the specific dissipated energy, ω_d and 'n c), the total specific work, ω_{tot} , for a given state of damage.

The constitutive law is formed by an initial elastic region, before damage initiation, and a softening region. The onset and proparation of delamination are limited by the onset mixed-mode displacement jump, λ_c , and the critical mixed-mode displacement jump, λ_c , such that the applicability of the energy-based damage variety (e, \mathcal{D}^e) , is restricted to:

$$\begin{cases}
\mathcal{D}^e = 0 & \text{for} \quad \lambda_{\mathcal{D}} \leq \lambda_o \\
\mathcal{D}^e = \frac{\omega_d}{\mathcal{G}_c} & \text{for} \quad \lambda_o \leq \lambda_{\mathcal{D}} \leq \lambda_c
\end{cases} \tag{6}$$

$$\mathcal{D}^e = 1 & \text{for} \quad \lambda_{\mathcal{D}} \geq \lambda_c$$

where $\lambda_{\mathcal{L}}$ is the mixed-mode displacement jump associated to the current damage state.

When applied to delamination modeling in laminated composite materials, the cohesive behavior is lumped into the interface between subsequent plies. In [35], it is demonstrated that the *J*-integral

of Equation (3), generalized in terms of curvilinear coordinates for cracks with cur ed front and non-planar crack surfaces, is path-area-independent. Then, for the measurement of the delamination extension force in 3D laminated structures modeled using a CZM approach the path-area-independence of Equation (3) can be employed to reduce the contour Γ to the cohesive interface (cf. Figure 3), similar to what is done with the two-dimensional form of the J-integral [5]. The refore, because of the zero-thickness of the cohesive interface, and taking into account the three pening displacements are very small, the differentials $n_1 d\Gamma \approx d\theta^3$ and dS in Equation (3) vanish. Thus, Equation (3) is reduced to:

$$J = -\frac{1}{\sqrt{g_{11}}} \int_{\Gamma} \left(T^{i} \frac{\partial u_{i}}{\partial \theta^{i}} \right) d\Gamma \tag{7}$$

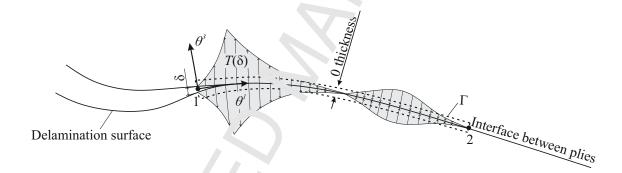


Figure 3: The integration path, (deshed line), is reduced to the zero-thickness cohesive interface.

Let σ^{ij} be the contravariant α nponents of the cohesive stress tensor. Then, the contravariant traction vector at the carek aces is given by:

$$T^i = \sigma^{ij} n_i \tag{8}$$

where n_j is one outward unit normal vector on the contour Γ , i.e., on the crack surfaces. Thus, n_j vanishes for $\zeta \neq 3$ and Equation (7) reads:

$$J = -\frac{1}{\sqrt{g_{11}}} \int_{\Gamma} \left(\sigma^{i3} \frac{\partial u_i^+}{\partial \theta^1} + \sigma^{i3} \frac{\partial u_i^-}{\partial \theta^1} \right) d\theta^1$$
 (9)

where u^+ and u^- are the displacements at the upper (+) and lower (-) crack surfaces, respectively.

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Finally, by introducing the displacement jump as the separation of two initially coinciding points on
the interface, defined as:

$$\delta_i = \left(u_i^+ - u_i^-\right) \tag{10}$$

the curvilinear CZ J-integral, when applied to cohesive interfaces, can be expressed as:

$$J = -\frac{1}{\sqrt{g_{11}}} \int_{CZ} \left(\sigma^{i3} \frac{\partial \hat{c}}{\partial \theta^1} \right) d\theta^1 \tag{11}$$

Observe, in Figure 3, that the integration path is the entire CZ so that all the cohesive stresses contribute to the CZ *J*-integral. Further details on the integration path shape and limits in 3D applications are provided in Section 2.3.

As demonstrated in Section 2.2, the integration domain of the curvilinear CZ J-integral applied

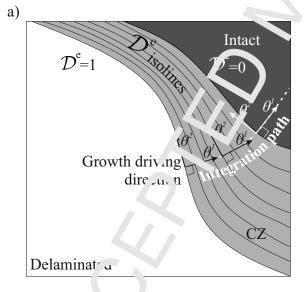
163 2.3. Integration paths

to cohesive interfaces is a slice of in. Titesim 1 thickness, dl_2 , lumped into the delamination interface. Thus, the integration domain is edu ed to a path contained in the delamination interface that follows the direction of crack proparation, θ . In order to compute the J-distribution in three-dimensional structures, the interface can be divided into infinite slices. Obviously, the J-value of each slice is unique and is obtained . 'n the integration path is covered in its entirety, i.e., going through the entire cohesive zone from the completely damaged zone (point 1 in Figure 3, with zero cohesive stress) to the end of the zone in elastic regime (point 2 in Figure 3, with zero cohesive stress). In LEFM, the remarkable gation direction, θ^1 , is assumed to be the normal to the crack front at the point P, where t e crack front is the line separating the damaged and undamaged parts (cf. Figure 4.b). Howeve time 'finition of the propagation direction as the normal to the crack front does not apply for CZM, due to the existence of a cohesive zone of variable length. The authors have recently introduced the concept of the growth driving direction (GDD) for CZM [39], as an analog to the crack propagation direction in LEFM. The GDD is defined as the gradient vector field of the scalar energy-based damage,

 \mathcal{D}^e , with respect to the coordinates tangent to the cohesive interface mid-surface:

$$GDD = -\nabla \mathcal{D}^e \tag{12}$$

Thus, the GDD is normal to the energy-based damage, \mathcal{D}^e , isoling (cf. Figure 4.a) and it converges with the normal to the crack front in LEFM (cf. Figure 4.b) in the limiting case where the length of the CZ approaches zero. Therefore, by making use of the criter in presented in [39], θ^1 can be defined according to the GDD. In this way, the integration raths, defined along the θ^1 -coordinate, never intersect and the three-dimensional structure can be and ratiod as the aggregation of infinite individual slices of infinitesimal thickness which contains a chack propagating in the GDD. It is worth mentioning that the damage isolines may not be realled along the CZ, leading to slices with double curvature if, in addition, the cohesive interfact mides are rather than the cohesive interfact mides are rather than an addition.



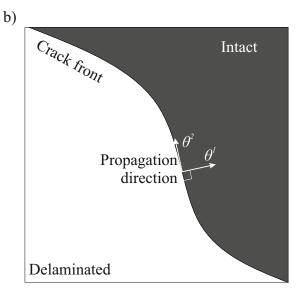


Figure 4: a) The game direction (GDD) is assumed to be the normal direction to the energy-based damage isolities in the CZM framework. The integration paths are tangent to the local GDD direction. b) The propagation direction is assumed to be the normal direction to the crack front in the LEFM framework.

It is not d that, to compute the *J*-value in cohesive interfaces using Equation (11), the contribution of the stress, σ^{i3} , and displacement jump slope in the GDD direction, $\frac{\partial \delta_i}{\partial \theta^1}$, in the elastic regime is needed. However, the criterion in Equation (12) for identifying the GDD, based on the negative

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gradient of the energy-based damage, \mathcal{D}^e , is only meaningful for $\mathcal{D}^e \in]0,1[$ (see Equation (6)). Therefore, a new criterion to identify the GDD in the elastic regime must be used. In this regard, another
criterion, which is also active before the initiation of the degradation process, a proposed in [39]:

$$GDD = -\nabla \left(\frac{\omega_{tot}}{\mathcal{G}_c}\right) \tag{13}$$

where $\frac{\omega_{tot}}{G_c}$ is the ratio between the total specific work (cf. Fig. re 2), and the fracture toughness. Both the conservative and the non-conservative work are computed in unis criterion. This implies that as soon as two initially coinciding points separate from each other $(\lambda > 0)$, some elastic energy is stored which makes this criterion active before damage of the damage is initiated, both criteria lead to the same GDD solution.

2.4. Mode-decomposition of the CZ J-integral j. r. application to cohesive interfaces

A crack can grow under a combination of the loading modes [36]: the opening mode (mode I), the sliding mode (mode II) and the tearing mode (mode III). Mode I is defined as normal to the cohesive interface mid-surface, mode II, to gent to the mid-surface in the propagation direction and mode III, tangent to the mid-surface and perion of ular to mode II. In this work, the crack propagation direction is defined as the GDD (cf. pectron 2.3). This implies that the mode II direction is also defined as the GDD, and the mode III direction is defined as the direction perpendicular to the mode I and mode II direction.

For the mode decomposition of the *J*-integral, the integrands in Equation (11) must be decomposed according to the local basis vectors, aligned with the three loading modes directions. Thus, θ^1 is locally coincident with the GDD (i.e. tangent to the mid-surface), θ^3 is normal to the mid-surface, and θ^2 is normal to ψ^i and θ^3 . Moreover, since θ^i are orthogonal curvilinear coordinates, the local covariant and contrataint basis vectors are collinear.

At an interface modeled using a CZM approach, only three uncoupled components of cohesive stresses (σ^{13} , σ^{23} and σ^{33}) result from the displacements jumps between crack faces (δ_1 , δ_2 , δ_3). The

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quantities σ^{13} and $\frac{\partial \delta_1}{\partial \theta_1}$ contribute to mode II, σ^{23} and $\frac{\partial \delta_2}{\partial \theta_1}$, to mode III, and σ^{33} and $\frac{\partial \delta_3}{\partial \theta_1}$, to mode III, and σ^{33} and $\frac{\partial \delta_3}{\partial \theta_1}$, to mode II crack loading. Hence, the mode-decomposed CZ *J*-integrals are defined for doing to the local θ^i coordinate system such that the terms with i=3 are attributed to Mode I, i=1, to Mode III and i=2, to Mode III:

$$J_{II} = -\frac{1}{\sqrt{g_{11}}} \int_{CZ} \left(\sigma^{33} \frac{\partial \delta}{\partial \theta} \cdot \right) d^{3} d^{3$$

release rates in arbitrarily shaped delamination. The volving a large fracture process zone modeled using a CZM approach. The integration paths a curved lines crossing the CZ formed according to the GDD, which is rendered taking into accept the loading state at each point. Moreover, the mode II is collinear with the GDD and mode li. 's per pendicular to it. This results in the mode directions not being constant along the integration paths. On the contrary, in LEFM approaches, mode II and mode III directions are the norma'nd tangent to the crack front, respectively. For 3D planar cracks are ribed by a rectangular Cartesian coordinate system, the work by Rigby and Aliabadi [30] and L. son [40] propose equivalent expressions for the mode-decomposed J-integrals, which are in agreement with those presented in Equation (14) in the limiting case where the length of t' e CZ ands to zero. Moreover, by limiting the integration domain to the cohesive interface, the cror committed in the decomposition of the far-field quantities due to the out-of-plane stress grad nts [2⁻, 40] is avoided, thus, allowing the integration domain of the CZ J-integrals to extend t la ge racture process zones.

Note that Equation (14) represents an expression to the word edecomposed energy

3. FE-discretized mode-decomposed CZ J-integrals

In the following, the formulation presented in Section 2 is applied in an FE ramework. The CZM used in this work, and its implementation to FE, was presented by Turcher al. in [10, 11]. Complying with the cohesive element definition, the interfacial tractions and displatment jumps are expressed in a local Cartesian coordinate system, x_i , located on the deform d miderarface, \overline{S}_{coh} , defined as the average distance between two initially coinciding points, P^- and C^+ (cf. Figure 5). The direction cosines of the local Cartesian coordinate system are the normal, \hat{C}_3 , and tangential, \hat{e}_1 and \hat{e}_2 , unit vectors to \overline{S}_{coh} . Furthermore, employing the criterion developed in [39], the local tangential coordinates can be oriented in such a way that x_1 and x_2 are the tangential and normal coordinates to the GDD, respectively.

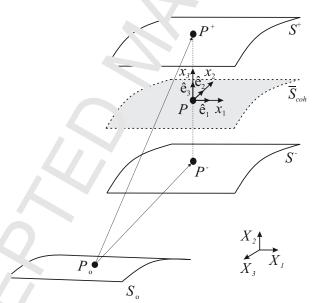


Figure 5: Description of the undeformed, S_o , and deformed, S^+ and S^- , configurations of the delamination interfaces. The quantities of the CZM are calculated at the deformed misurface, \overline{S}_{coh} , in terms of the local Cartesian coordina as P is a point located at the mid-surface in the deformed configuration, while points P^+ and P^- are points belonging to the upper and lower crack surfaces, respectively. P, P^+ and P^- coincide at P_o in the undeformed configuration.

To nu. ν rically integrate Equation (14), trapezoidal integration is employed (although any other numerical integration method could be used). Thus, the curved integration pathline is discretized into small linear subintervals tangent to the curvilinear coordinate θ^1 . The quantities in the integrand

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of Equation (14) must, therefore, be defined according to the local Cartesian coordina e system, x_i ,
with a locally coincident direction with the covariant and contravariant basis recors of the orthogonal
curvilinear coordinate system, θ^i . Further details on the discretization of the remulation with the FE
method, such as the tracking of the integration path, as well as its limits are addressed in Appendix
A.

After the discretization of the cohesive interface into FE. The numerical integration of Equation

(14), performed under the trapezoidal rule, reads:

$$J_{II} \simeq -\sum_{k} \left[h^{k} \left(\frac{\sigma_{3}^{k}}{2} \frac{\partial \delta_{3}^{k}}{2} + \sigma_{3}^{k-1} \frac{\partial \delta_{3}^{k+1}}{\partial x_{1}} \right) \right]$$

$$J_{II} \simeq -\sum_{k} \left[h^{k} \left(\frac{\sigma_{1}}{2} \frac{\partial \frac{\kappa}{1}}{2} + \sigma_{13}^{k+1} \frac{\partial \delta_{1}^{k+1}}{\partial x_{1}} \right) \right]$$

$$J_{III} \simeq -\sum_{k} \left[h \left(\frac{\sigma_{23}}{2} \frac{\partial \delta_{2}^{k}}{\partial x_{1}} + \sigma_{23}^{k+1} \frac{\partial \delta_{2}^{k+1}}{\partial x_{1}} \right) \right]$$

$$(15)$$

where h^k is the integration interval p_0 of p_0 , approximated to the Euclidean distance between two consecutive points along the integration path, P^k and P^{k+1} .

The accuracy on the cor.p. ation of the CZ J-integral depends both explicitly on the integration interval length, and implicit f on the size of the cohesive elements due to the discretization of the displacement field in the FL model.

4. Compariso with mode-decomposed energy release rates extracted by VCCT

The car abilities of the CZ *J*-integral formulation presented are assessed by comparing the energy release rate n. As components of a moment-loaded DCB model obtained by VCCT. The specimen is 30 mm lo. c, 6 mm wide and 3 mm thick (Figure 6). The elastic properties, corresponding to a unidirectional laminate made of a carbon fiber reinforced polymer (CFRP) material used in aeronautical applications, are listed in Table 1. The fracture properties of the interface are presented in Table 2.

The fracture toughnesses, \mathcal{G}_{Ic} , \mathcal{G}_{IIc} and \mathcal{G}_{IIIc} , are close to typical values for this materal. The interlaminar strengths, τ_{Ic} , τ_{IIc} and τ_{IIIc} , have been selected such that the fractu. process zone is small, to enable a fair comparison between the VCCT and the cohesive zone model wm. ensuring a minimum number of 3 damaged elements spanning the cohesive zone, to provide an accurate distribution of the tractions ahead of the crack tip [41, 42]. The specimen arms are nodeled in the commercial FE code ABAQUS [43] using C3D8I hexahedral elements. The undeformed elements are 0.4 mm wide, 0.2 mm long and 0.5 mm thick. The delamination front is completely suraigh, and located at the mid-surface at a distance of 15.1 mm from the loading application edges. A combined I, II and III fracture mode is created by applying four force pairs (Figure 6). M1 and M2 generate uneven opening Y-moments at the upper and lower arms, respectively. M3 and M4 go erate even tearing Z-moments at both arms. The resultant bending moments are listed in Table 3

Lamina e p. operties		
E_{11} : longitudinal Young's $\frac{1}{2}$	154	GPa
$E_{22} = E_{33}$: transversal Young 5 modulus	8.5	GPa
$G_{12} = G_{13}$: shear modulus in the longitudinal planes	4.2	GPa
G_{23} : shear modulus i' the ι ansversal plane	3.0	GPa
$\mu_{12} = \mu_{13}$: Poisson's refficient in the longitudinal planes	0.35	-
μ_{23} : Poisson's coefficient . ne transversal plane	0.4	-

Table 1: Elastic properties of the la.: ate ased in the simulation studies of the moment-loaded DCB and the ELS specimens.

Interface properties		
\mathcal{G}_{Ic} : mode 1 $\stackrel{\frown}{\cdot}$ cture toughness	0.3	N/mm
$\mathcal{G}_{IIc} = \mathcal{C}_{IIIc}$ modes II and III fracture toughness	3	N/mm
$ au_{Io}$: m de interlaminar strength	10	MPa
$\tau_{IIo} = \tau_{II_I}$. r.odes II and III interlaminar strengths [11]	31.62	MPa
η: Denzegragn-Kenane's interpolation parameter [44]	2	-
K: penalty stiffness	10^{5}	N/mm^3

Table 2: Fre ture properties of the interface used in the simulation study of the moment-loaded DCB specimen.

In the Fermilal Landsus using the VCCT [4], the energy release rates are evaluated locally, at every node forming the delamination front, using the nodal forces, F_i , and relative displacements between released nodes on the upper and lower crack faces, $u_i^{upper} - u_i^{lower}$:

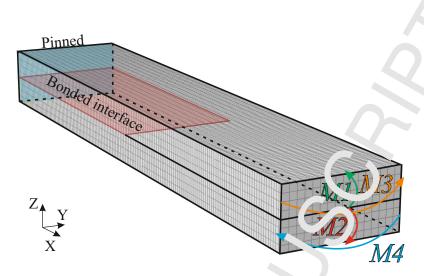


Figure 6: DCB specimen dimensions with four force pairs: 11 and 12 generate uneven opening Y-moments, while M3 and M4 generate even tearing Z-moments.

Bending moment	[Nmm]
$\overline{M1}$	270
M2	135
M3	960
M4	960

Table 3: Bending moment resultants from the application of the four force couples to the double-cantilevered-beam model.

$$\mathcal{G}_{I} = \frac{1}{2l_{1}^{e}l_{2}^{e}} F_{3} \left(u_{3}^{upper} - u_{3}^{lower} \right)
\mathcal{G}_{II} = \frac{1}{2l_{1}^{e}l_{2}^{e}} F_{1} \left(u_{1}^{upper} - u_{1}^{lower} \right)
\mathcal{G}_{III} = \frac{1}{2l_{1}^{e}l_{2}^{e}} F_{2} \left(u_{2}^{upper} - u_{2}^{lower} \right)$$
(16)

where l_i^e is the 1emen length in the *i*-direction. A local crack coordinate system, x_i with i = 1, 2, 3,
defines the regle-components, such that mode II $(x_1$ -direction) and mode III $(x_2$ -direction) are normal
and tangent all to the delamination front, respectively, and mode I $(x_3$ -direction) is normal to mode II
and III area tons. For a straight front, like the one under study, the orientation of this local coordinate
system is constant along the front and aligned with the mesh [45]. The same results are obtained using
the built-in implementation available in the commercial FE code ABAQUS [43].

To evaluate the *J*-values, the interface undergoing delamination has been modeled using userdefined cohesive elements. To this end, the method presented in [10, 11] has been enhanced with
the formulation for the numerical evaluation of the mode-decomposed CZ of integrals presented in
Appendix A. For the purpose of comparison with VCCT, a fixed CD is defined normal to the
straight delamination front.

The mode-decomposed energy release rate distributions 2' ig two width of the specimen, from both the VCCT and the CZ J-integral extraction methods, are plotted in Figure 7. Both results are in good agreement, although there are small differences at some points. However, determining which is the most accurate is not straightforward. On the one . and, in a real specimen, a damage process zone develops ahead of a crack tip, thus increasing the empliance of the specimen. Using the VCCT approach, the development of a damage process zon. I need of the crack tip is neglected. Using cohesive elements, the development of this damage process are is captured and therefore, the compliance of the specimen increases with respect to the con. Since of the VCCT specimen. On the other hand, the penalty stiffness of the cohesive law can manduce an error into the computation of the energy release rate [46], especially when the dam use process zone is not fully developed. However, it is worth noting that the initial stiffness that has been selected is very high to minimize this effect. In any case, the good agreement between both proposches validates the methodology presented here.

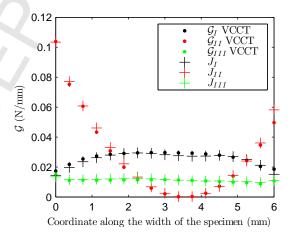


Figure 7: Comparison of the mode-components of energy release rate between VCCT and CZ J-integral extraction methods.

Furthermore, the standard formulations for VCCT require having orthogonality or ,'he mesh with the delamination front in order to obtain accurate energy release rate con. for ents [47]. Therefore, its application to three-dimensional FE models requires the option of being ab. to move meshes that conform according to the delamination front, something which is no available in commercial finite element codes [48]. Alternative solutions that enable the use of station ry meshes are presented in [49, 50]. These techniques consist of tracing a smooth virtual f. int around the stepped front. Either way, the basic assumption of these formulations is that the nodes at the delamination front will propagate along a normal vector to the current front. However, we en the delamination originates from an artificial initial defect, e.g. caused by a Teflon insert, or "hen the loading conditions change, there is a transient stage during which the shape of the crack fro. changes according to the current propagation conditions. The formulation for the evaluation of 'b GDD does not depend on the geometry of the crack front (which is historical information), but I ther on the current displacement field. Further details are given in [39]. Thus, any variatio, in the displacements due to a change in the loading scenario is captured by the GDD cri error, at the current time. Therefore, the mode-decomposition scheme according to the GDD car be at it d during transient propagation.

5. Comparison with the FFM analytical solution of a penny-shaped crack

In this section, the f rmu ation of the CZ J-integral is applied to a penny-shaped crack embedded at the centre of a ste f cylinder of 20 mm radius, f, and 20 mm height, f (c.f. Figure 8.a). The radius of the penny-shaped f crack f is 5.1 mm. A shear force, f is applied at the center of each crack face, pointing in opp site directions, as shown in Figure 8.b. The cylinder is modeled in the commercial FE code A f AQUC [43] using C3D8I hexahedral elements. Exploiting f symmetry, only one half of the specimen f codeled (c.f. Figure 8.c). The crack interface is modeled using user-defined cohesive elements f (f in the finite element implementation). The undeformed cohesive elements are 0.32 mm wide and 0.1 mm long (tangential and radial direction to the crack front, respectively). The elastic and fracture properties used in the

FE model are listed in Table 4.

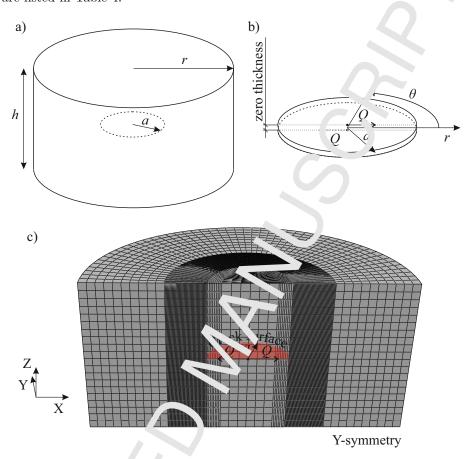


Figure 8: a) Penny-shaped crack onbe ded at the center of a cylinder. b) Detail of the penny-shapped crack with the applied shear load [38]. c) r. mouel.

Propertie	S	
E: Young's modulus	210	GPa
μ : Poisson's coefficient	0.3	-
\mathcal{G}_c : fracture toughness	11	N/mm
τ_o : strength	400	MPa
K: penalty stiffness	10^{5}	N/mm^3

Table 4: E. stic ar a fracture properties used in the simulation study of the penny-shaped crack.

The CZ integral mode II and III components computed according to Equation (15) are represented in Figure 9 together with the LEFM analytical solution of a penny-shaped crack in an infinite domain available in [38]. The mode I component is not plotted since it is negligible under these loading conditions. The represented results have been normalized by:

$$F = \frac{\left(\frac{2Q}{(\pi a)^{3/2}}\right)^2}{E} \tag{17}$$

The energy release rates extracted using the CZ *J*-integral formulation are in good agreement with those from the LEFM analytical solution [38]. However, likewise in the VCCT example presented in Section 4, the LEFM analytical solution does not take into account the development of a damage process zone ahead of the crack tip. Even though the parameters of the cohesive law have been selected such that a fair comparison with LEFM can be made (small fracture process zone), there still exist a small discrepancy between the results of the are rethods. In any case, the derivation and implementation of the proposed CZ *J*-integral accomposition scheme is validated with high accuracy.

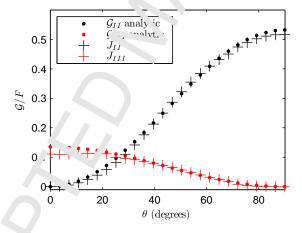


Figure 9: Comparison C the n. de-components of the energy release rate between the LEFM-based analytical solution [38] and CZ C-integral method.

337 6. Application L. partially reinforced ELS specimen

In [39], an "A Loaded split (ELS) test on a symmetric run-out specimen is presented. A Teflon insert is placed at the mid-plane of the specimen and acts as an initial delamination. A pulling displacement is applied to the cracked end of the specimen causing the two specimen beams to deflect. The test rig allows the applied displacement to be maintained in the initial direction (usually the vertical direction)

by clamping the opposite end of the specimen between rollers. Consequently, the mc ement in the horizontal direction is not constrained and axial forces are avoided. Because of this test configuration, the specimen is subjected to large deflections. Moreover, the particularity of this kind of test is that the delamination shape changes during propagation as it approaches the stiffened region created by bonded reinforcements on the upper and lower faces (cf. Figure 17). The reinforcements do not span the entire width of the specimen in order to promote a curved of lamination. As a consequence, during propagation, both the delamination front and the crack surfaces are curved. Therefore, the partially reinforced ELS specimen is considered to be suitable to exemplify the applicability of the generalized CZ J-integral methodology for 3D curved and non-planar delamination fronts.

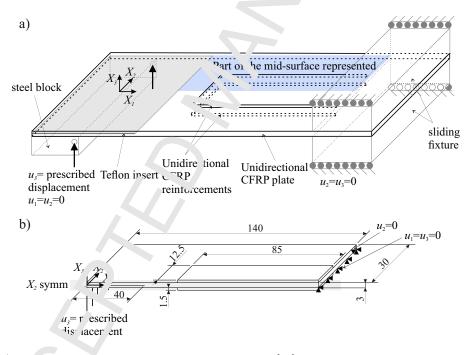


Figure 10: a) Ske ch of the partially reinforced ELS specimen [39], consisting of a CFRP plate with an initial delamination can be down a Teflon insert and two CFRP reinforcements bonded to the upper and lower faces. The grey-shadowed area is the part of the mid-surface represented in figures 11, 12 and 13. b) Simplified model for FE simulation and dimensions (units in mm).

The mage face is modeled using user-defined cohesive elements which incorporate the formulation present 1 in [10, 11], enhanced with the GDD criterion presented in [39] and the CZ *J*-integral formulation described in Appendix A. The undeformed cohesive elements are 0.27 mm wide, 0.23 mm long and have zero thickness. To reduce the computational resources required, only one half of

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Interface properties		
$\mathcal{G}_{Ic} = \mathcal{G}_{IIc} = \mathcal{G}_{IIIc}$: mode-independent fracture toughness	2	$\overline{ m V/mm}$
$\tau_{Io} = \tau_{IIo} = \tau_{IIIo}$: mode-independent interlaminar strength	<u> </u> 25	MPa
K: penalty stiffness	105	$^{\rm mm^3}$

Table 5: Fracture properties of the interface used in the simulation s udy ... he ELS specimen.

the specimen is modeled by exploiting X_2 -symmetry. The elastic properties of the laminate and the fracture properties of the interface are listed in Tables 1 and r , respectively. Note that, as a simple way to check the CZ J-integral implementation, the fracture to ghine s is set to be mode-independent ($\mathcal{G}_c = \mathcal{G}_{Ic} = \mathcal{G}_{IIc} = \mathcal{G}_{IIIc} = 2 \text{ N/mm}$) to ensure a constant J value ($J = \mathcal{G}_c$) during static crack propagation. Thus, the sum of the three mode-decomposed CZ J-integrals in Equation (15) must be constant and equal to 2 N/mm at every integration contour, regardless of the loading mode. In the following figures, only the blue-shadowed area of the nid-surface in Figure 10 is represented.

The historical evolution of the 0.5-valued energy based damage isoline is plotted in Figure 11.a. The energy-based damage, \mathcal{D}^e , distribution is projected onto the deformed mid-surface (cf. Figure 11.c) for a prescribed end displacement of \mathcal{L}^{e} rum. Note that a large fracture process zone is developed (the maximum length of the CZ is apport nately 20 mm). The GDD distribution within the CZ is represented in Figure 12. As most need in Appendix A, the CZ J-integral can be evaluated at any point within the CZ and, therefore, infinite integration paths can be tracked. For illustrative purposes, only a few selected integration paths are plotted on top of the GDD distribution. Note that the trajectory of the integration paths is established according to the GDD. Thus, since the $\frac{\omega_{tot}}{G_c}$ isolines are not parallel, the $\frac{d^2 G_c}{G_c}$ isolines throughout the CZ.

The total J- value is valuated at each of the 30,000 integration points forming the CZ. The result is represented in Figure 13.a. The step length ${}^*h^k$ used is 0.3 mm (1.3 times the element length), where the superscription of ${}^*h^k$). Note that the J-distribution is constant and equal to the fracture toughness, which, during static propagation and for any mode mixity, amounts to 2 N/mm. The total J-value computed is equal to the fracture toughness at all the integration points within the cohesive

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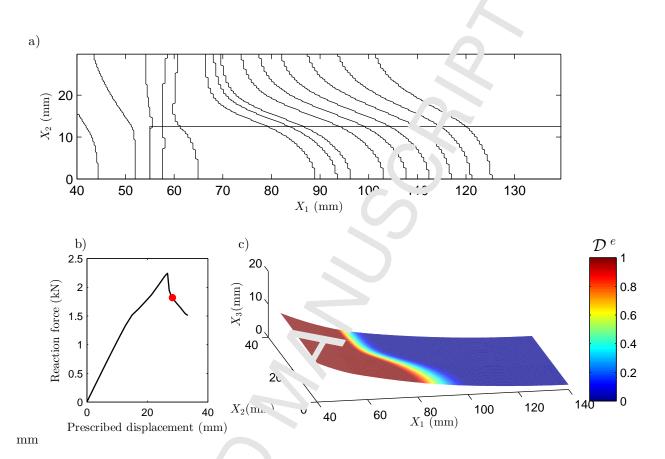


Figure 11: a) Historical evolution of the 0.5-valued energy-based damage isoline extracted at the integration points. b) Reaction force vs prescriled disposement curve with the current loading state highlighted in red. c) Energy-damage projected onto fue denormed mid-surface at the current loading stated marked in (b).

zone with a maximum erro of 3.7% (cf. Figure 13.a). By reducing $^*h^k$, more accurate results may be obtained. However, for such a large CZ, the computational cost increases significantly with the number of segments ^{j}h which the integration paths are discretized.

The decompositio. If the CZ J-integral into modes, computed according to Equation (15), is also represented in Figure 13. The mode II and III components of the CZ J-integral are predominant, while mode a slightly appears at a small region close to the specimen's edge (cf. Figure 13.b). The contribution of the J-value of the tangent quantities to the mid-surface is decomposed into modes II and III according to the GDD. The bonded reinforcements cause the loading state to be uneven throughout the specimen's width, leading to a curved crack, so that the GDD amounts to 60° with respect to the X_1 at the zones with the highest delamination front curvature. Due to the test configuration, the

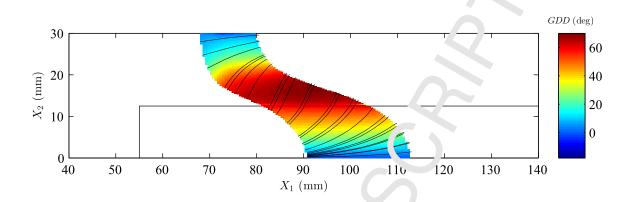


Figure 12: Growth driving direction (GDD) distribution along the con-sive zone and a few selected integration paths (black solid lines) plotted on top of it. The current load. ** state ** s marked in Figure 11.b.

maximum interlaminar shear stress is applied in the global X_1 -direction. For straight cracks where
the GDD is aligned with the X_1 -direction, the shear stress is applied in the global X_1 -direction. For straight cracks where
the GDD is aligned with the X_1 -direction, the shear stress ponent would be pure mode II. However, in
the studied case with a curved delamination that the maximum contribution of the external loading
to the mode III CZ J-integral is at the regret X_1 -direction (cf.
Figures 13.c and 13.d).

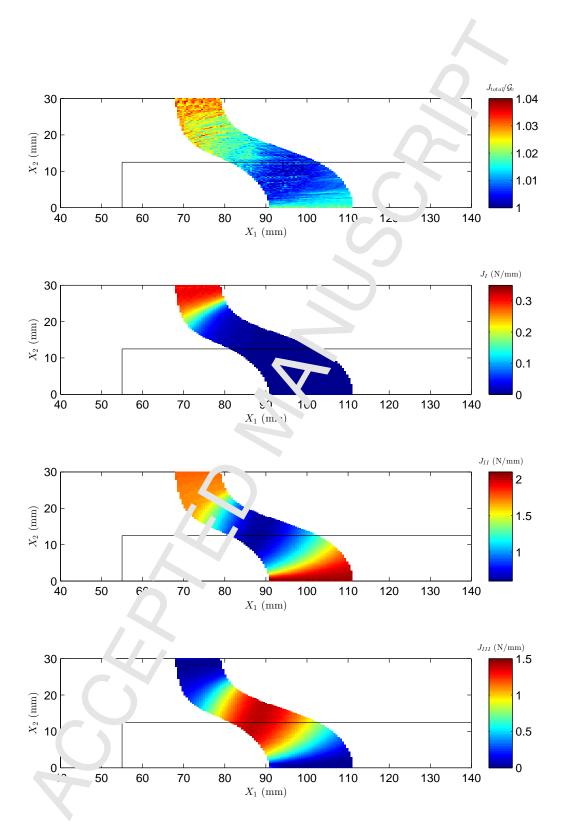


Figure 13: Distribution of a) J_{total}/\mathcal{G}_c (where $J_{total}=J_I+J_{II}+J_{III}$ and $\mathcal{G}_c=2$ N/mm), b) J_I , c) J_{II} and d) J_{III} within the cohesive zone at current loading state marked in Figure 11.b.

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7. Conclusions

A novel methodology for calculating the mode-decomposed J-integrals in the redimensional delam-ination simulation using a cohesive zone model approach is presented. The methodology incorporates the growth driving direction criterion, recently developed by the authors, thrack the integration paths and to determine the local directions of mode I, II and III compounts. The generality of the formu-lation makes it applicable to curved fronts with non-planar d lam; ... ion interfaces and large fracture process zones. The application of the described methodolog, results in curved integration paths. The calculation of the J-integral is based on dividing the dislamination interface into elemental thickness slices so that the J-value of each slice is unique. The curvature of such slices is defined according to the growth driving direction. Since the growth driving direction is mesh independent, the definition of the slices is not affected by the rash share. By applying the formulation presented here, a global measure of the energy release rate in three-dimensional structures modeled using a cohesive zone model approach can be obtained. To the authors knowledge, this has not been prevously a dressed. Furthermore, the energy release rate can be decomposed into mode I, II and III omponents. The decomposition of the shear component of the energy release rate into mode II and I.f. to date, has only been addressed under the assumption of elastic fracture mechanics. In aquition, the new formulation enables a global measure of the mode mixity to be obtained, Gereaming the limitation of the current 3D cohesive zone model formulations

The limitation of the presented formulation are related to the use of cohesive zone models, and therefore, the crick is onfined to propagate within the interface between layers. The possibility of crack migrating to nother interface is not accounted for.

where the mode mix ty; only obtained at integration point level in terms of opening displacements.

Bes: 100 the immediate applications of the formulation, the authors believe that more applications
will be unc vered in future research. The CZ *J*-integral presented here is a decisive contribution to
fracture mechanics-based procedures in a cohesive zone model framework, which will allow the design of
lighter and more reliable structures. In addition, a direct application of the CZ *J*-integral formulation

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is its implementation in combination with existing fatigue simulation methods formula ed in a CZM approach that rely on mode-dependent Paris law' like expressions. Thus, the mode-decomposed CZ Jintegral formulation developed becomes a new solution for extracting mode-decomposed energy release
rates of real complex three-dimensional structures.

422 8. Acknowledgements

This work has been partially funded by the Spanish Government Ministerio de Economia y Competitividad) under contract TRA2015-71491-R, co-finan ed by the European Social Fund.

Using Equation (15), the mode-decomposed CZ -integrals, which may vary for every slice, can be

Appendix A. Discretization with the FE method

evaluated everywhere within the CZ. Moreover, any point within the CZ belongs to a single slice, i.e. to a single integration path. The integration path is are defined according to the local GDD. Therefore, one can randomly select any location of the "Z and, by means of the GDD, identify the tangent to the integration pathline at that point in order to move, either forward or backward, along the integration path. The mode-decomposed CZ , int grals corresponding to such slice are obtained when the path is tracked in its entirety. The procedure for the evaluation of the mode-decomposed CZ J-integral of Equation (15) is shown in Figure A.15 and c escribed in the following. Consider a point, P^k , belonging to the CZ. In order to assess the mode-docomposed CZ J-integrals at the slice which the point P^k belongs to, the numerical integration of E vaction (14) is performed along the integration path, defined as tangent to the local GDD direction and limited by vanishing stress conditions at both ends (cf. Figure 3). In the general case, the initial point P^k is not located at one end of the integration path, i.e. point P^k is located in the middle of the CZ. In this case, the path will be tracked from P^k in the GDD (Loop 1 in Figure A.15) and in the opposite direction to the GDD (Loop 2 in Figure A.15). In other words, in the positive GDD until vanishing elastic stress is reached (point 2 in Figure 3), while in the negative GDD

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until the intersection with the 1-valued energy-based damage isoline, where the cohesive stress also equals zero (point 1 in Figure 3). The condition for vanishing cohesive stress as reads:

$$\mu < tol$$
 (A.1)

where μ is the norm of the cohesive stresses and tol is a user-defined thre hold close to zero.

To move along the integration path, the following procedure is $_{*}p_{1}$ 'ied. Starting from P^{k} , the next point along the integration path is established by moving in straight line a $^{*}h^{k}$ -length step further in the local GDD, which is tangent to the cohesive interface \bar{S}_{coh} , at P^{k} . Then, a new point, $^{*}P^{k+1}$, in the space is found. Nevertheless, $^{*}P^{k+1}$ is not necessarily placed on the mid-surface, \bar{S}_{coh} . This becomes evident when \bar{S}_{coh} is highlar planar (cf. Figure A.14). Thus, the real next point constituting the integration path, P^{k+1} is found by projecting $^{*}P^{k+1}$ on \bar{S}_{coh} in the normal x_{3} -direction of point P^{k} .

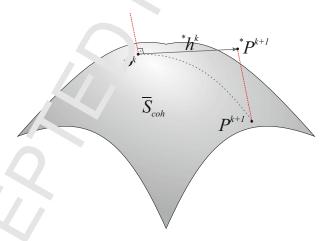


Figure A.14: Point P^k is a point on the integration path of a curved cohesive interface, \bar{S}_{coh} . The following point on the integration path, P^{k+1} , is found by projecting point P^{k+1} along the normal direction to the interface at P^{k+1} is at an P^{k+1} is at an P^{k+1} in the tangential GDD.

The integrands in Equation (15), σ_{ii} and $\frac{\partial \delta_i}{\partial x_1}$, are evaluated at every point P^k along the integration path. σ_{ii} a 2 the components of the cohesive stress tensor expressed according to the local Cartesian coordinate system. On the other hand, the derivative of the displacement jumps, δ_i , with respect to the local Cartesian coordinate aligned with the GDD, x_1 , is addressed in the following. X_j is the

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Cartesian reference system, x_i is the local Cartesian coordinate system and R_{ii} is the consormation tensor which relates the global to the local coordinate system. The derivation of the rotation matrix, R_{ij} , with respect to the coordinate x_1 can be approximated to zero by assuming that the curvature of the interface within the integration subinterval is small. This is activated by setting an $*h^k$ -length step similar to the element length. Moreover, its derivation would increase the complexity of the formulation without a substantial improvement in the accuration of the solution. Thus, by assuming that the derivative of R_{ij} with respect to x_1 can be omitted, the derivative $\frac{\partial \delta_i}{\partial x_1}$ reads:

$$\frac{\partial \delta_i}{\partial x_1} = R_{ij} \frac{\partial M_{j}}{\partial x_1} \mathcal{O}_{r} \tag{A.2}$$

where M_{jm} is the transformation matrix that i. ... the global displacement jump with the nodal global displacement, Q_m . The size of Q_m is ... number of degrees of freedom of the element (in the case of 8-noded cohesive elements, $m = \frac{1}{2} \frac{1}{2} \frac{1}{2} \frac{1}{2}$. The derivative of the transformation matrix, M_{jm} , with respect to the local coordinate, x_1 , is obtained by applying the chain rule:

$$\frac{\partial I_{jm}}{\partial x_1} = \frac{\partial M_{jm}}{\partial \eta_{\alpha}} \frac{\partial \eta_{\alpha}}{\partial x_1} \tag{A.3}$$

The first partial derivative is the right hand side of Equation (A.3) is the variation of the transformation matrix, M_{jm} with the isoparametric coordinates of the cohesive element formulation, η_{α} (α =1,2):

$$\frac{\partial M_{jm}}{\partial \eta_{\alpha}} = \left[-\frac{\partial N_{jk}}{\partial \eta_{\alpha}}, \frac{\partial N_{jk}}{\partial \eta_{\alpha}} \right] \tag{A.4}$$

where N_{jk} s the slape function matrix and the subscript k runs from 1 to the number of degrees of freedon real actively, of the top and bottom surface of the cohesive element. In the case of an 8-noded element, k = 1...12. In [10, 11], the material coordinates and the displacement fields are interpolated within the domain of the interface element using isoparametric bilinear shape functions:

$$L_{1} = \frac{1}{2} (1 - \eta_{1}) (1 - \eta_{2}); \qquad L_{2} = \frac{1}{2} (1 + \eta_{1}) (1 - \zeta_{1})$$

$$L_{3} = \frac{1}{2} (1 + \eta_{1}) (1 + \eta_{2}); \qquad L_{4} = \frac{1}{2} (1 - \eta_{1}) (1 + \eta_{2})$$
(A.5)

organized in N_{jk} as follows:

$$N_{jk} = \begin{bmatrix} L_1 & 0 & 0 & L_2 & 0 & 0 & L_3 & 0 & 0 & l_4 & 0 & 0 \\ 0 & L_1 & 0 & 0 & L_2 & 0 & 0 & L_3 & 0 & 0 & L_4 & 0 \\ 0 & 0 & L_1 & 0 & 0 & L_2 & 0 & 0 & L_3 & 0 & 0 & L_4 \end{bmatrix}$$
(A.6)

where the local isoparametric coordinates, η_1 and η_2 , 1. The from -1 to 1 over the element domain.

The derivatives $\frac{\partial \eta_{\alpha}}{\partial x_1}$ are the inverse of the derivatives of the local coordinate, x_1 , with respect to the isoparametric coordinates, η_{α} , defined as:

$$\frac{\partial x_1}{\partial \eta_{\alpha}} = \iota_{1j} \frac{1}{2} \frac{\partial N_{jk}}{\partial \eta_{\alpha}} \left(C_k^+ + C_k^- + Q_k^+ + Q_k^- \right)$$
 (A.7)

where C_k^- and C_k^+ are the global coordinates of the nodes at the lower and upper surfaces, and Q_k^- and

 Q_k^+ are the nodal displaceme its, relative to the global coordinates, of the lower and upper surfaces.

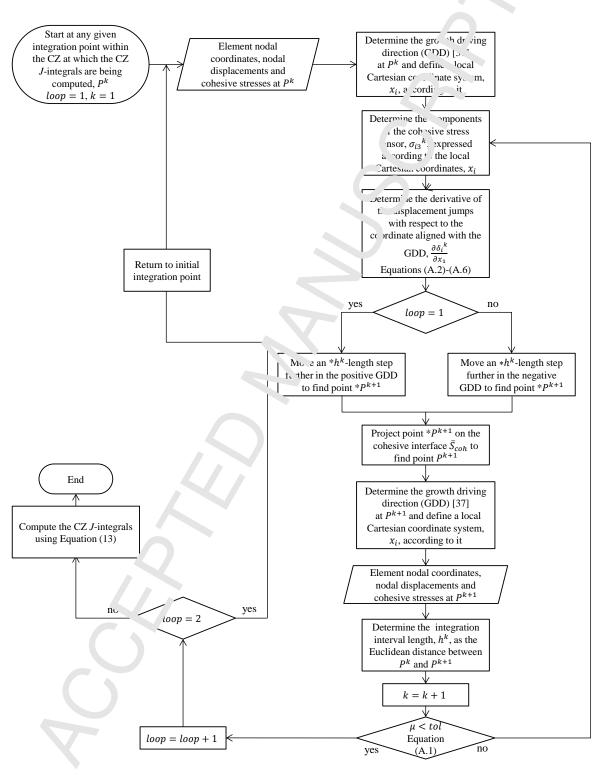


Figure A.15: Flow chart of the calculation of the CZ J-integrals at a given point within the cohesive zone discretized with the FE method.

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